

# Introduction to small-angle scattering –the use of X-rays and Neutrons

A practical guide with examples to highlight specific advantages

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FASEM School

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**HELMHOLTZ** RESEARCH FOR  
GRAND CHALLENGES



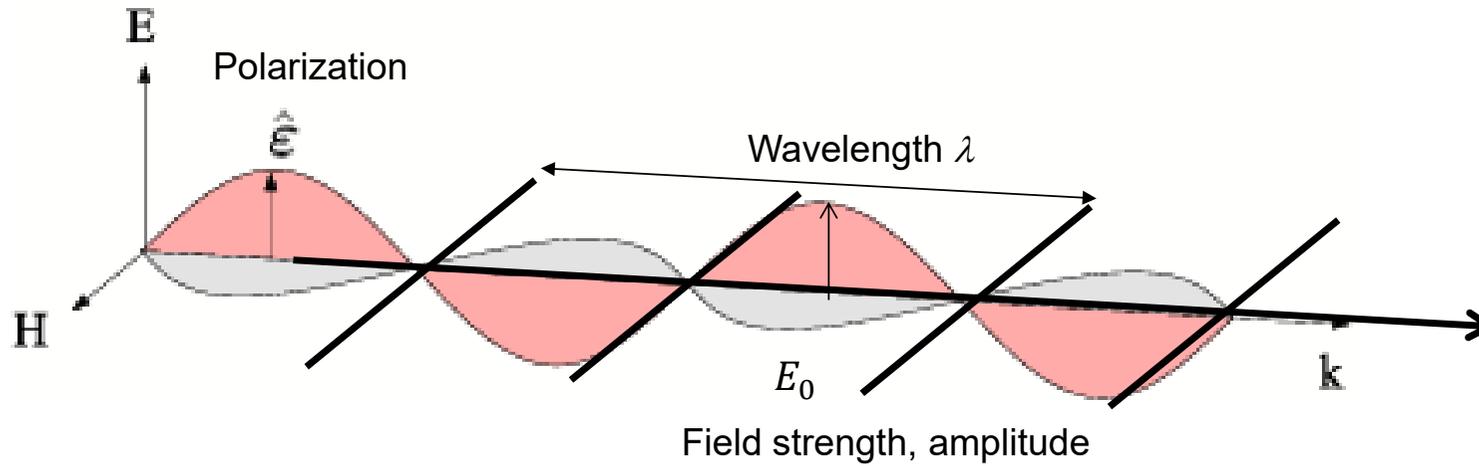
# Outline

## Introduction to small-angle scattering – the use of X-rays and Neutrons

- **Basics SAXS (hold ~SANS)**
- SAXS approximations (holds for SANS)
- Example I – Aerogels
- Example II – Kinetics of colloidal droplet drying
- Example III – Chocolate
- Example IV – Microfluidics
- Example V – Superalloys
- Example VI – Contrast variation using X-rays: Iron nanoparticles in metal hydrid matrix
- Neutron scattering
- Example VII– Contrast variation using deuteration and D<sub>2</sub>O

# Photons

- Type equation here. Electromagnetic wave

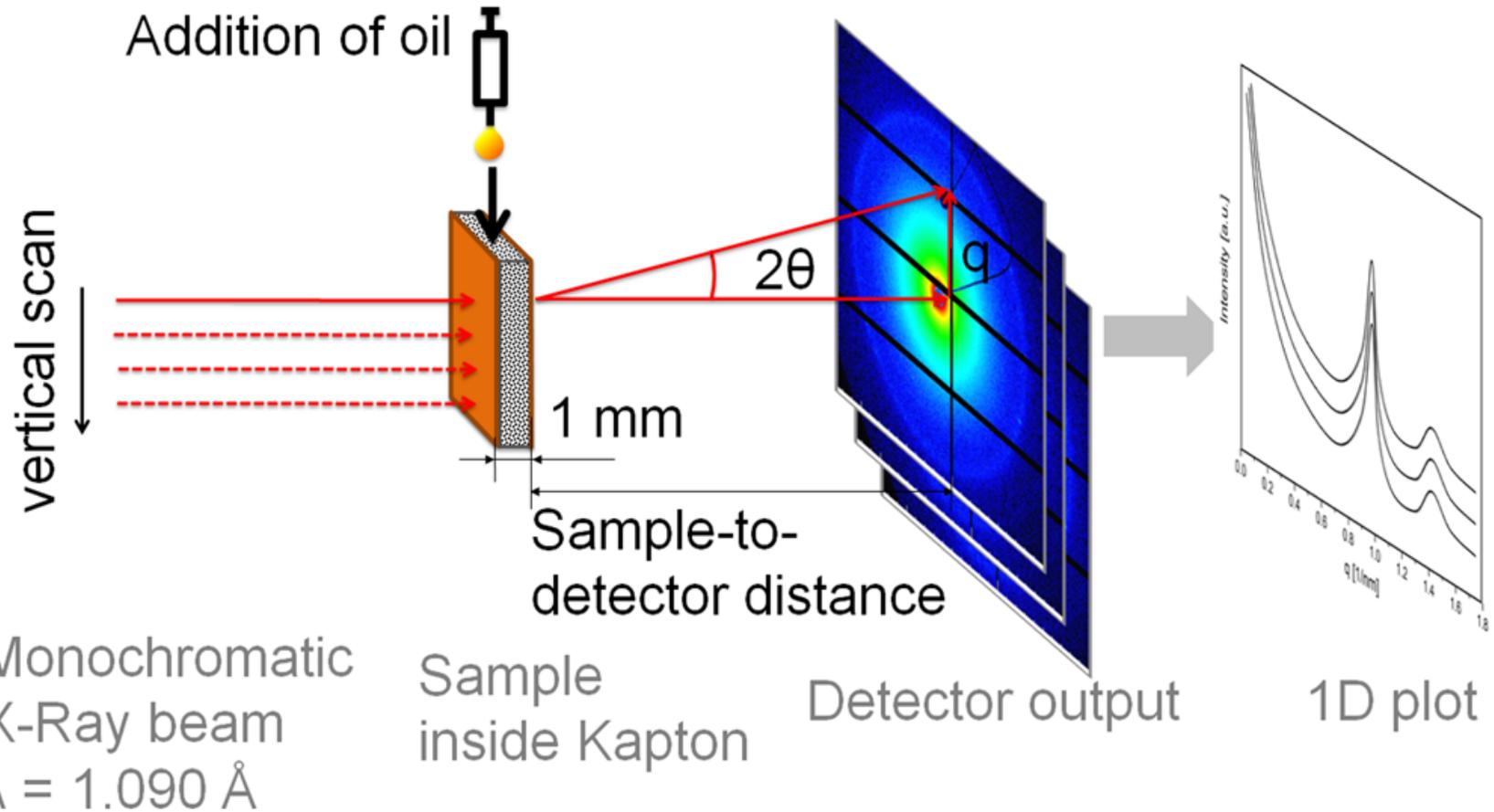


$$\mathbf{E} = \hat{\epsilon} E_0 \exp(i(\mathbf{k}\mathbf{r} - \omega t))$$
$$\mathbf{p} = \hbar \mathbf{k}$$
$$|k| = \frac{2\pi}{\lambda}$$

Als-Nielsen, McMorrow, "Elements of modern X-ray Physics", Wiley, 2010

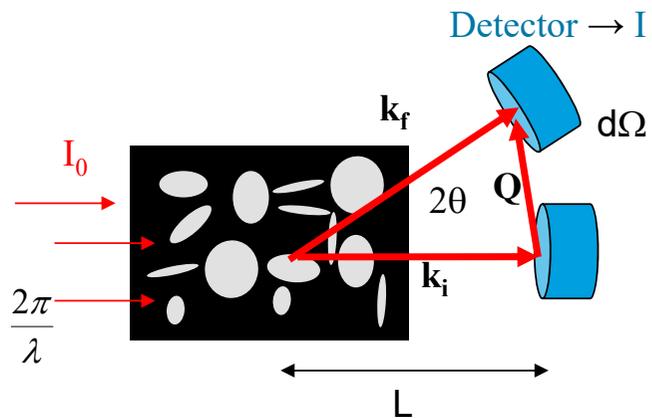
# Chocolate

A real multicomponent system



# Cross-section

- Differential cross section



$$d\sigma = \frac{I}{I_0} (L^2 d\Omega)$$

$$\frac{d\sigma}{d\Omega} = \frac{I}{I_0} (L^2) \Rightarrow \frac{d\Sigma}{d\Omega} = \frac{1}{V} \frac{d\sigma}{d\Omega}$$

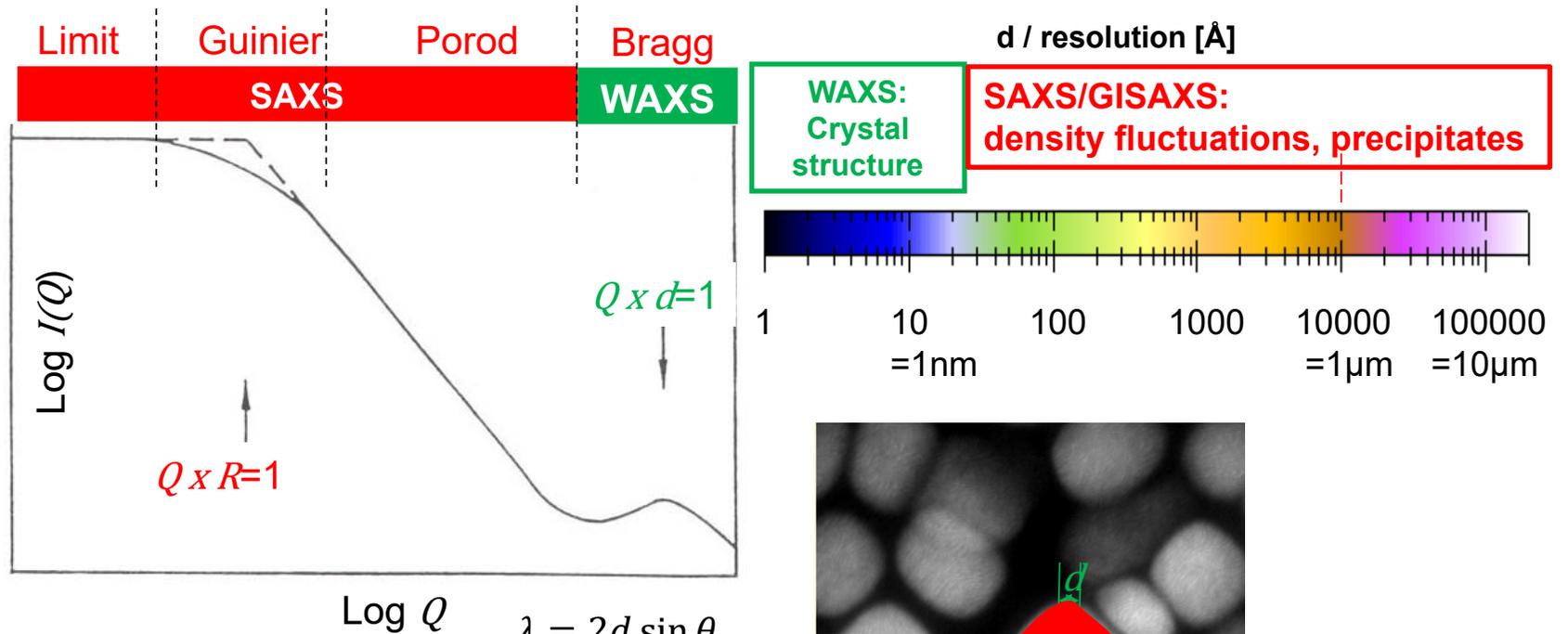
$V$  = Sample volume

$$\vec{Q} = \vec{k}_f - \vec{k}_i \quad |\vec{k}_f| = |\vec{k}_i| = \frac{2\pi}{\lambda} \quad |\vec{Q}| = 2 \frac{2\pi}{\lambda} \sin(\theta)$$

- Scattering occurs due to density differences

# WAXS, SAXS, GISAXS

Source: Streumethoden zur Untersuchung kondensierter Materie  
1996; ISBN 978-3-89336-180-9



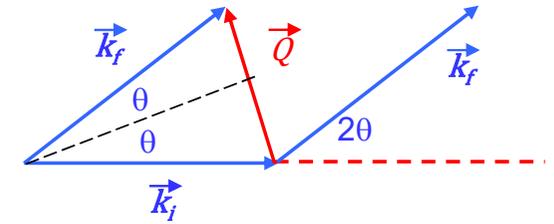
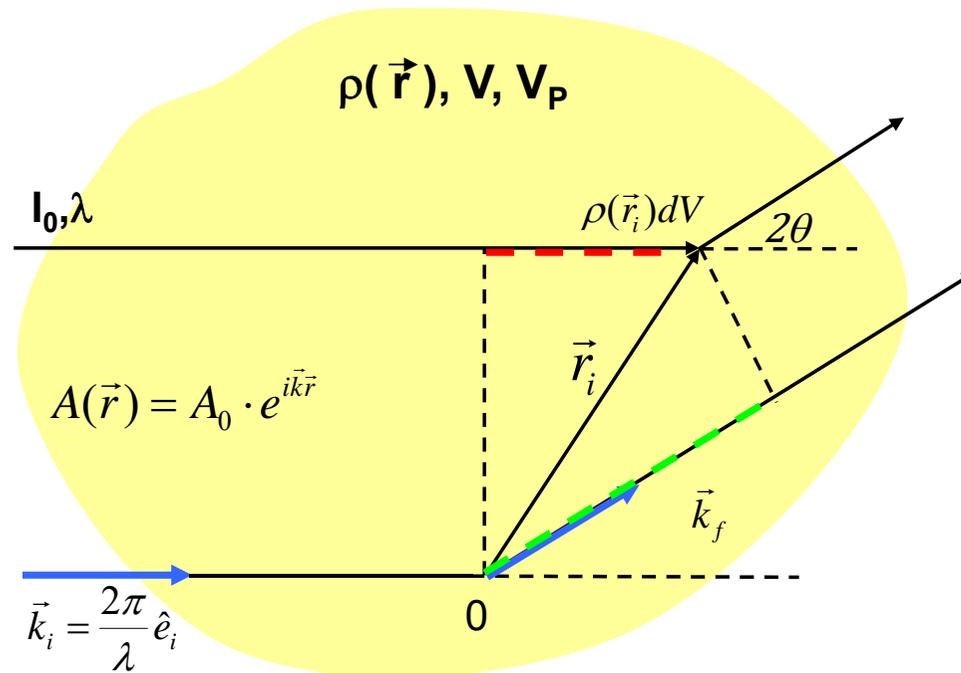
- $R$ ~particles "radius"
- $d$ ~interatomic distance
- SAXS:  $\theta < 5^\circ$

$$\lambda = 2d \sin \theta$$

$$\lambda = 1.54 \text{\AA}$$

# Scattering Amplitude

- Interference in far field



$$|\vec{Q}| = \frac{4\pi}{\lambda} \sin \theta$$

- Phase difference:

$$\Delta\varphi_i = (\vec{k}_f - \vec{k}_i) \cdot \vec{r}_i = \vec{Q} \cdot \vec{r}_i$$

- Scattering amplitude:

$$A(\vec{Q}) = \int \rho(\vec{r}) e^{-i\vec{Q}\vec{r}} dV = \int \rho(\vec{r}) e^{-i\vec{Q}\vec{r}} d^3\vec{r}$$

- Intensity:

$$I(\vec{Q}) = \frac{1}{V} |A(\vec{Q})|^2$$

# Form factor and structure factor: Fourier transform

Single particle: Fourier transformation

$$A(\vec{Q}) = \int \rho_P(\vec{r}) e^{-i\vec{Q}\vec{r}} dV = \int \rho_P(\vec{r}) e^{-i\vec{Q}\vec{r}} d^3\vec{r}$$

Particle distribution function  $G(\mathbf{r}) \rightarrow$  Electron density distribution

$$\rho(\vec{r}) = \sum_i \rho_P(\vec{r}_i) = \int \rho_P(\vec{r}') G(\vec{r} - \vec{r}') d^3r' = \rho_P(\vec{r}) * G(\vec{r})$$

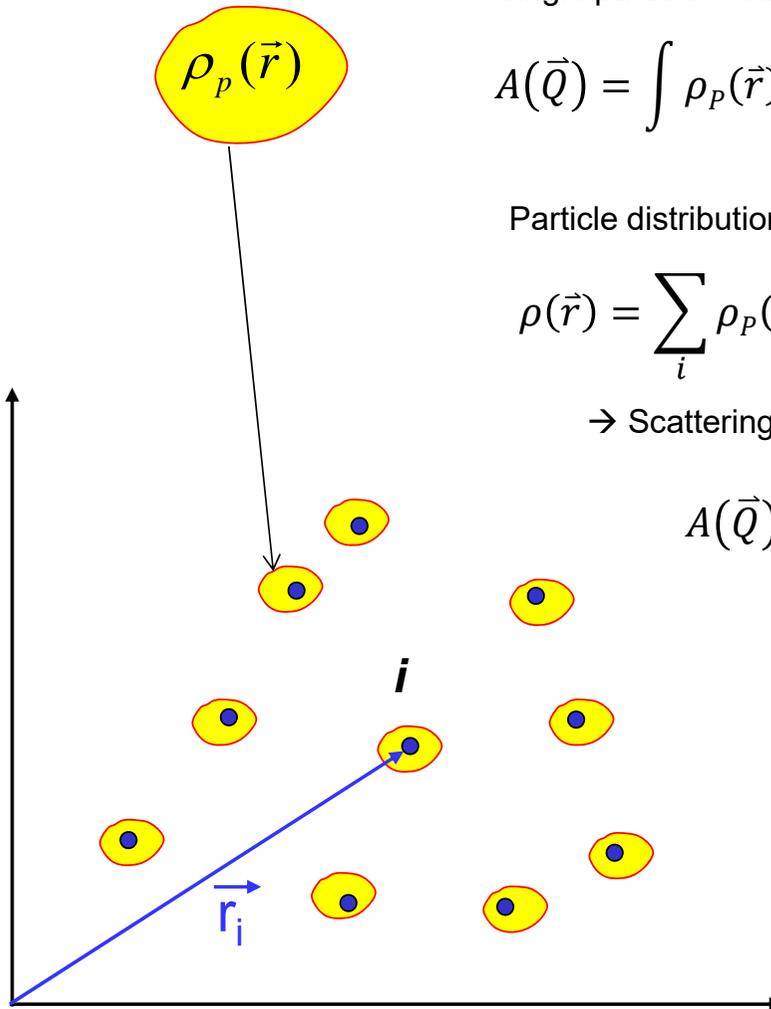
$\rightarrow$  Scattering amplitudes of the whole arrangement

$$\begin{aligned} A(\vec{Q}) &= \int \rho(\vec{r}) e^{-i\vec{Q}\vec{r}} dV = \int [\rho_P(\vec{r}) * G(\vec{r})] e^{-i\vec{Q}\vec{r}} d^3\vec{r} \\ &= \int \rho_P(\vec{r}) e^{-i\vec{Q}\vec{r}} dV \cdot \int G(\vec{r}) e^{-i\vec{Q}\vec{r}} dV \end{aligned}$$

$\rightarrow$  Scattered Intensity

$$I(\vec{Q}) = \frac{1}{V} |A(\vec{Q})|^2 = P(\vec{Q}) S(\vec{Q})$$

Form factor    Structure factor

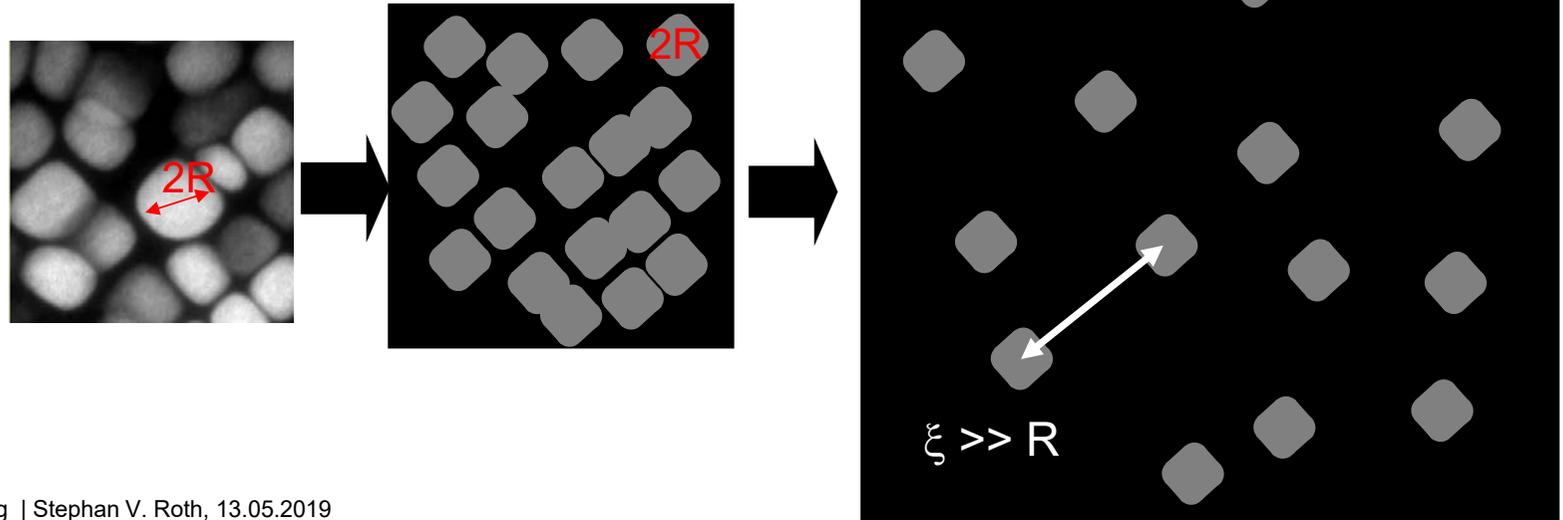


# Two-phase model: Dilute systems

- > Only form of particle relevant
- > Matrix  $M$ , volume fraction  $\Phi$   
Particles  $P$ , volume fraction  $(1-\Phi)$   
Electron density:  $\rho_{M,P} = n_{M,P} * f_{M,P}$

$f_{M,P}$ : atomic form factor („extension of the electron cloud“, resonances)  
 $n_{M,P}$ : number density of atoms

- > Consider  $\rho_{M,P}$  as constant resp.



# Two phase Model

- Scattering amplitude:

$$A(\vec{Q}) = \int \rho(\vec{r}) e^{-i\vec{Q}\vec{r}} d^3\vec{r} = \int_{\Phi V} \rho_M(\vec{r}) e^{-i\vec{Q}\vec{r}} d^3\vec{r} + \int_{(1-\Phi)V} \rho_P(\vec{r}) e^{-i\vec{Q}\vec{r}} d^3\vec{r}$$

$$A(\vec{Q}) = (\rho_M - \rho_P) \int_{\Phi V} e^{-i\vec{Q}\vec{r}} d^3\vec{r}$$

$$A(\vec{Q}) = \Delta\rho \int_{\Phi V} e^{-i\vec{Q}\vec{r}} d^3\vec{r}$$

- $I(\vec{Q}) = \frac{1}{V} |A(\vec{Q})|^2 \sim \Delta\rho^2$
- Porod Invariant  $Q$  (Porod, 1982):

$$Q = \int I(\vec{Q}) d^3Q = 4\pi\Phi(1-\Phi)\Delta\rho^2$$

- Only dependent on density contrast  $\Delta\rho$

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- Example I – Aerogels
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# Two phase Model – single particle approximation

- > Amplitude:  $A(\vec{Q}) = \Delta\rho \int_{\Phi_V} e^{-i\vec{Q}\vec{r}} d^3\vec{r}$
- > Intensity:  $I(\vec{Q}) = \frac{1}{V} |A(\vec{Q})|^2$
- > Closer look at  $I(q)$  for dilute systems:  $N_P$  independent scatterers
- > Incoherent sum of intensities:

$$I(\vec{Q}) \sim NP V_P^2 \Delta\rho^2 \left| \frac{1}{V_P} \int_{V_P} e^{-i\vec{Q}\vec{r}} d^3r \right|^2$$

$n_M f_M = \rho_M$        $n_P f_P = \rho_P$

$V_P \sim R^3$

$\Delta\rho$

$n_P f_P$

$n_M f_M$

## Two phase Model – single particle approximation

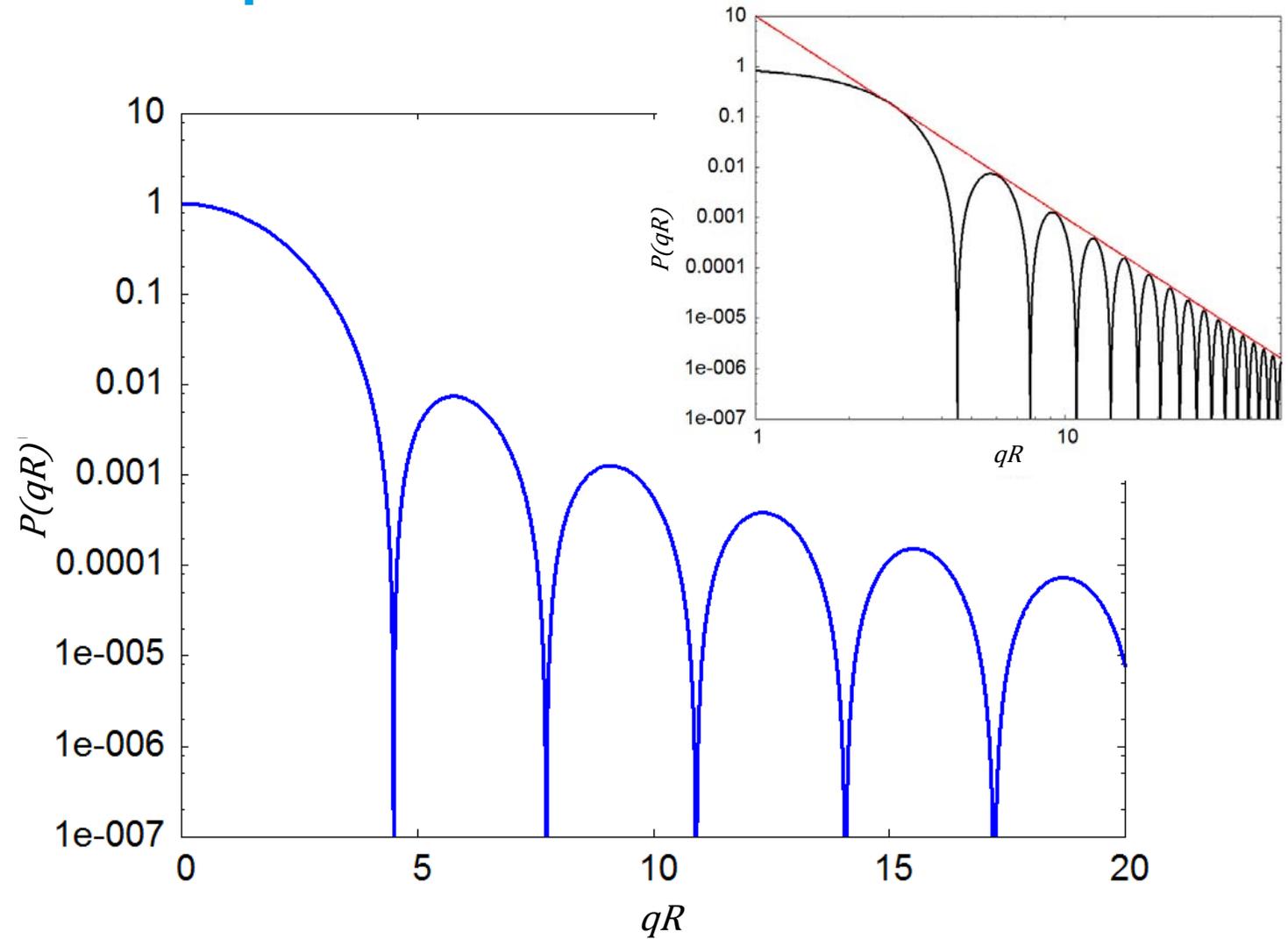
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- > Closer look at  $I(Q)$  for dilute systems:  $N_P$  independent scatterers
- > Incoherent sum of intensities:

$$I_m(\vec{Q}) \sim N_P V_P^2 \Delta\rho^2 \underbrace{\left| \frac{1}{V_P} \int_{V_P} e^{-i\vec{Q}\vec{r}} d^3r \right|^2}_{P(Q)}$$

$$P(Q) = \left| 3 \frac{\sin(QR) - QR \cos(QR)}{(QR)^3} \right|^2$$

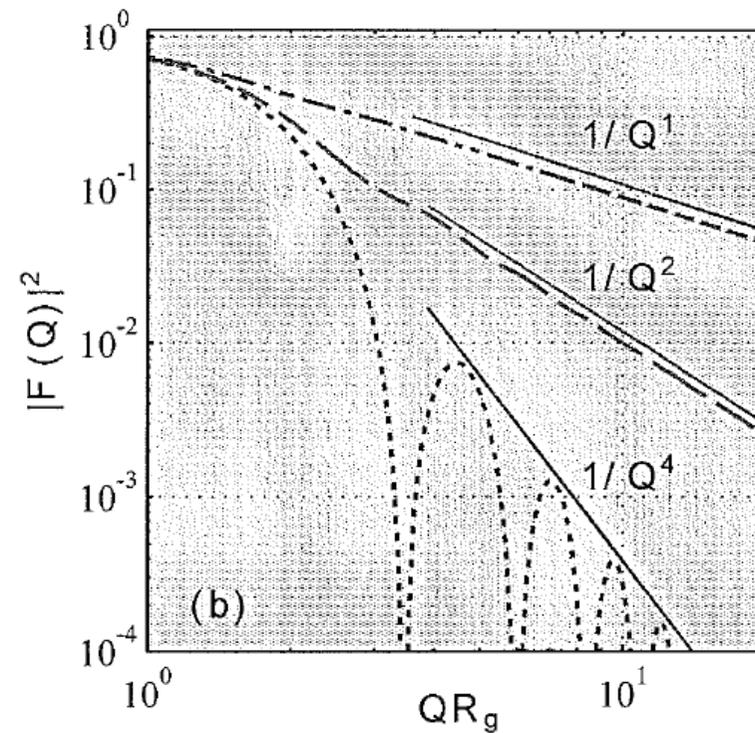
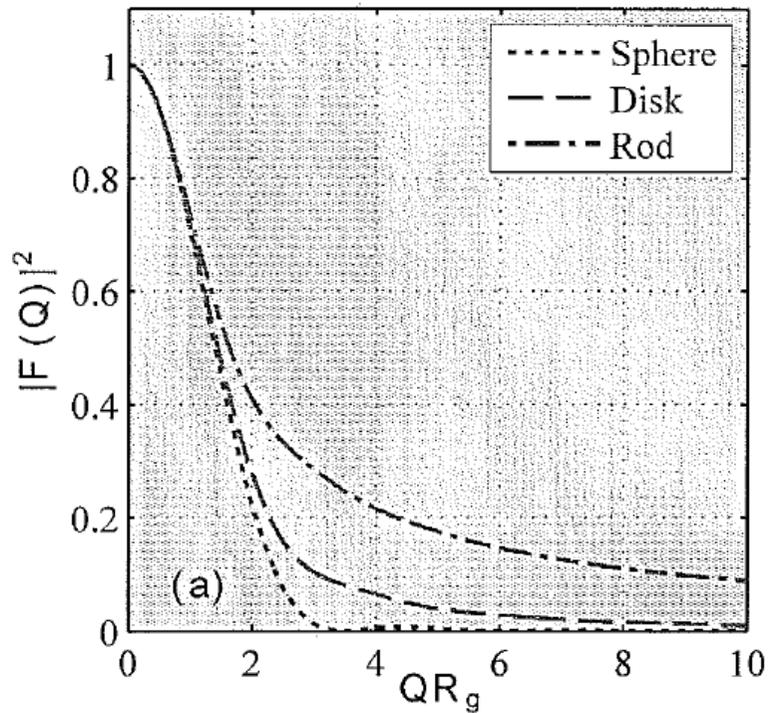
- Form factor of a **sphere of radius  $R$**
- Isotropic scattering

# Colloid: homogeneous sphere of radius $R$



# Comparison of SAXS pattern

- Influence of different shapes



Als-Nielsen, McMorrow, "Elements of modern X-ray Physics", Wiley, 2010

# Guinier Approximation

- $Q \rightarrow 0$
- Homogenous sphere of radius  $R$

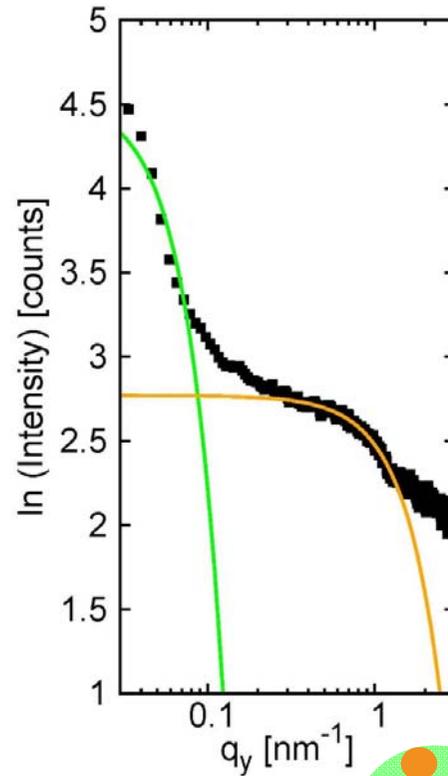
$$P(Q) = \left| 3 \frac{\sin(QR) - QR \cos(QR)}{(QR)^3} \right|^2 \sim 1 - \frac{1}{5} Q^2 R^2 \sim \exp\left(-\frac{1}{5} Q^2 R^2\right)$$

- Radius of gyration:  
rms distance from the particle's center of gravity:  $R_g$

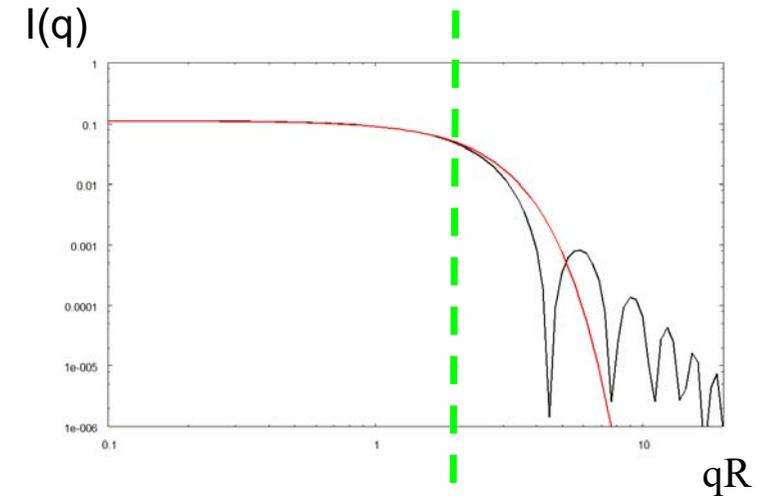
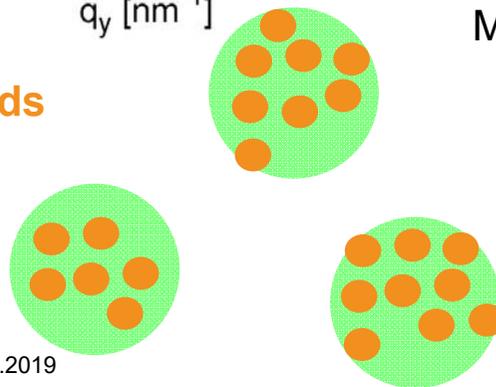
$$R_g^2 = \frac{\int_{V_p} \rho(r) r^2 dV}{\int_{V_p} \rho(r) dV}$$

- Sphere  $R_g = \sqrt{3/5} R$
- $P(q) \sim \exp\left(-\frac{1}{3} q^2 R_g^2\right)$  general form of Guinier law [Guinier (1955)]
- Independent of particle form

# Guinier Approximation



2nm Colloids  
domains



$$\lim_{q \rightarrow 0} I(q) = \Delta\rho^2 \cdot V^2 \cdot \exp(-q^2 \cdot \frac{R_g^2}{3})$$

Radius of Gyration  $R_g$

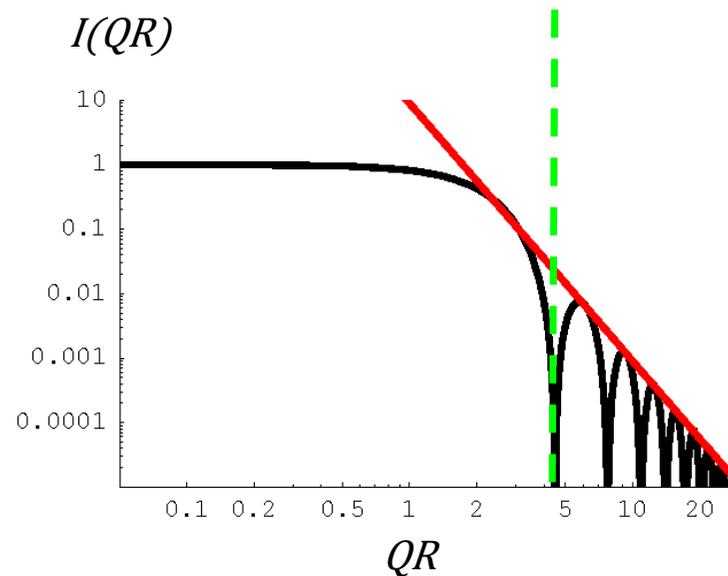
Monodisperse spheres of radius  $R$ :  $R_g = \sqrt{3/5} \cdot R$

# Porod's law: large $q$

Scattered intensity:  $\sim \left| 4\pi R^3 \rho_0 \frac{(\sin(qR) - qR \cos(qR))}{(qR)^3} \right|^2$

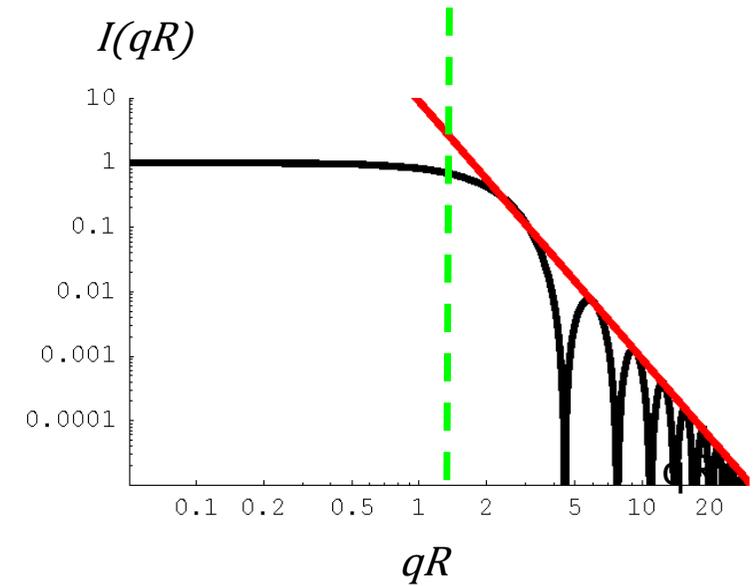
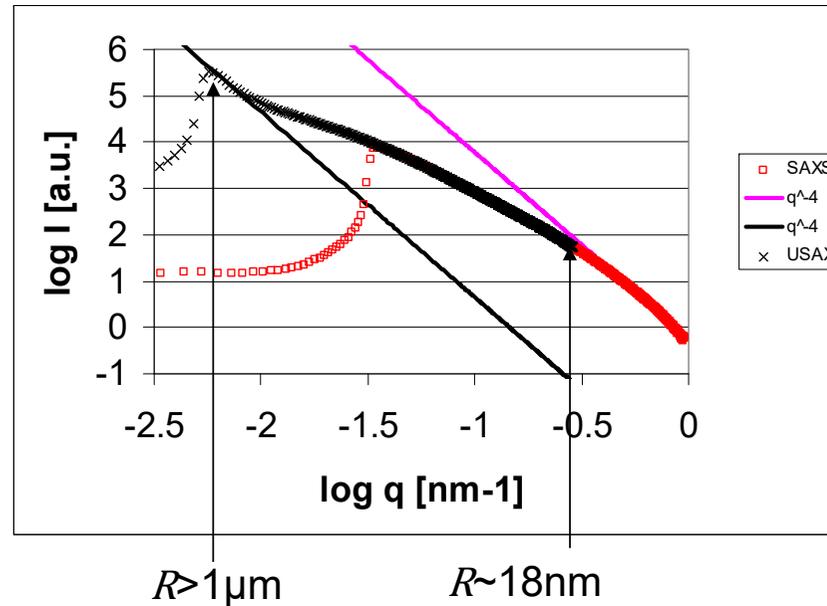
Look at maxima of form factor

$$\begin{aligned} &\sim \left| 4\pi \rho_0 \frac{(\sin(qR) - qR \cos(qR))}{(qR)^3} \right|^2 \\ &\leq \left( 4\pi \rho_0 \frac{|\sin(qR)| + qR |\cos(qR)|}{(qR)^3} \right)^2 \\ &\sim \left( 4\pi \rho_0 \frac{1 + qR}{(qR)^3} \right)^2 \sim \left( 4\pi \rho_0 \frac{qR}{(qR)^3} \right)^2 \\ &\sim \frac{1}{(q)^4} \frac{R^2}{R^6} \sim \frac{S}{V_P^2} q^{-4} \end{aligned}$$



Surface of sphere

# Porod's law: large q



$$P(qR > 4.5) = 2\pi \left( \frac{S}{V_P^2} \right) q^{-4}$$

- > Depends only on Surface and particle Volume
- > No shape dependence

# The structure factor – many particles, close distance

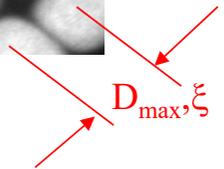
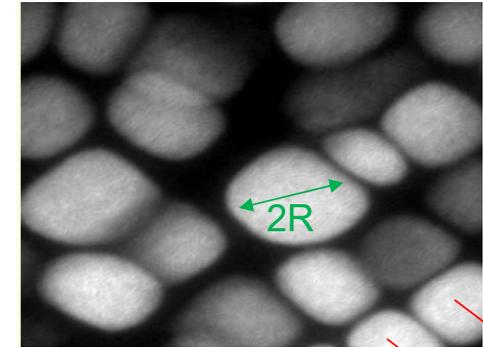
- Real systems: not dilute, many particles...
- Generalisation of Bragg's Law in crystallography:

$$I(q) = c P(q) S(q)$$

Form factor

Structure factor

Interference due to assembly of particles



- Periodic ordering with periodicity  $d, \xi$  in the electron density :
- $I(q)$  shows a corresponding maximum at  $q = 2\pi / (D_{max}, \xi)$

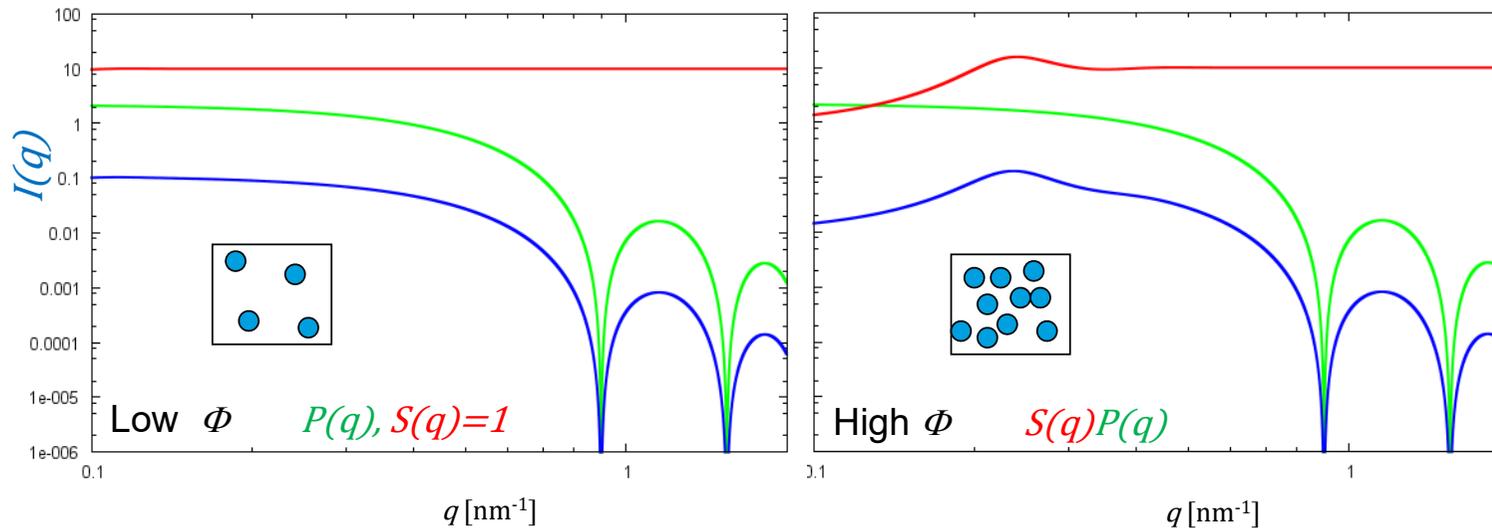
$$S(q) \propto \frac{1 - \exp(-2\sigma_D^2 q^2)}{1 - 2 \exp(-\sigma_D^2 q^2) \cos(qD_{max}) + \exp(-2\sigma_D^2 q^2)}$$

Smearing

Distance of particles

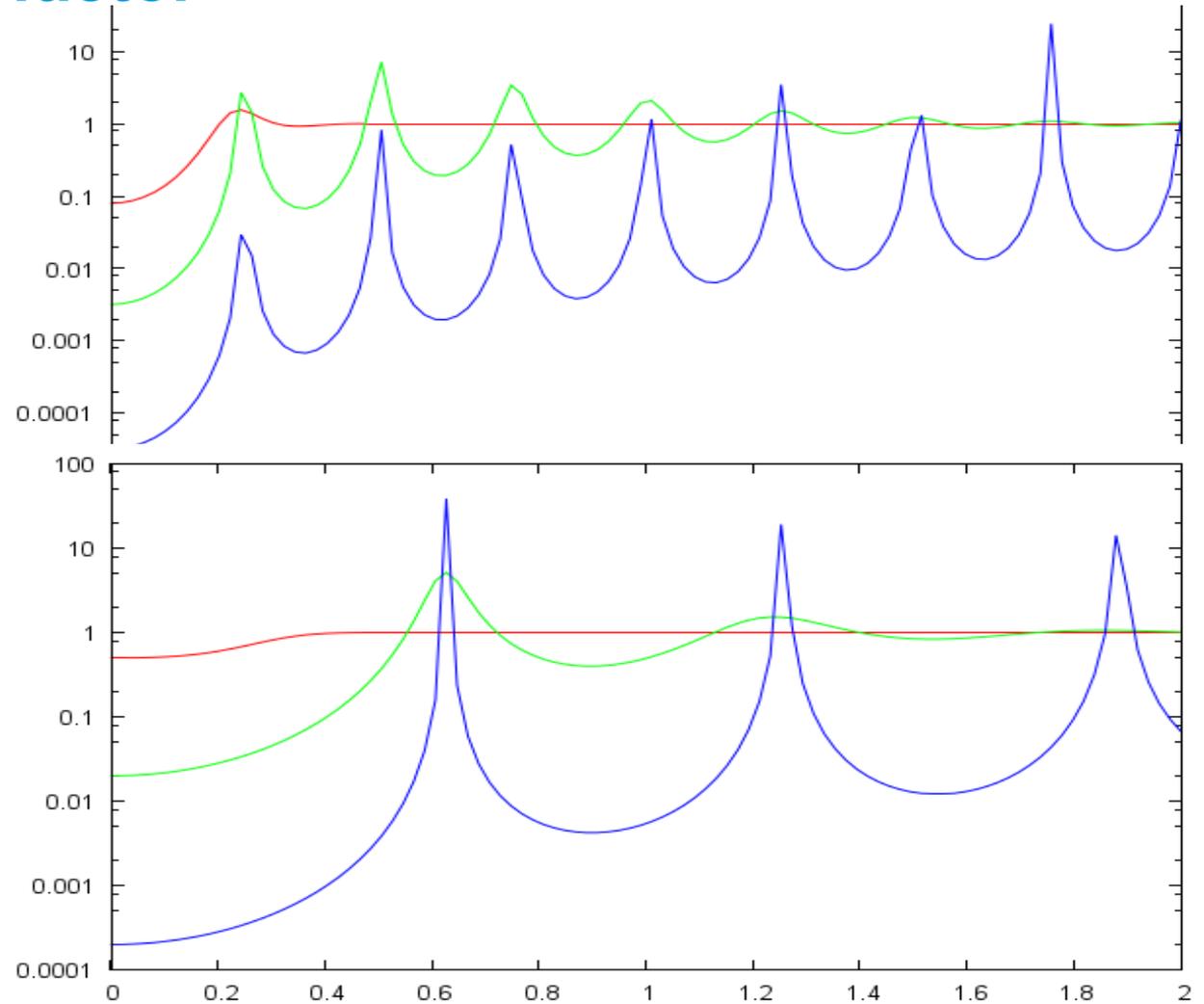
# The structure factor – many particles, close distance

- Real systems: not dilute, many particles...
- Generalisation of Bragg's Law in crystallography:  
 $I(q) = c P(q) S(q)$
- Examples:  $R=5\text{nm}$ ,  $D_{max}=100\text{nm}$ ,  $25\text{nm}$ ,  $\sigma_D/D_{max}=25\%$



# Structure factor and form factor

- $D_{max}=25\text{nm}$   
 $D_{max}=10\text{nm}$
- $\sigma_D = 5\text{nm}, 1\text{nm}, 0.1\text{nm}$
- $S(q) \rightarrow 1 \quad q \rightarrow \infty$   
well separated particles

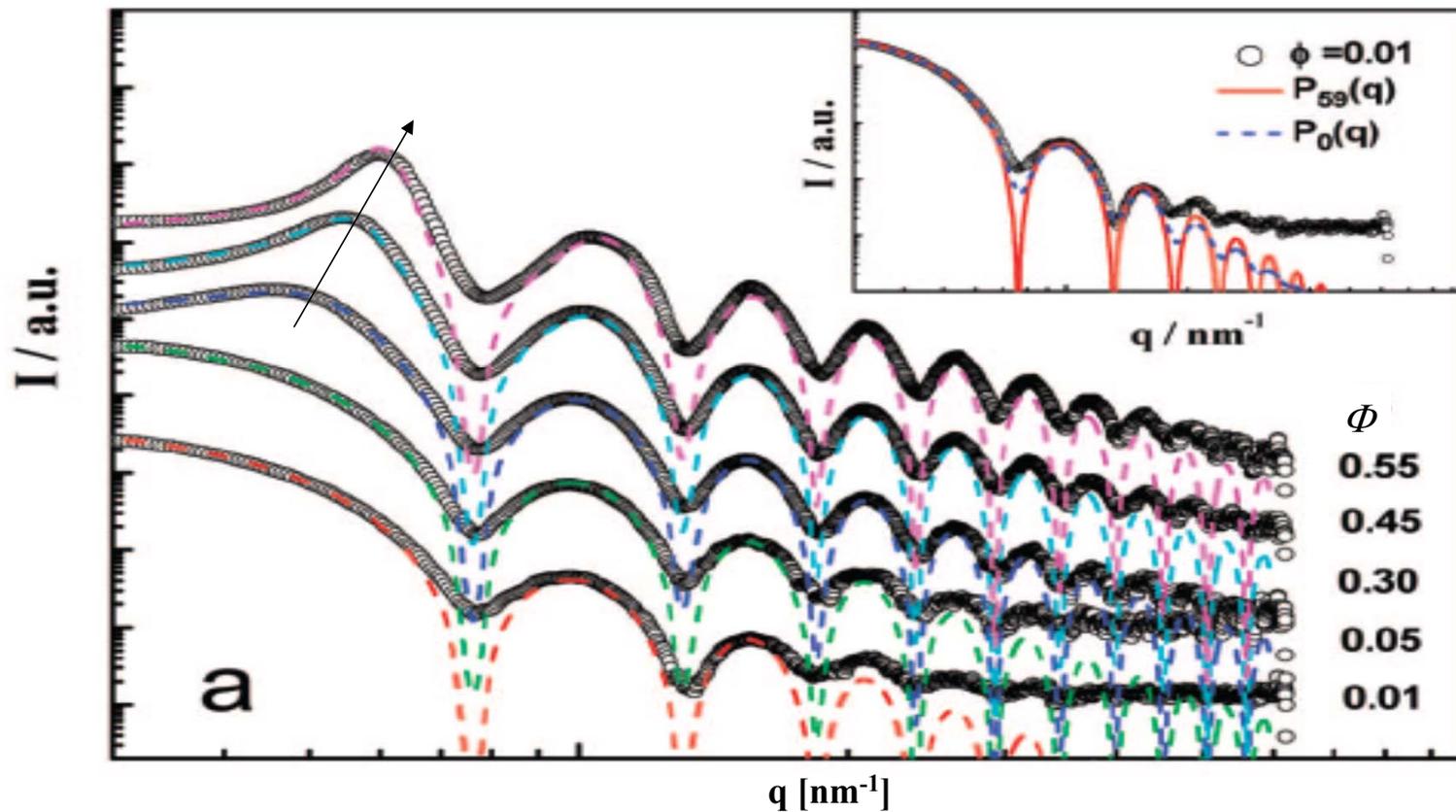
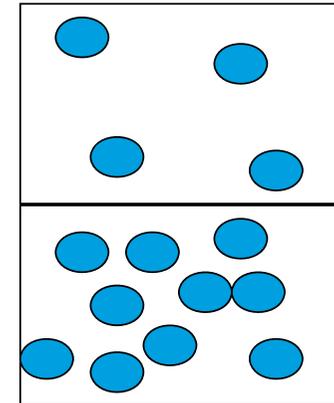


# Colloidal systems

> Latex spheres in water

$$I(q) = c P(q) S(q)$$

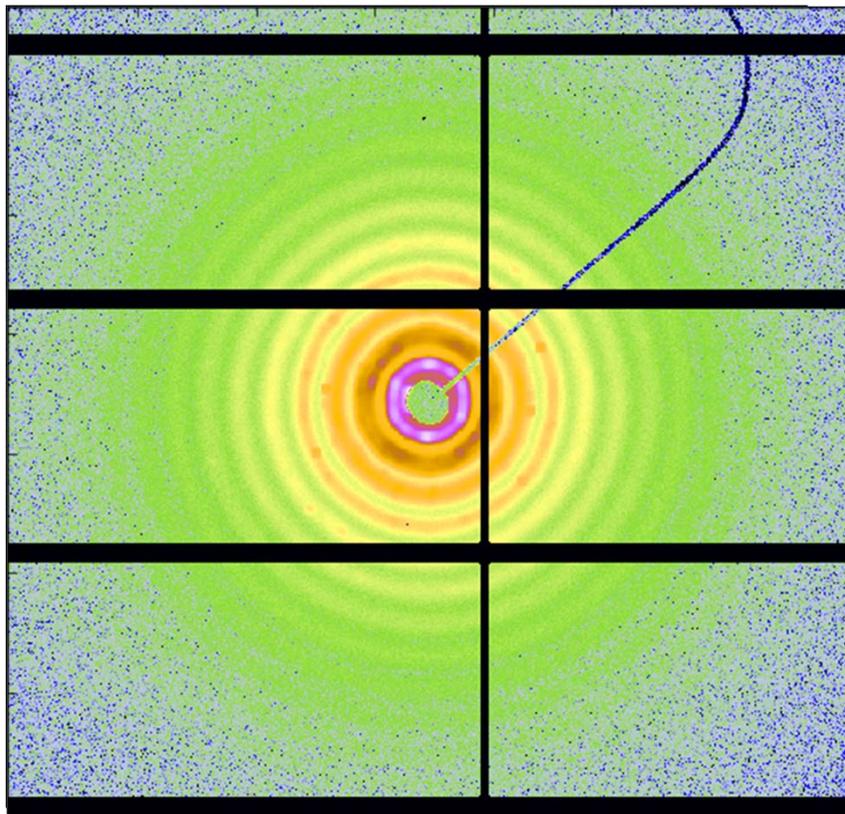
Low  $\phi$      $P(q), S(q)=1$   
 High  $\phi$      $S(q)P(q)$



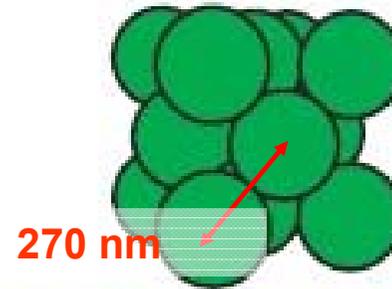
- > Gaussian distribution of particle sizes
- > Shift in maximum: Decreasing distance

# Illustration

- > USAXS at photonic crystals
- > USAXS in highly concentrated colloidal suspensions



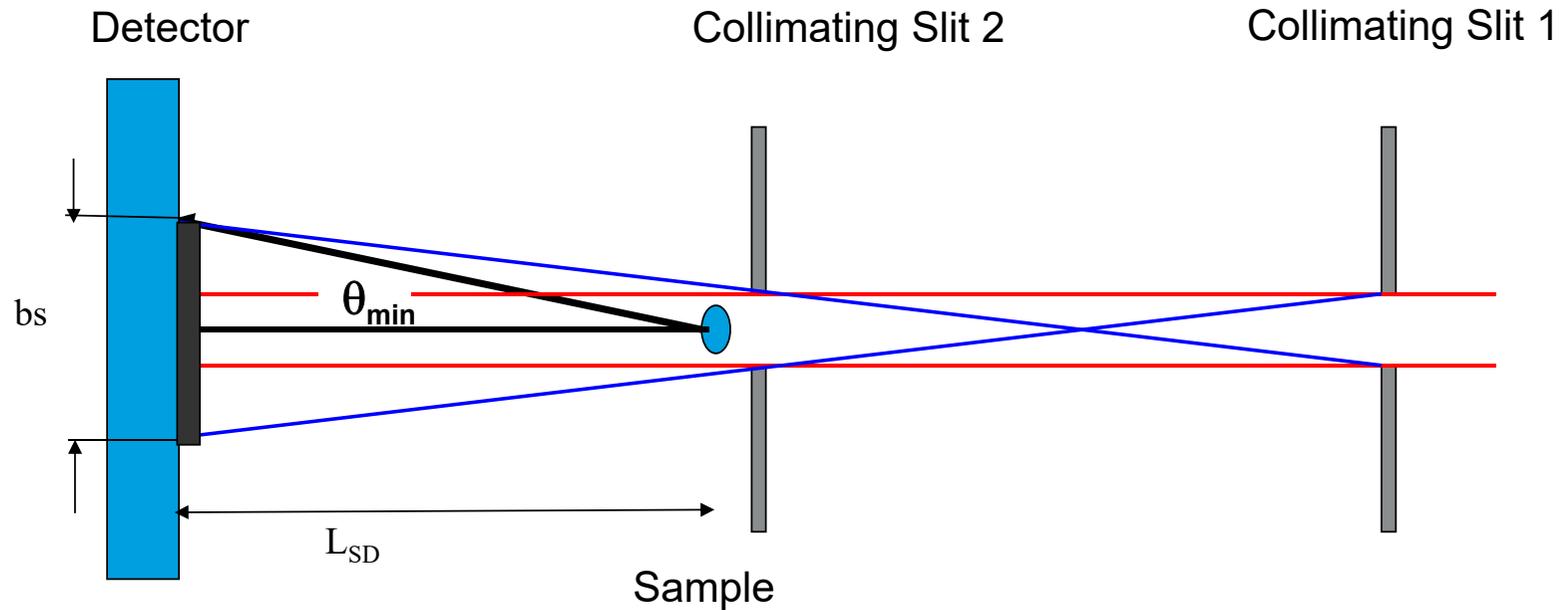
Beamstop **fcc**



<http://ab-initio.mit.edu/book>

[http://lamp.tu-graz.ac.at/~hadley/ss1/emfield/photonic\\_crystals/photonic\\_table.html](http://lamp.tu-graz.ac.at/~hadley/ss1/emfield/photonic_crystals/photonic_table.html)

# SAXS collimation and scattering geometry



$L_{SD}$  determines resolution

$$\theta_{\min} = bs / (2 L_{SD})$$

Use Bragg's law:

$$d_{\max} = \frac{\lambda}{\theta_{\min}}$$

# Example: P03 / MiNaXS @ PETRA III, DESY

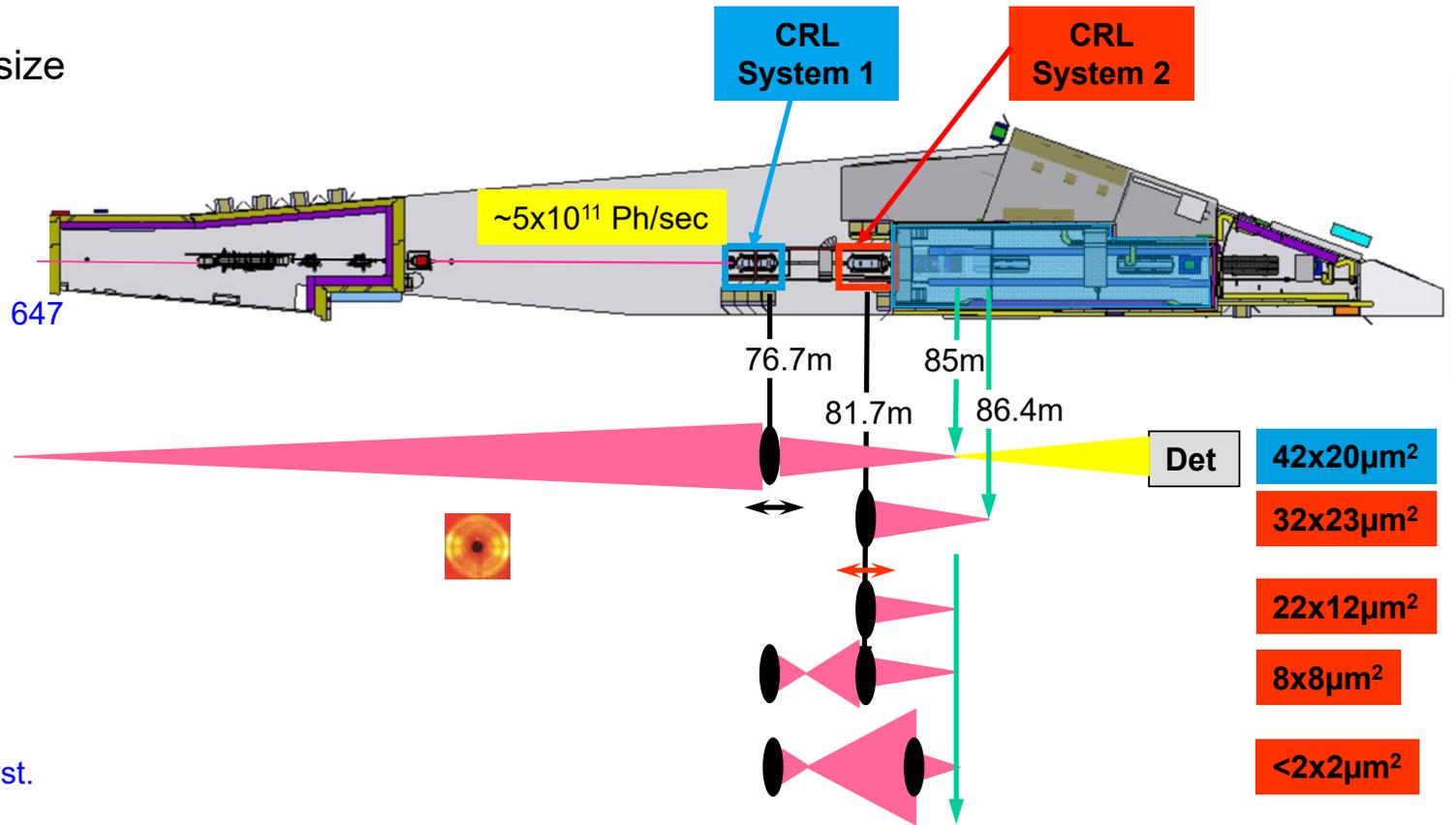
Flexible choice of beam size and divergence

Fixed focal spot position and size

- E=13keV 22x12 $\mu\text{m}^2$
- E=15keV 24x17 $\mu\text{m}^2$

Roth et al., J. Phys.: Cond. Matter 23, 254208 (2011)

Buffet, Roth et al., J. Synchr. Rad., 19, 647 (2012)



> Nanofocus end station:

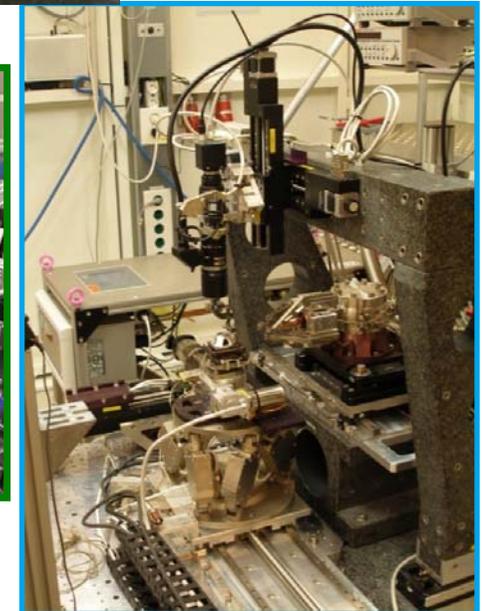
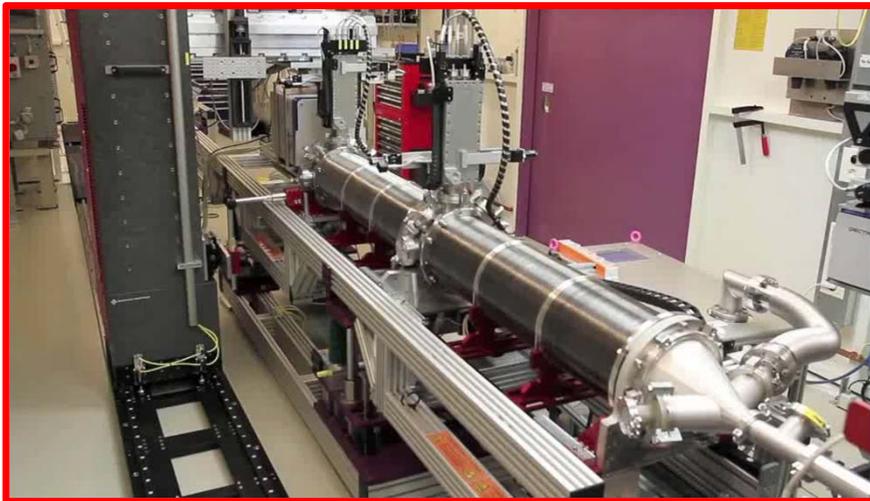
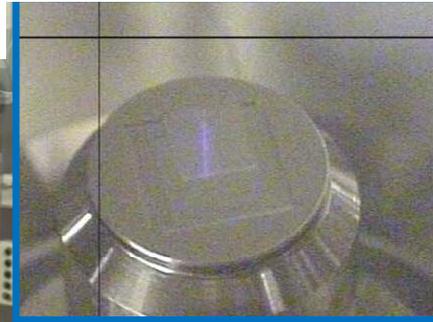
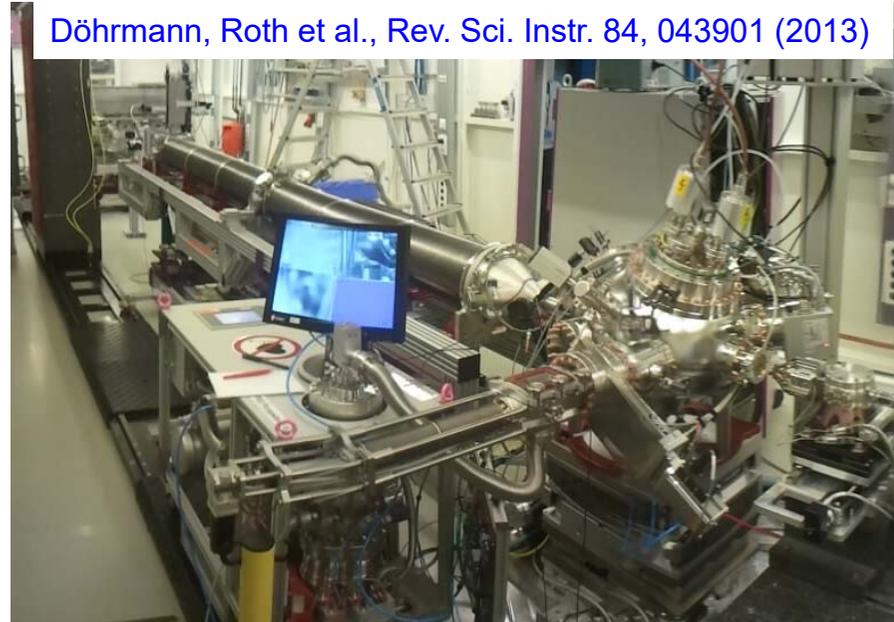
- <1.5x1.5 $\mu\text{m}^2$
- 500x500nm<sup>2</sup>

Krywka, SVR et al., J. of Appl. Cryst. 45, 85 (2012)

# Impression @ P03

- Adjust scattering angles  
↔  $d\Omega$   
↔ q-ranges
- $5\text{cm} < D_{\text{SD}} < 8.6\text{m}$
- Highly flexible
- Separate WAXS device
- GISAXS / GIWAXS

Döhrmann, Roth et al., Rev. Sci. Instr. 84, 043901 (2013)



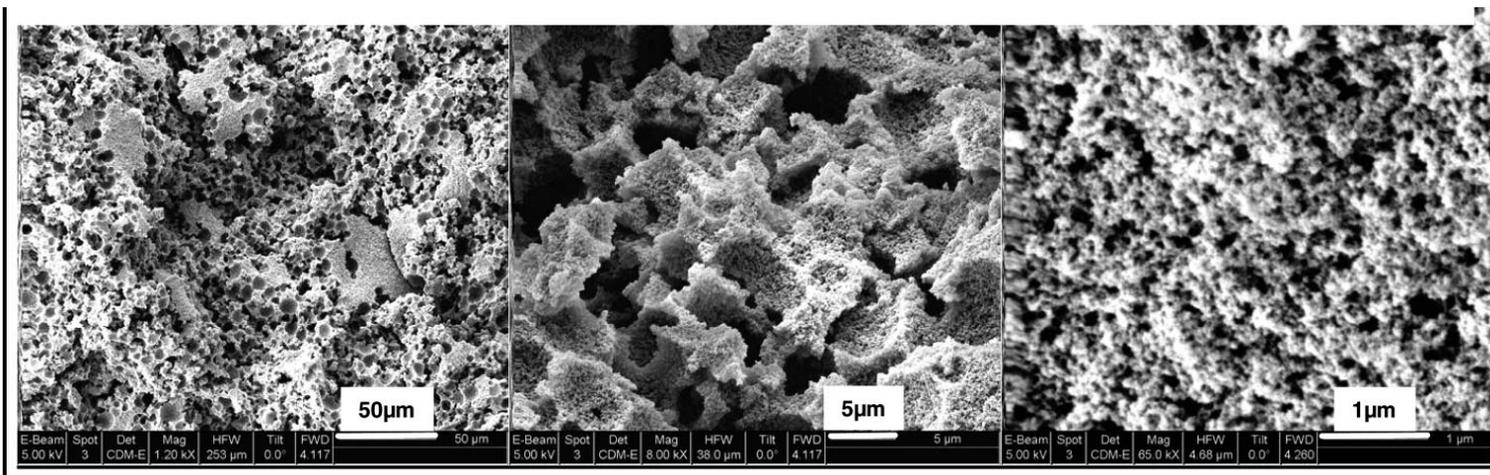
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# Aerogels

- Highly porous materials:
  - OLED: matching of refractive indices
  - molecular sieves
  - sensors
- Challenges
  - **Generation of pores** with dimensions greater than 100 nm, yet submicron
  - **Characterization of size**



# Quantitative analysis

- Monomodal distribution of particles
- $I(Q) = \Delta \rho^2 N \int_0^\infty V_P^2(R) P(R) * D(R) dR$

$$P(q, r) = \frac{9}{(qR)^6} [\sin(qR) - qR \cos(qR)]^2 \quad (7)$$

A first choice for the object size distribution is assuming a Schulz-Zimm distribution<sup>43</sup>

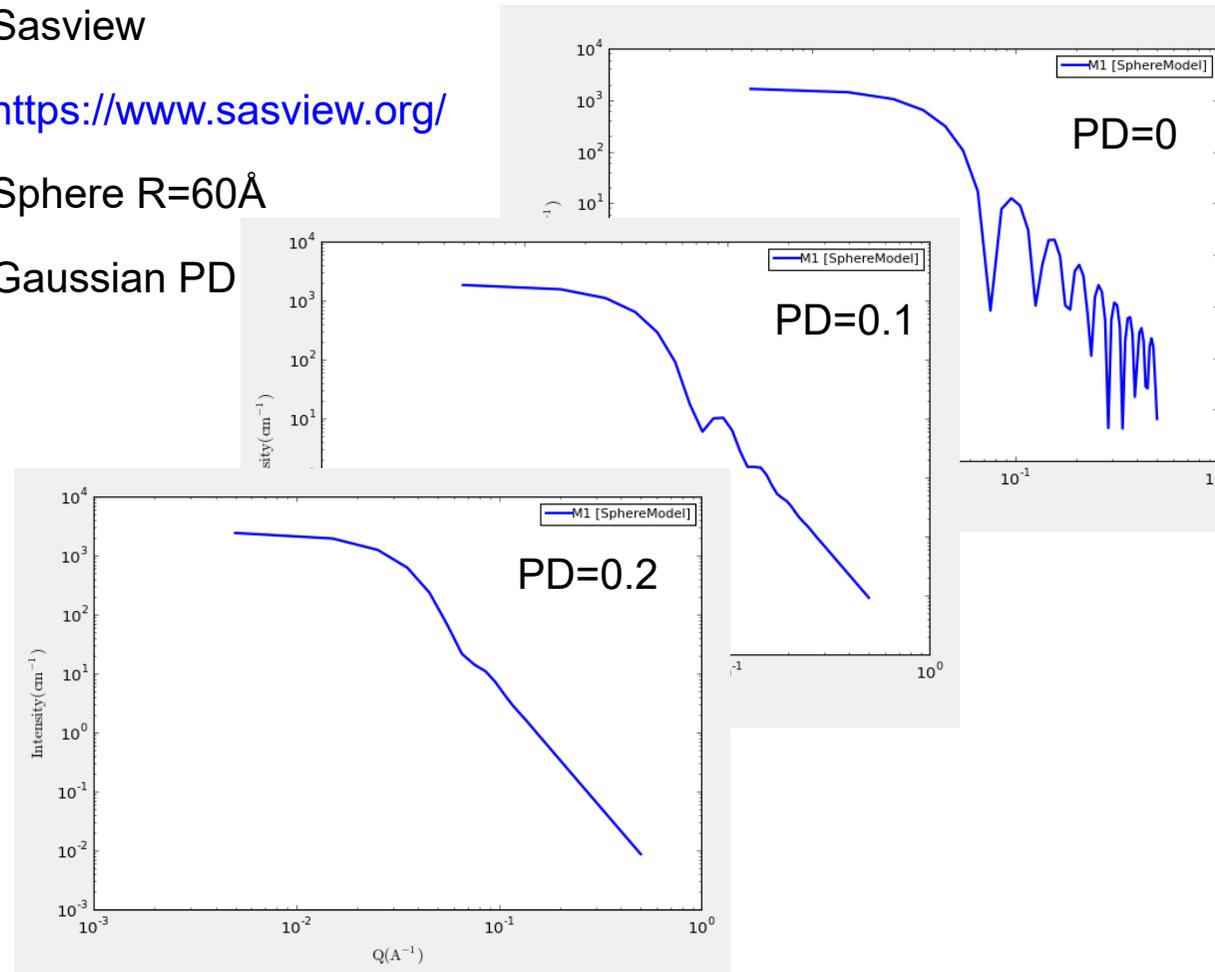
$$D(R) = \frac{1}{\Gamma(\beta)} \left(\frac{\beta}{\omega}\right)^\beta R^{\beta-1} \exp\left[-\frac{\beta R}{\omega}\right] \quad (8)$$

with  $\omega$  being the average radius,  $\sigma = \beta^{-1/2}$  being the standard deviation with the probability  $\beta$ , and  $\Gamma$  denoting the gamma function. The distribution is normalized such that

$$\int_0^\infty D(R) dR = 1 \quad (9)$$

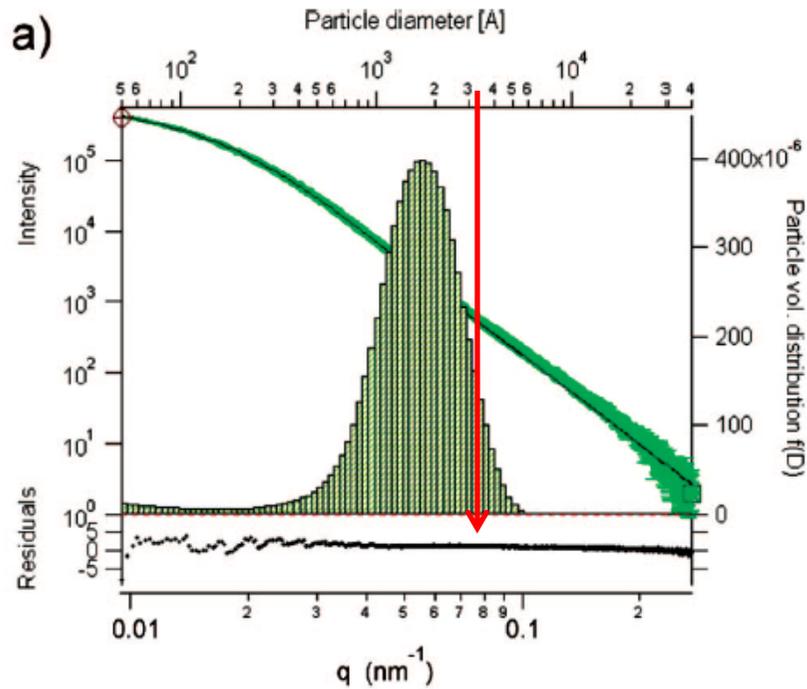
Eggers, Roth et al., Langmuir 24, 5887 (2008)

- Sasview
- <https://www.sasview.org/>
- Sphere R=60Å
- Gaussian PD

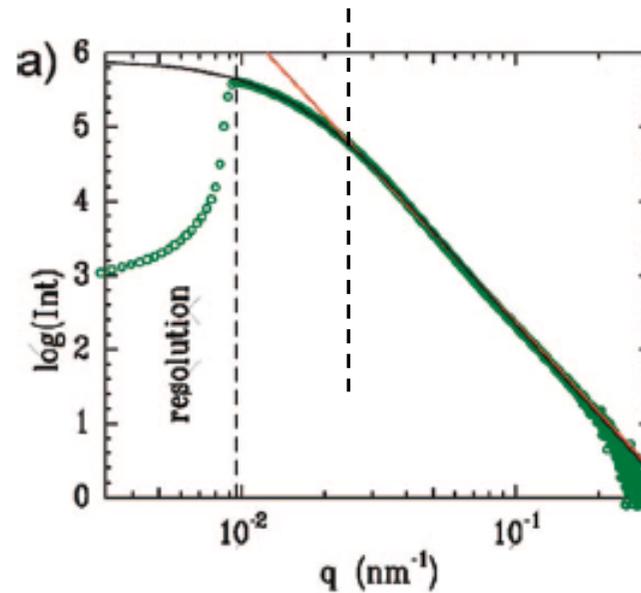


# Quantitative analysis

- Monomodal distribution of particles
- $I(Q) = \Delta \rho^2 N \int_0^\infty V_P^2(R) P(R) * D(R) dR$
- Porod law:
  - Particle size  $\sim 360$  nm



$$2R \sim 360 \text{ nm} = 3600 \text{ \AA}$$



# Quantitative analysis

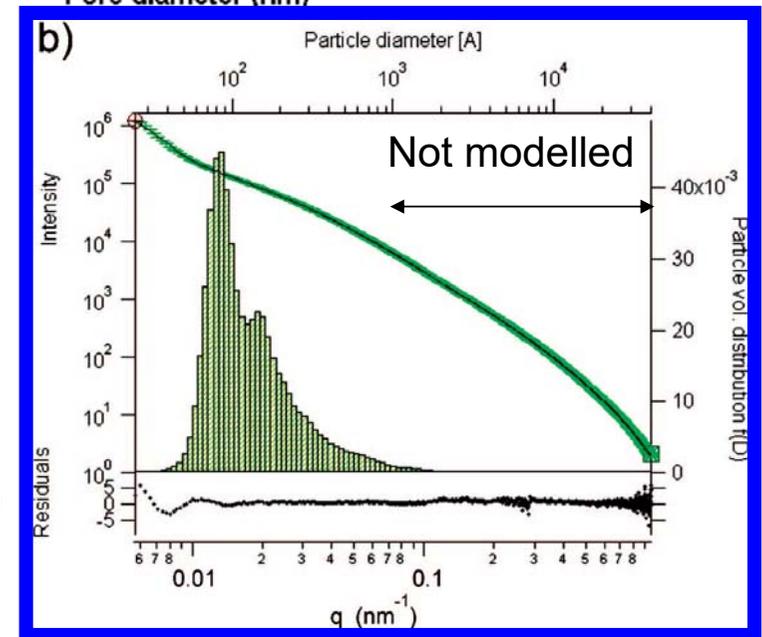
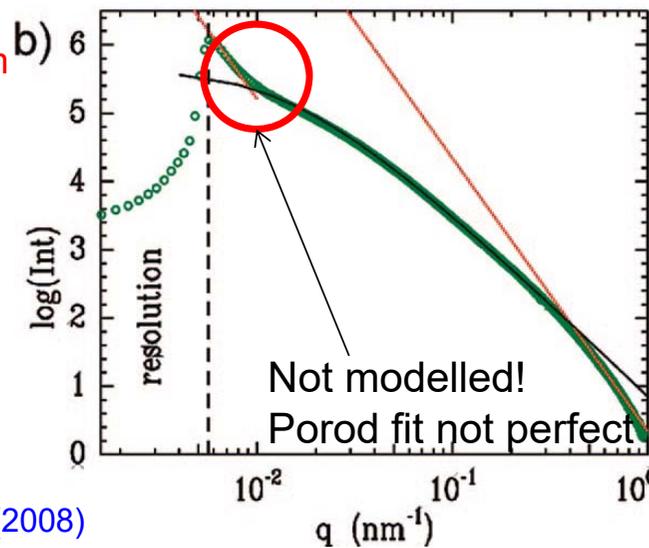
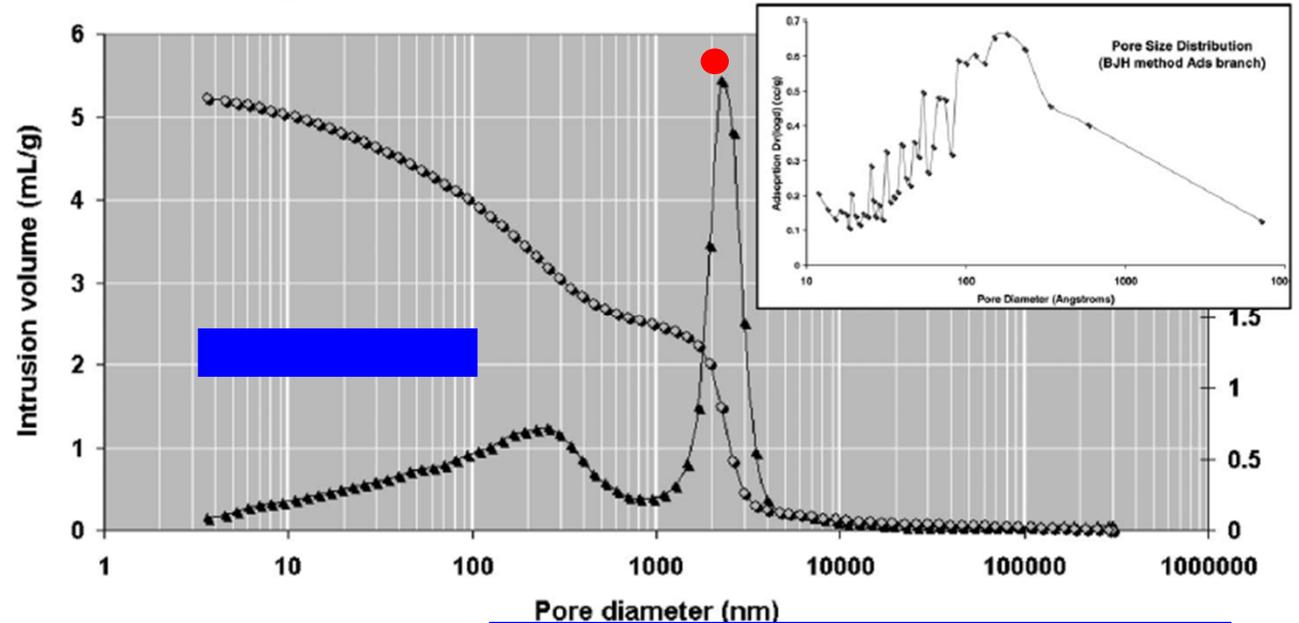
Bimodal distribution particles

$$I(Q) = \Delta \rho^2 N \int_0^\infty V_p^2(R) P(R) * D(R) dR$$

Influence of SAXS resolution

Porod law:

- Particle size ~ 30nm
- Pore size estimate >1600nm



Eggers, Roth et al., Langmuir 24, 5887 (2008)

# Outline

## Introduction to small-angle scattering – the use of X-rays and Neutrons

- Basics SAXS (hold ~SANS)
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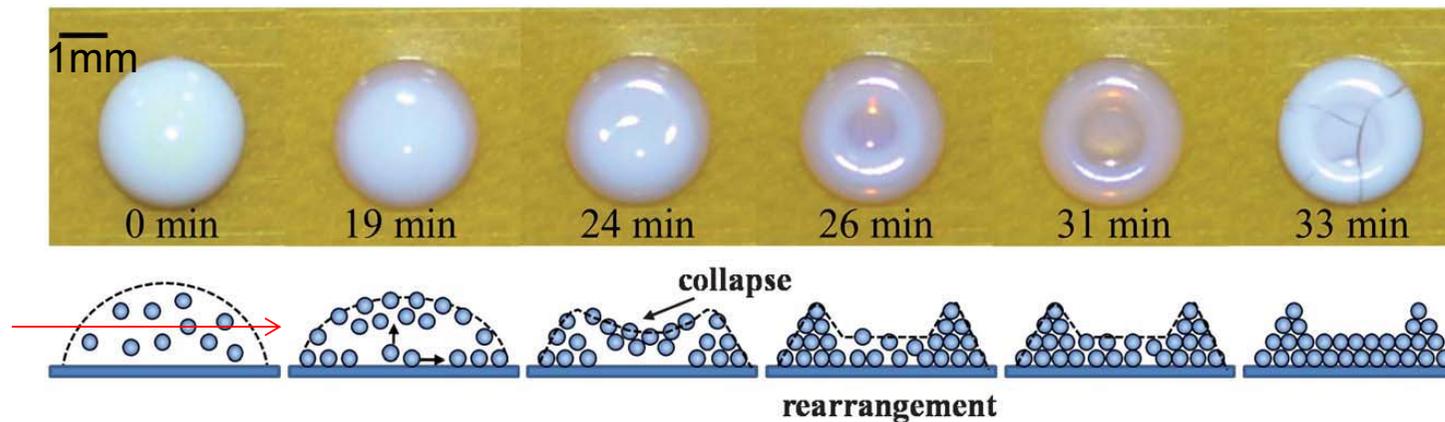
# The drying droplet

- Self-organisation: attractive capillary forces
- correlated nano-structures
- industrial processes
  - spray drying (see also GISAXS part)
  - food processing, pharmaceuticals
  - Paintings/coatings

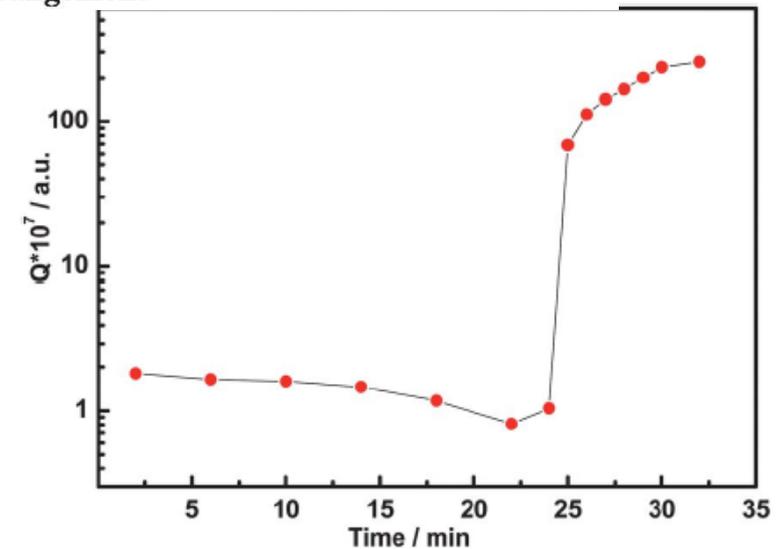


# Porod Invariant - practical application

- Colloidal solution: drying thick droplet



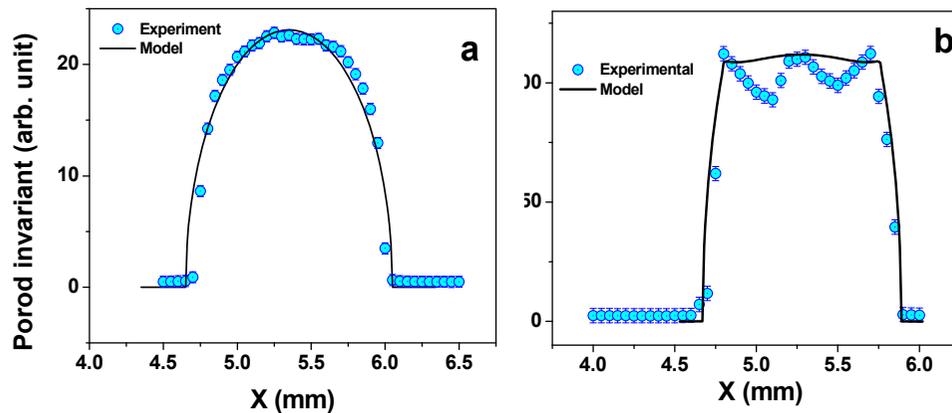
- Evaporation of water:
  - Irradiated volume becomes smaller: shrinking
  - Distance of colloidal particles decreases,  $\Phi \rightarrow 1$
  - $\Delta\rho$  increases (air!), as water removed from interstitial sites



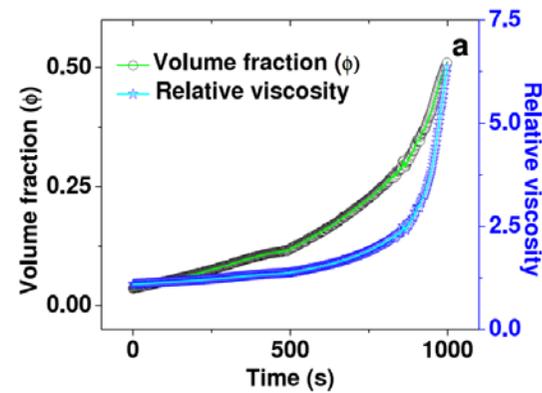
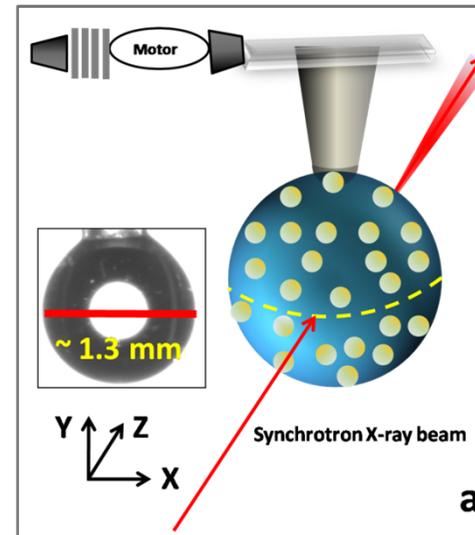
# The drying droplet

- > Slow / fast drying
- > Concentration of colloids:
  - Arresting of colloids
  - Homogenous
  - Core shell effect (,coffee ring')

- > Follow concentration profile in-situ



$$Q \sim \phi(1-\phi)(R^2 - x^2)^{0.5}$$



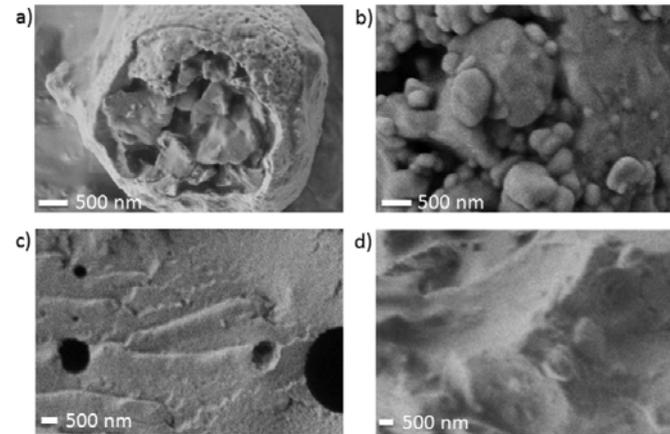
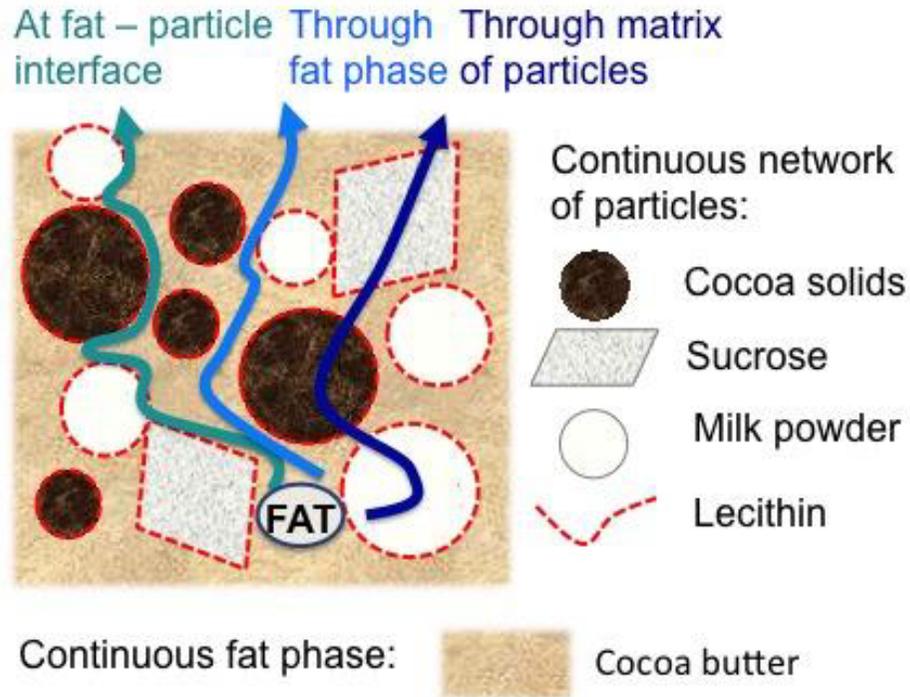
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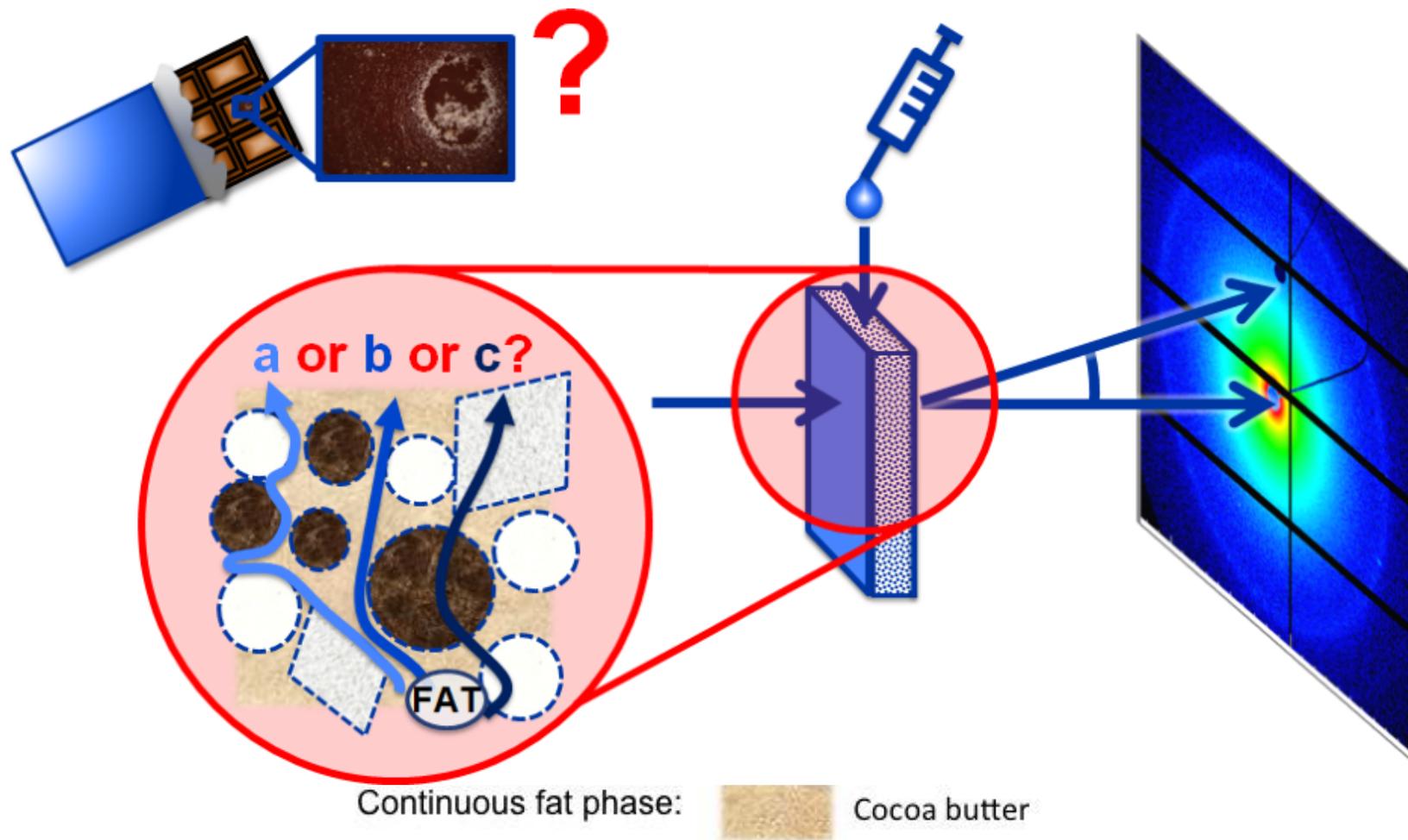
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# Chocolate

- A real multicomponent system
- Nestlé, TU HH, DESY
- Fat Blooming - pathways

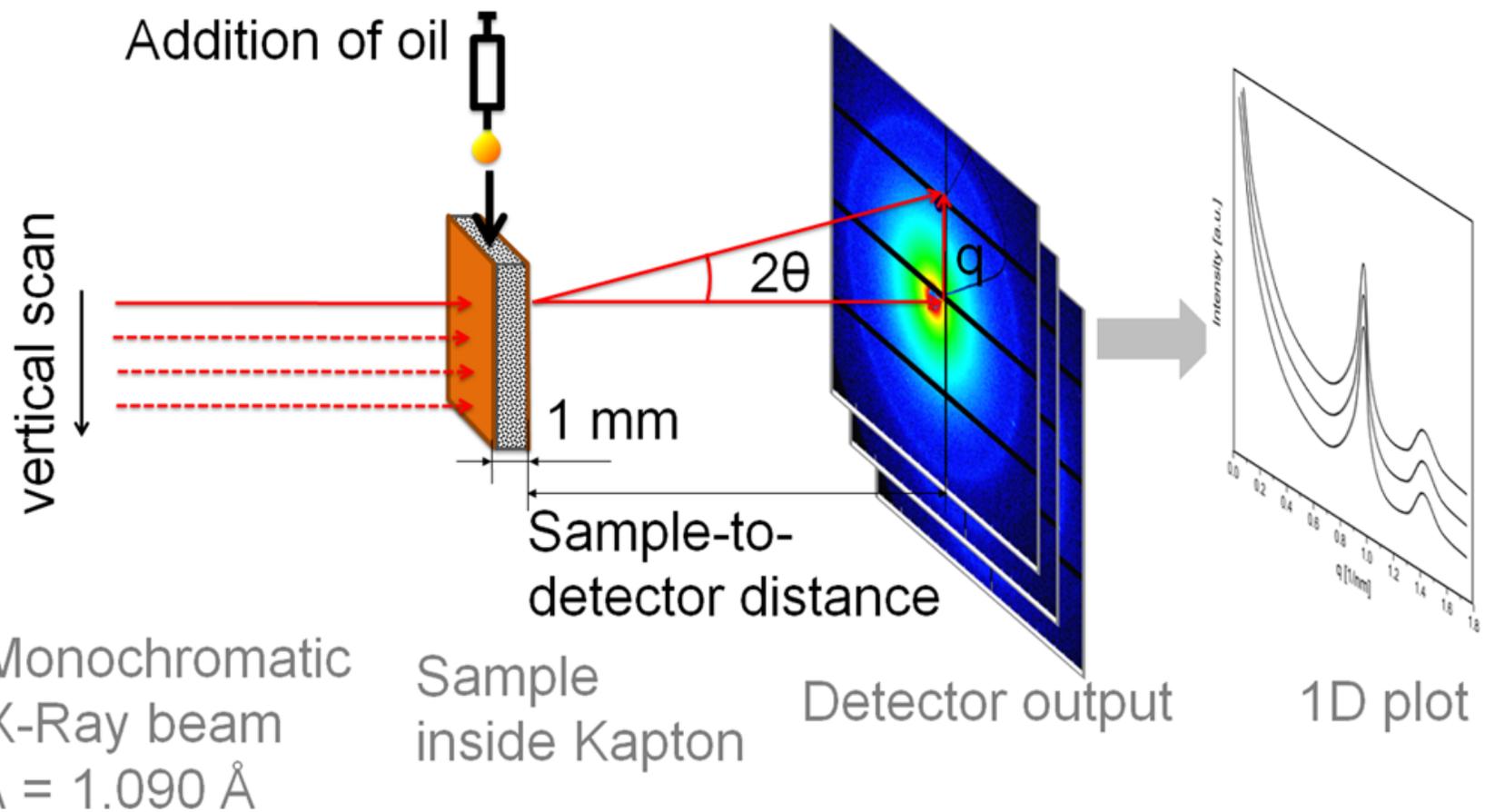


# Chocolate



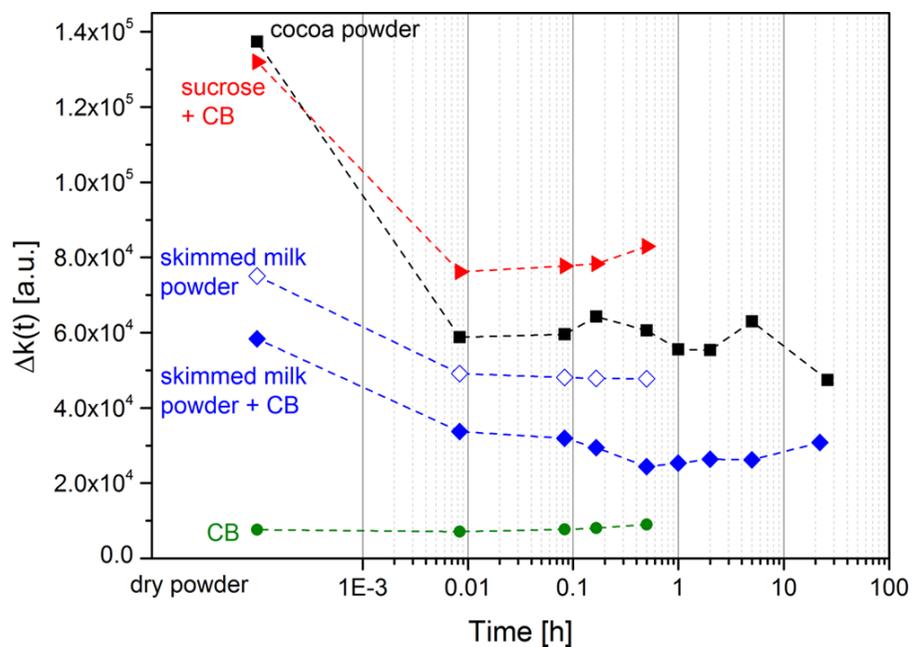
# Chocolate

A real multicomponent system

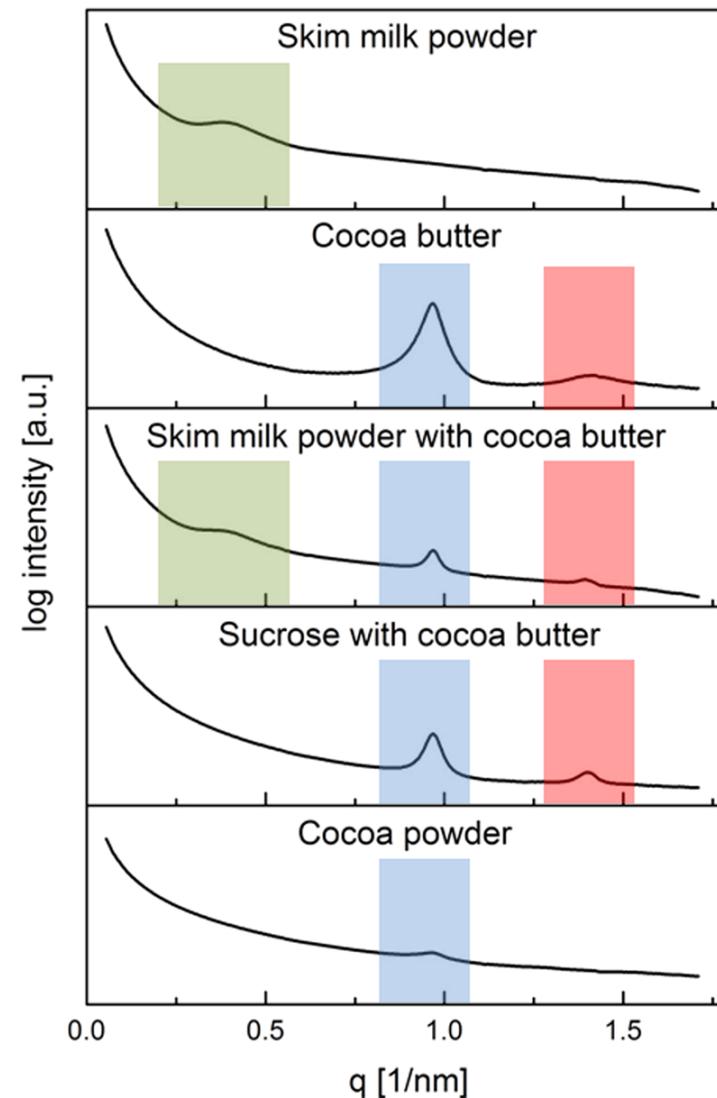


# Chocolate

- A real multicomponent system
- Superposition of SAXS contributions

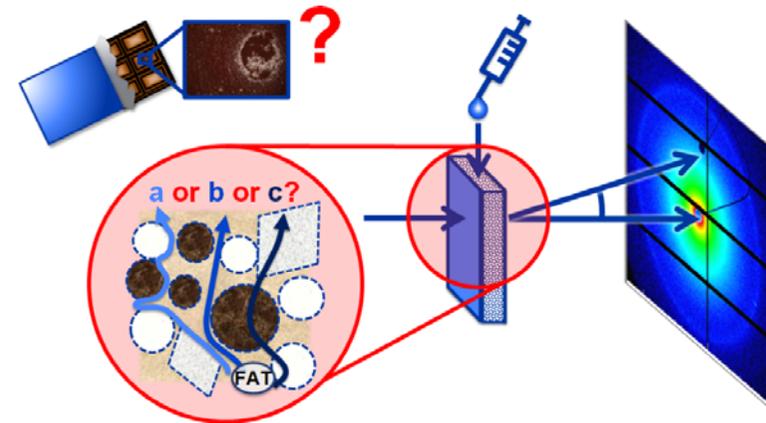
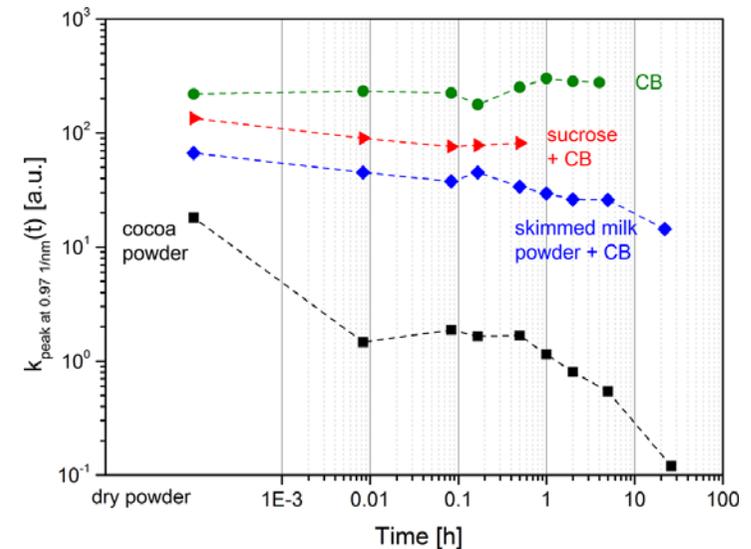


- Different density difference
- Migration: filling of voids by oil: Q decreases



# Chocolate

- Peak intensities
- Pores, cracks: capillary effect
- Then: “chemical migration through the fat phase by softening and partial dissolution of the crystalline cocoa butter.”
- reduction of porosity and a minimization of defects
- a reduced content of noncrystallized liquid cocoa butter
- **b or c**



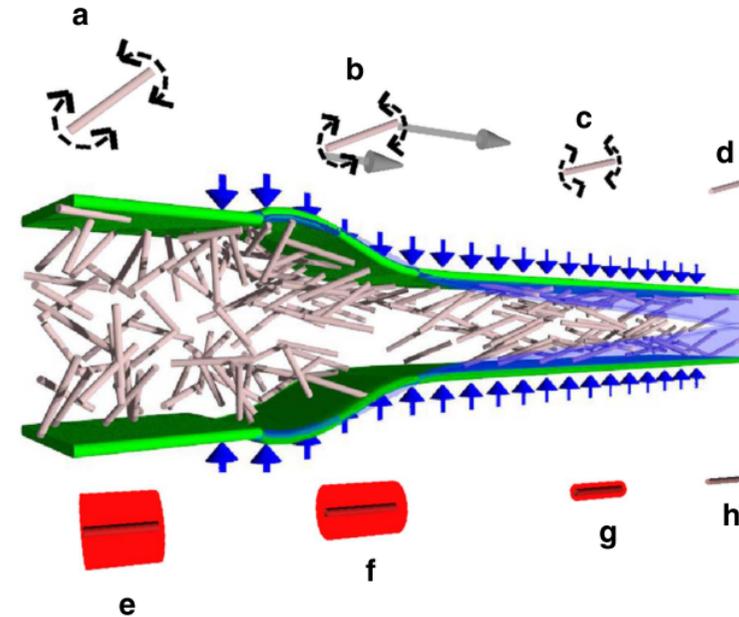
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# Microfluidics – orientation of CNF

- Example: Orientation of CNF in a microfluidic device
- Gulp: The use of rotation matrices in the mathematical description of molecular orientations in polymers



# Microfluidics – orientation of CNF

- Example: Orientation of CNF in a microfluidic device
- Gulp: The use of rotation matrices in the mathematical description of molecular orientations in polymers
- Distribution function  $f(\beta)$

$$f(\beta) = \sum_{i=0}^{\infty} a_i P_i(\cos \beta)$$

- Average of all possible orientations

$$\langle A \rangle = \int_0^{\pi} A(\beta) f(\beta) \sin \beta d\beta$$

Legendre polynomials

$$P_0(\cos \beta) = 1$$

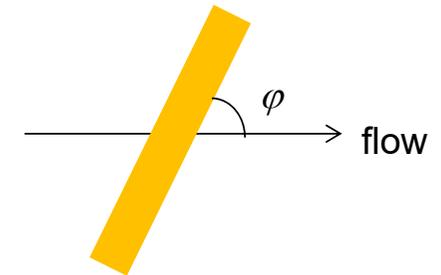
$$P_1(\cos \beta) = \cos \beta$$

$$P_2(\cos \beta) = 1/2(3\cos^2 \beta - 1)$$

$$S = \left\langle \frac{3}{2} \cos^2 \varphi - \frac{1}{2} \right\rangle$$

$$S = \int_0^{\pi} I(\varphi) \left( \frac{3}{2} \cos^2 \varphi - \frac{1}{2} \right) \sin \varphi d\varphi$$

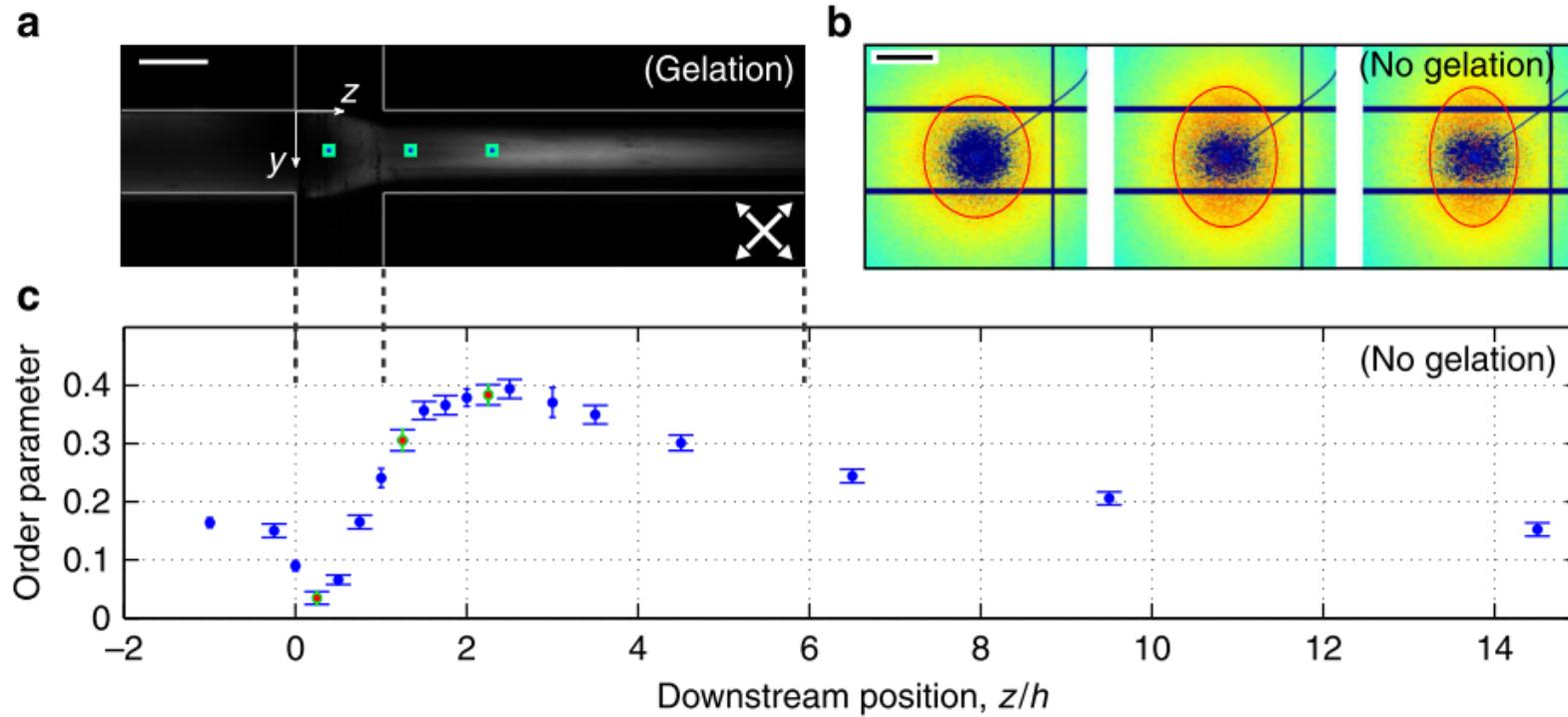
$$\int_0^{\pi} I(\varphi) \sin \varphi d\varphi = 1$$



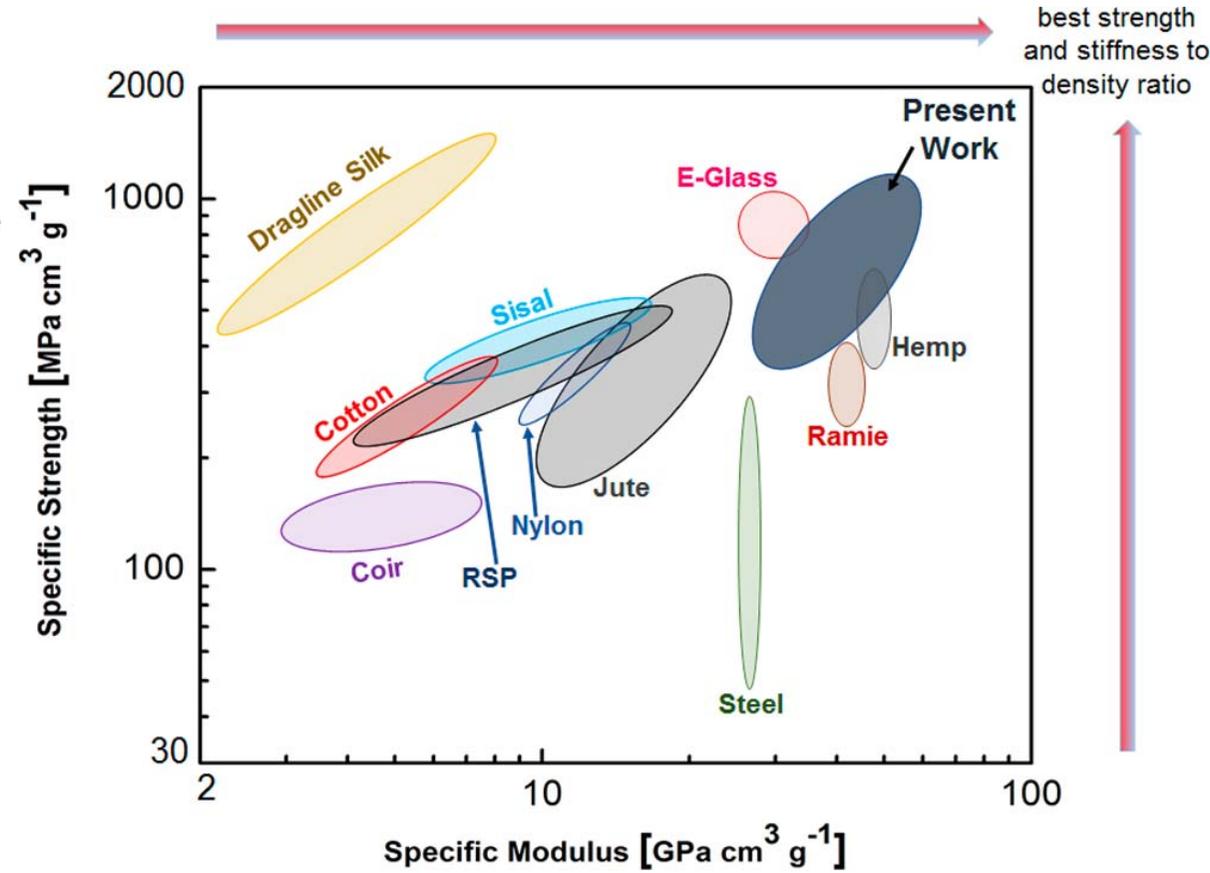
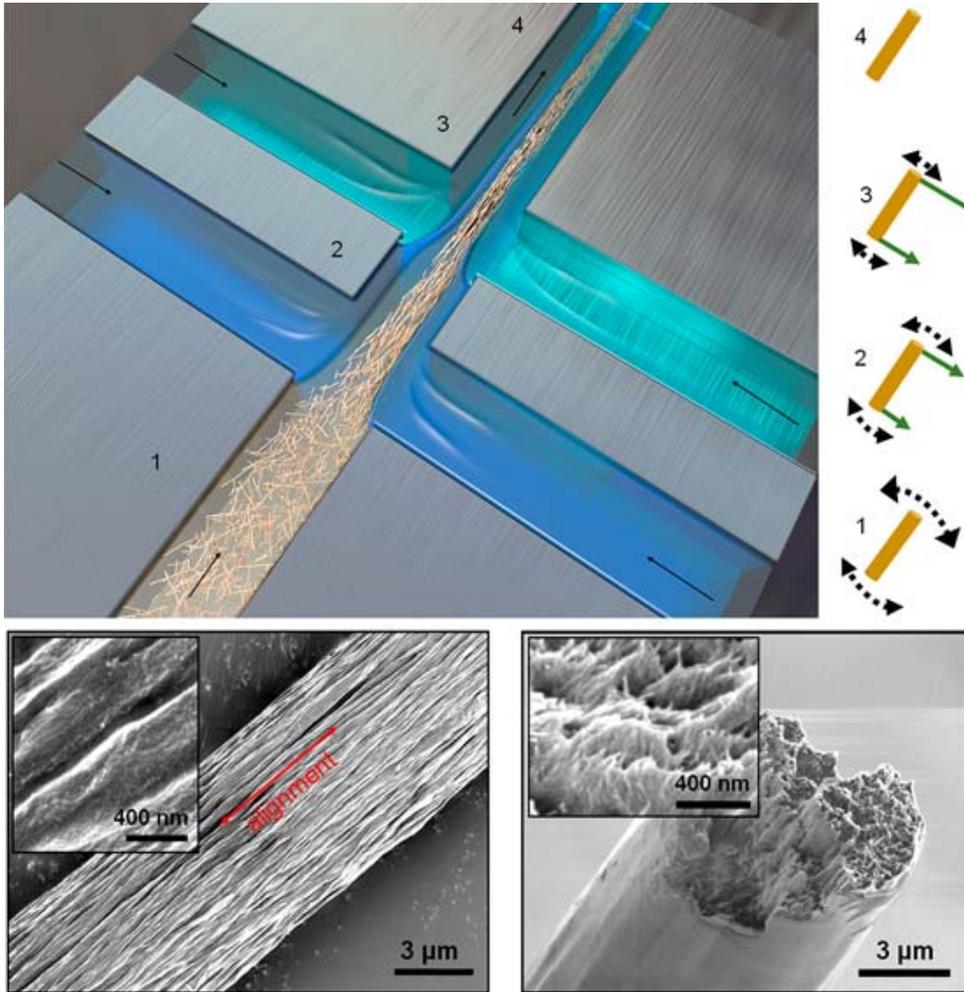
Gulp, Colloid Polym. Sci. 273, 607–625 (1995)

Håkansson, Roth et al., Nat. Commun. 5, 4018 (2014)

# Microfluidics – orientation of CNF



... and you get the best biofibre ever!



Mittal, ..., Roth, ... ACS Nano (2018) DOI: 10.1021/acsnano.8b01084

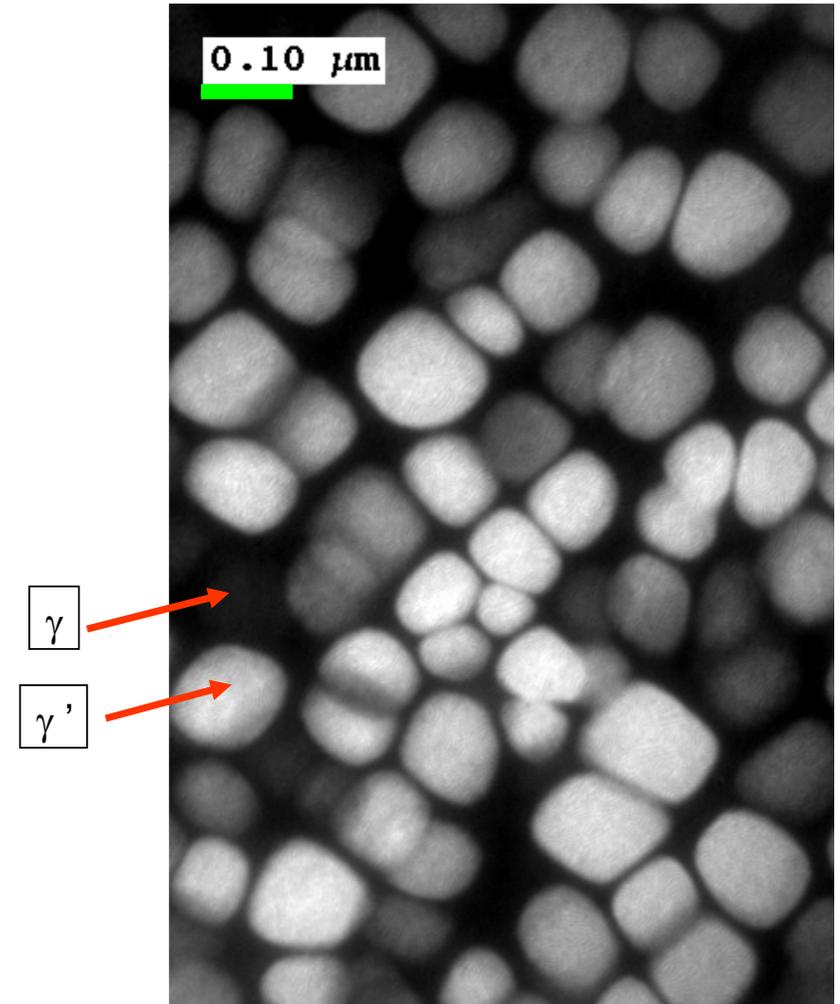
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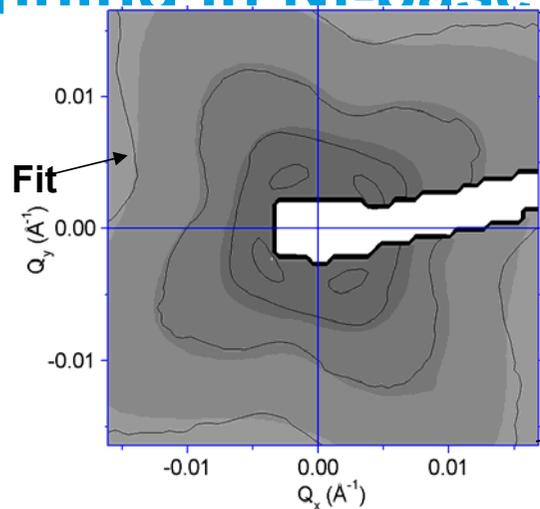
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# Precipitate scanning in Ni-base super alloys

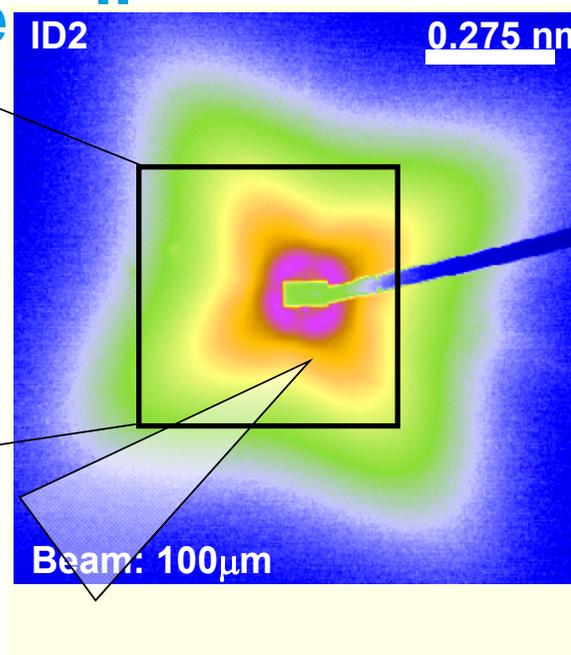
- Ni-base **W-rich** experimental single crystal superalloy (Ni-4.6Al-6.4Ta-5.7Cr-10.8W-2.1Mo)
- Ni-Al solid solution **Matrix** ( $\gamma$ ), fcc
- **Precipitates** ( $\gamma' \rightarrow \text{Al}, \dots$ ),  $\text{Ni}_3(\text{Al}, \text{Ti})$
- TEM:  $\gamma'$ -precipitates **R > 50 nm**
- **D > 100 nm**



# Precipitate scanning in Ni-base superalloy



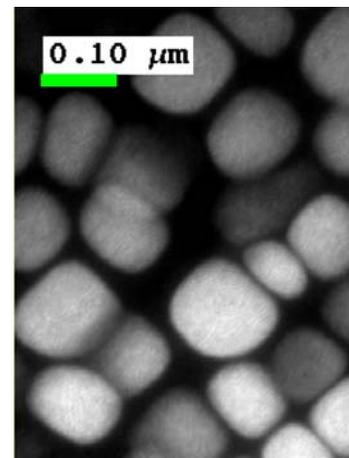
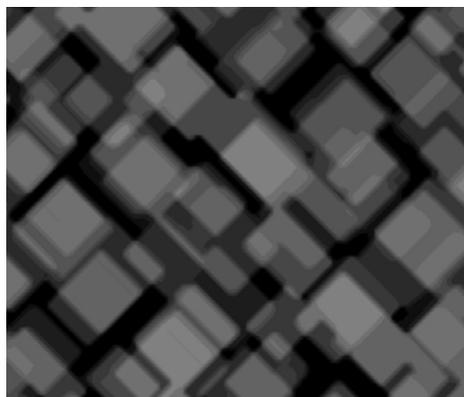
noc\_4: P. Strunz et al., J. Appl. Cryst. **36**, 854 (2003)



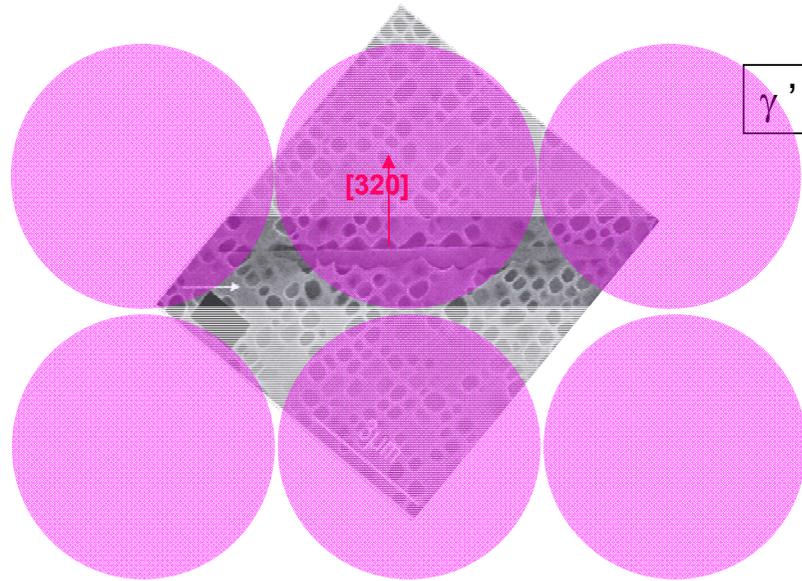
Only  $\omega'$  phase



Courtesy:  
Gilles  
Strunz



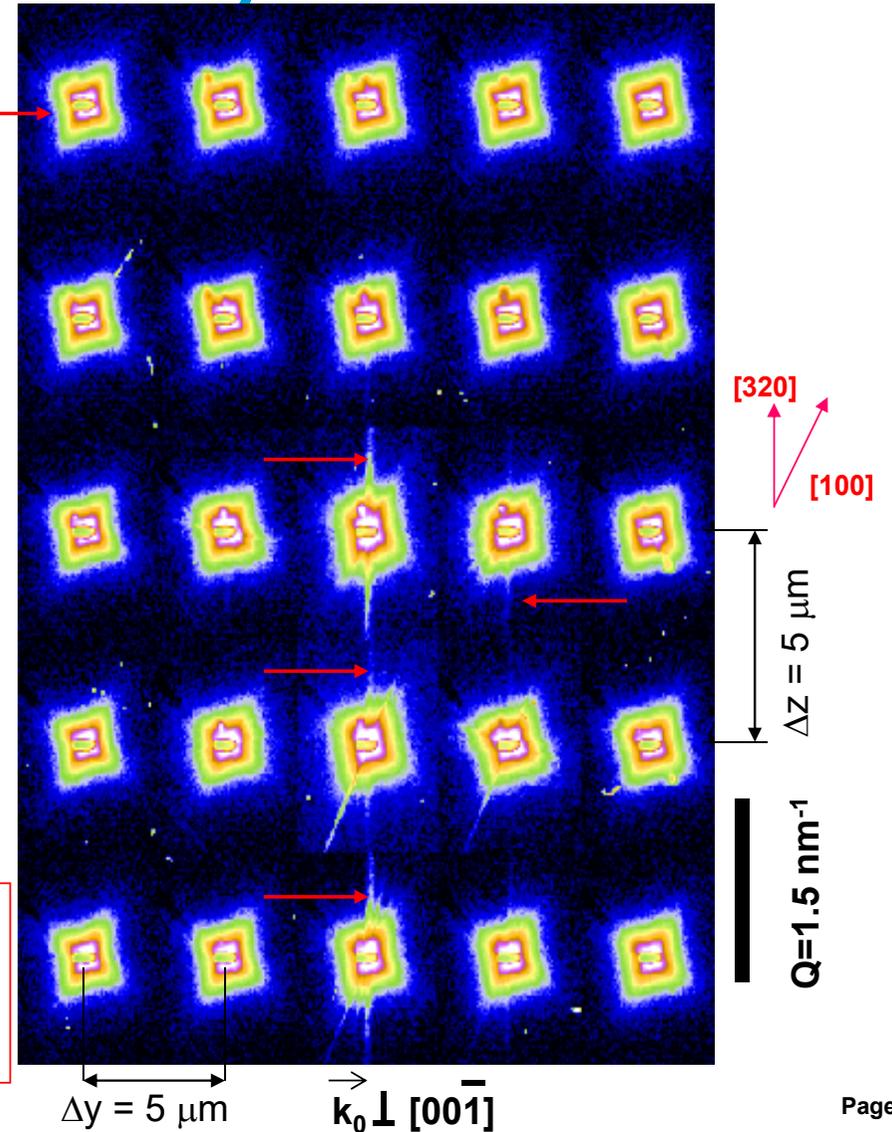
# Precipitate scanning in Ni-base super alloys



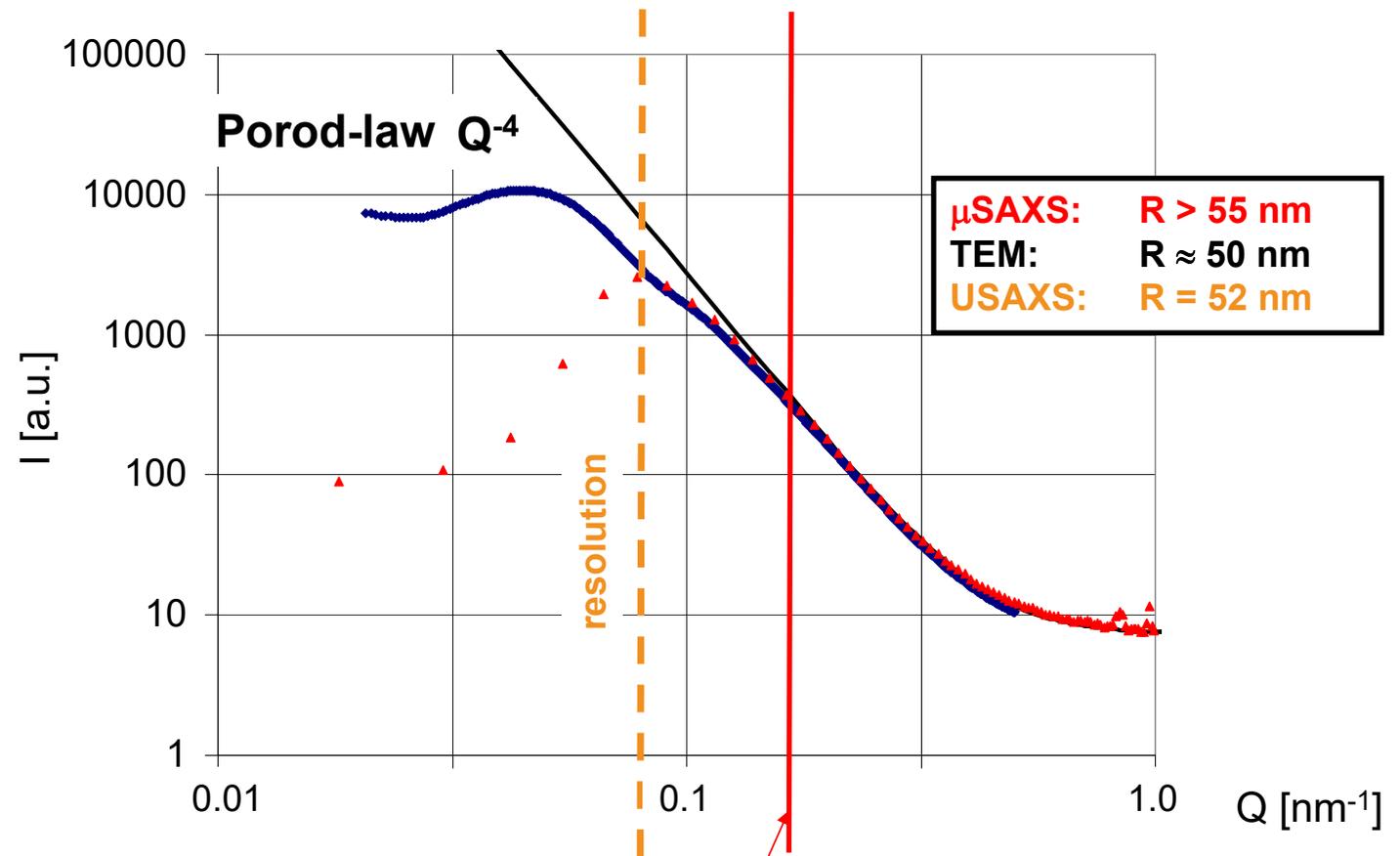
R. Gilles et al.  
 Scripta Mat. **39**, 715 (1998)

- $\sigma$  phase precipitate:  
 embrittlement of alloy  
 crack formation and propagation

- streaking: correct orientation  
 -  $\sigma$  phase: stack - distance  $5 - 15 \mu\text{m}$   
 diameter  $2R < 10 \mu\text{m}$   
 thickness  $t > O(100 \text{ nm})$



# Precipitate scanning in Ni-base super alloys



Lower minimum of particle size distribution

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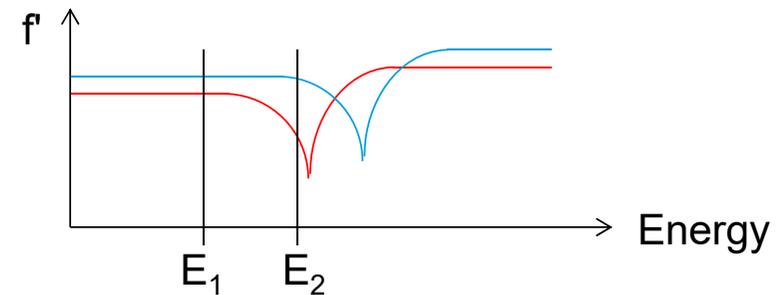
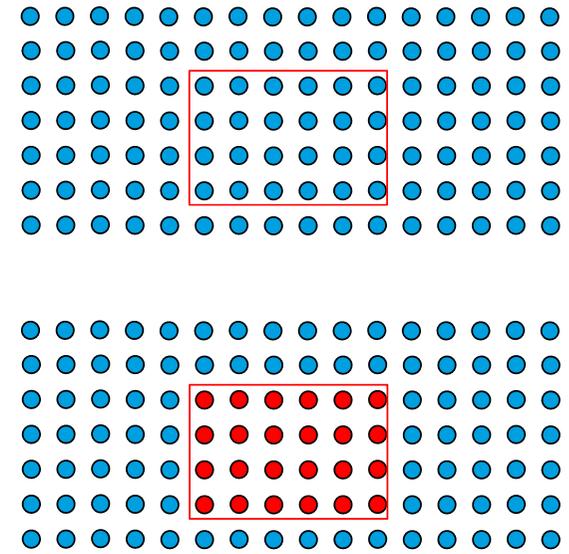
# Contrast variation SAXS

- Contrast variation
- $f(Q, \omega) = f_0(Q) + f'(\omega) + f''(\omega)$
- Synchrotron: variation of energy

$$I(q) = NF(q)F^*(q) = N(F_0^2(q) + \boxed{2f'F_0(q)N_R(q)} + \boxed{(f'^2 + f''^2)N_R^2(q)})$$

Resonant terms

- Resonance:
- Difference in  $f \rightarrow$  contrast:  
 $I \sim |\Delta\rho|^2 \sim |f_P - f_M|^2 \times \dots$
- $f'$  decrease  $\rightarrow$   $I(Q)$  decreases

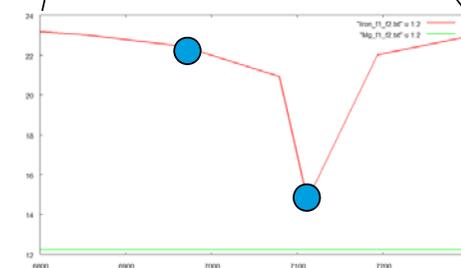
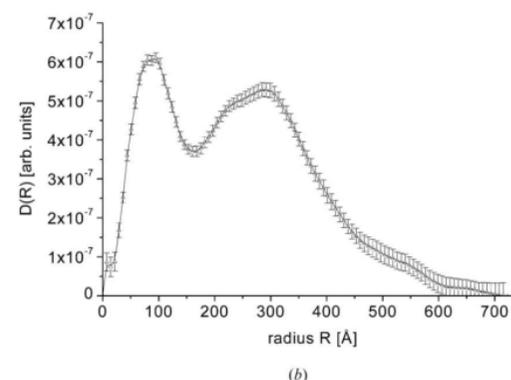
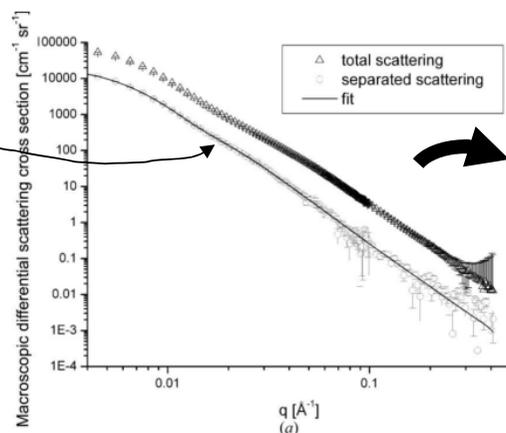
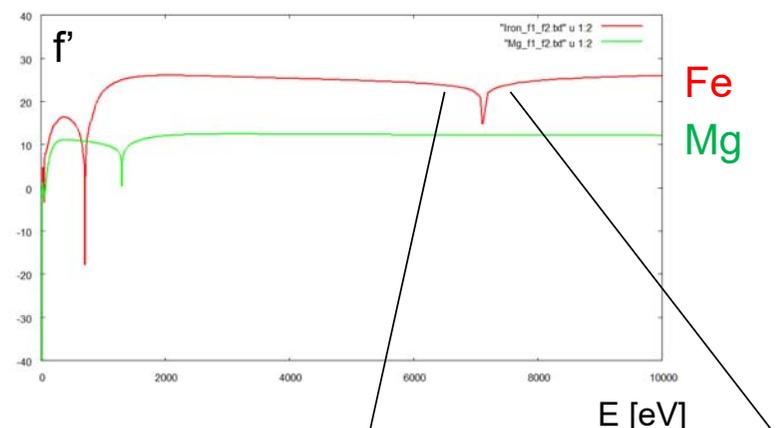
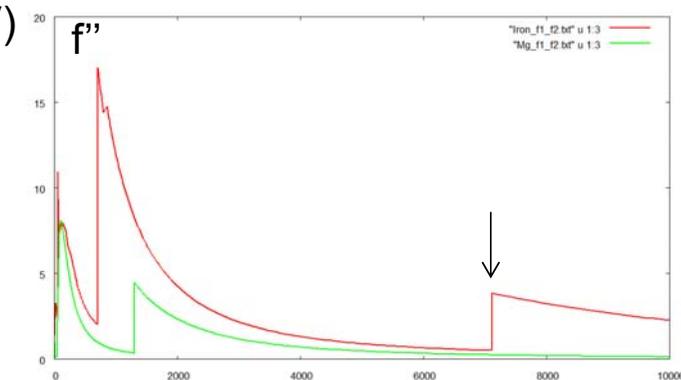


# Contrast variation SAXS

Fe<sub>2</sub>O<sub>3</sub> (1 mol%) nanoparticles in MgHx matrix

$$I(q) = NF(q)F^*(q) = N(F_0^2(q) + 2f'F_0(q)N_R(q) + (f'^2 + f''^2)N_R^2(q))$$

- Scattering contribution from iron oxide NP: 'separated scattering'
- $I(E=7112\text{eV}) - I(E=6884\text{eV})$



Source: CXRO

# Outline

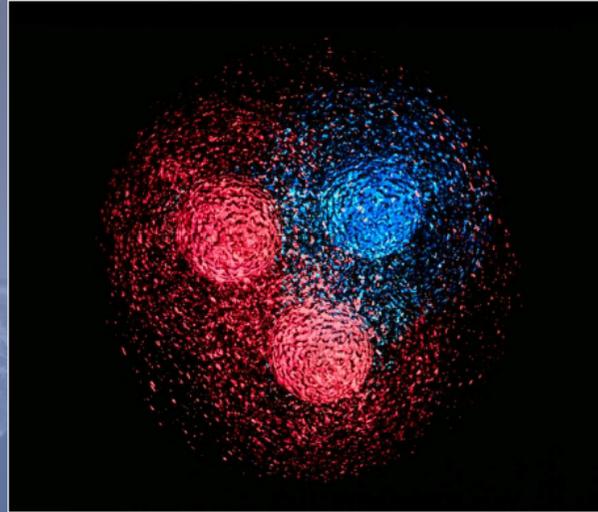
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# What is a Neutron ( $n^0$ )?

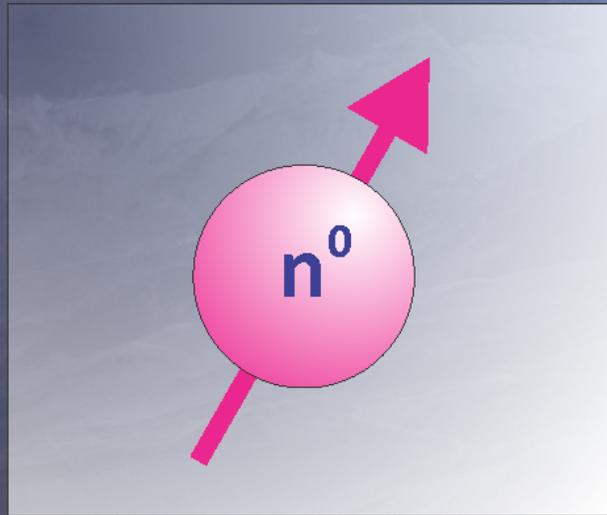
## For particle physicists:

- A subatomic baryon particle of the hadron family.
- Consists of three quarks (2 down & 1 up) of different flavours held together by gluons.



## For neutron scatterers:

- A neutral  $S = \frac{1}{2}$  particle used as an optimal tool to investigate microscopic / macroscopic materials / device properties.
- “Can show where atoms are and what they do” + **magnetism**



# Neutron properties

## NEUTRAL

Charge = 0 → infinitely small elec. dipole moment, neutrons do not see charge!

## HAS A SPIN

$S = \frac{1}{2}$  → Initial state can be polarized & polariz. of final state can be analyzed!

## HAS A MAGNETIC MOMENT

$\mu_{n0} = -1.913 \mu_{\text{Nuc}}$  → neutrons can see magnetism !!!

## RATHER STABLE

$\beta$ -decays but lifetime  $\tau = 881.5$  seconds (enough to survive the experiment!)

## VERY SMALL

Confinement radius  $R = 7 \times 10^{-14}$  m → All interactions are point-like!

## 'IDEAL' MASS

$m_{n0} = 1.675 \times 10^{-27}$  kg  $\approx m_{p+} \approx 1840 \times m_{e-}$

## PARTICLE- & WAVE-LIKE PROPERTIES

Dispersion relation:  $E = \hbar k^2 / 2m \rightarrow \dots$

$\lambda = 5 \text{ \AA} \rightarrow E = 3.3 \text{ meV}$

'Lingo'	E [meV]	$\lambda$ [nm]
Cold	0.1–5	3–0.4
Thermal	5–100	0.4–0.1
Hot	100–500	0.1–0.04

**Neutron wavelengths/energies are perfect for studying material properties from Ångström to mm + dynamics!!!**

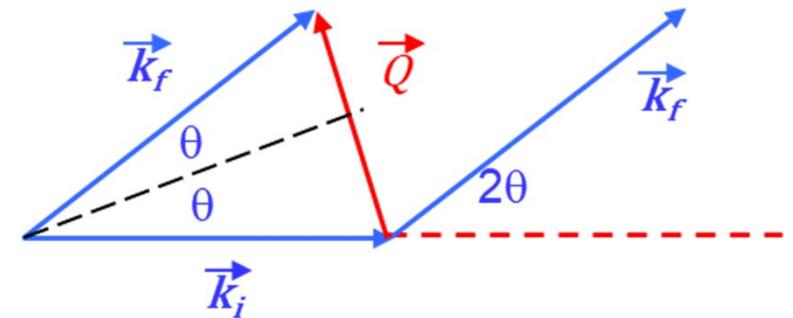
# Scattering of neutrons

- Point-interaction with nuclei (not only e )
- Possible to investigate also light elements, e.g. Hydrogen, which is more or less impossible with x-rays
- “Soft Probe” = no risk for degradation of delicate samples (c.f. X-rays + bio)
- Neutral particle that penetrates probe bulk (intrinsic material) properties as well as buried structures. [surface vs. bulk!!!]

$$Q = (\mathbf{k}_i - \mathbf{k}_f)$$

$$E = \hbar\omega = \hbar^2(\mathbf{k}_i^2 - \mathbf{k}_f^2) / 2m$$

- If the scattering occurs without any loss of neutron energy (  $E = 0$  i.e.  $|\mathbf{k}_i| = |\mathbf{k}_f|$ ) this is called Elastic Neutron Scattering



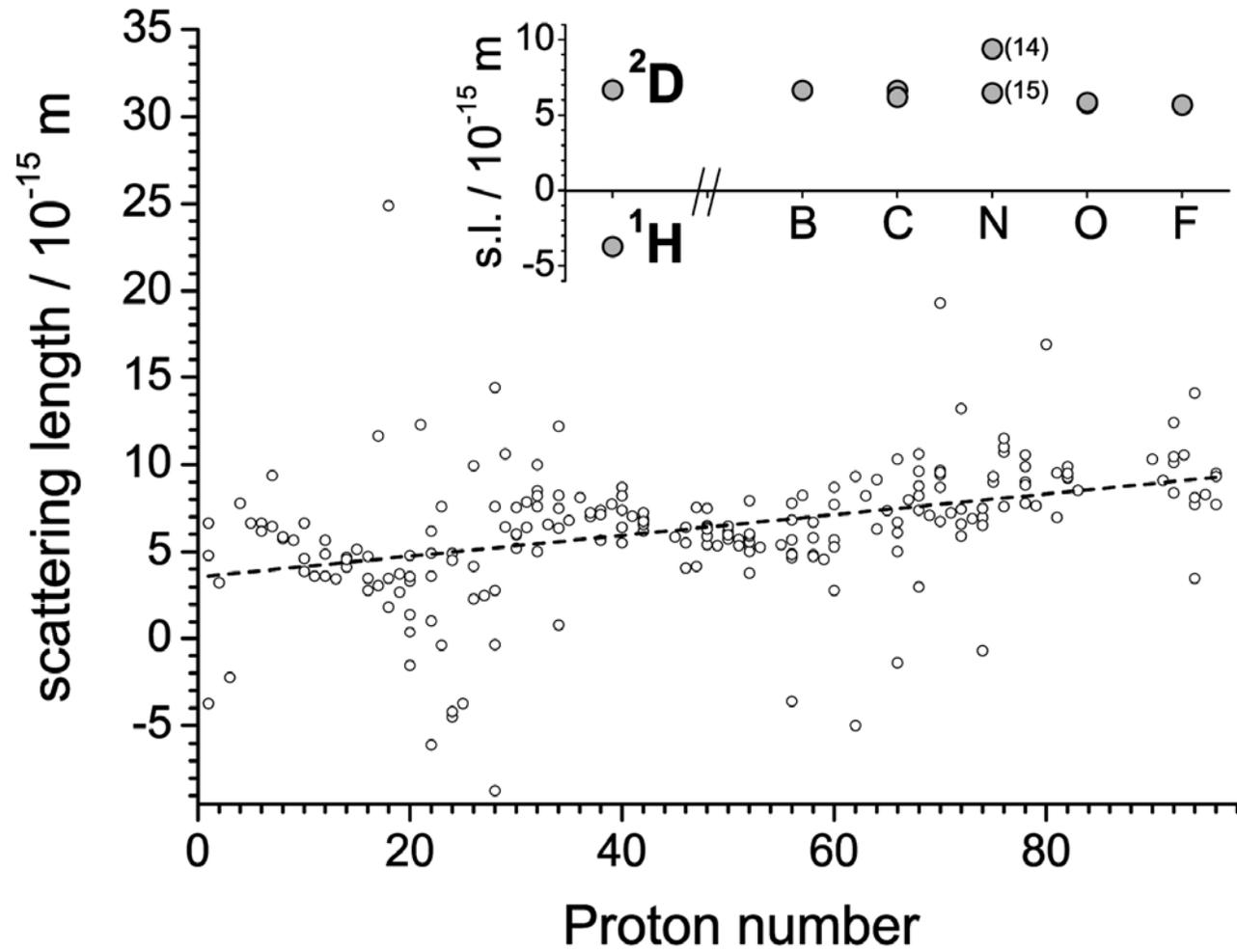
X-rays		Neutrons	
○	H/D	⊖	⊕
○	C	○	○
○	O	○	○
●	Ti	●	●
●	Fe	●	●
●	Ni	●	●

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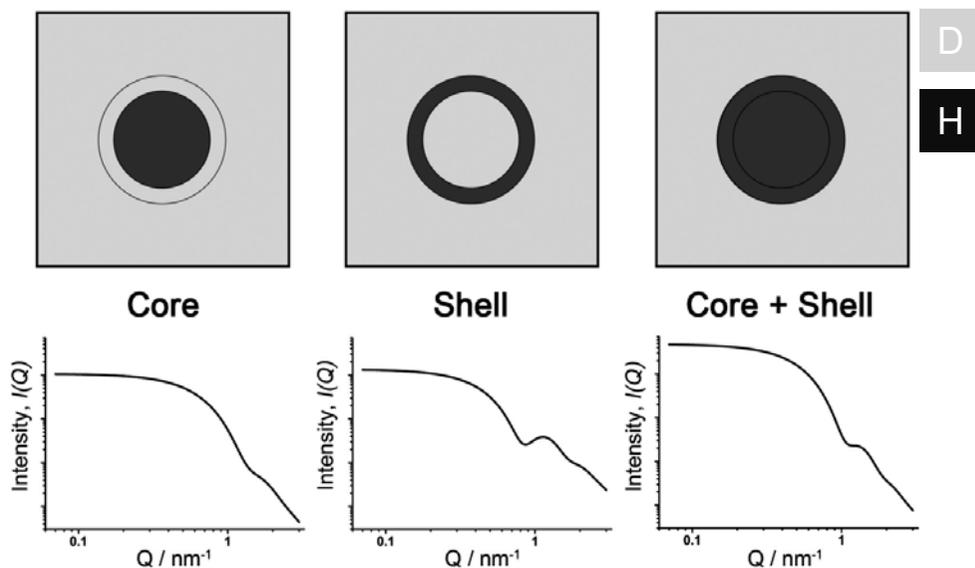
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# Scattering length



# Contrast variation

- Adjusting scattering lengths



Hollamby, *Phys.Chem.Chem.Phys.* **15**, 10566 (2013)

$$\langle b \rangle = x b_D + (1 - x) b_H - b_0$$

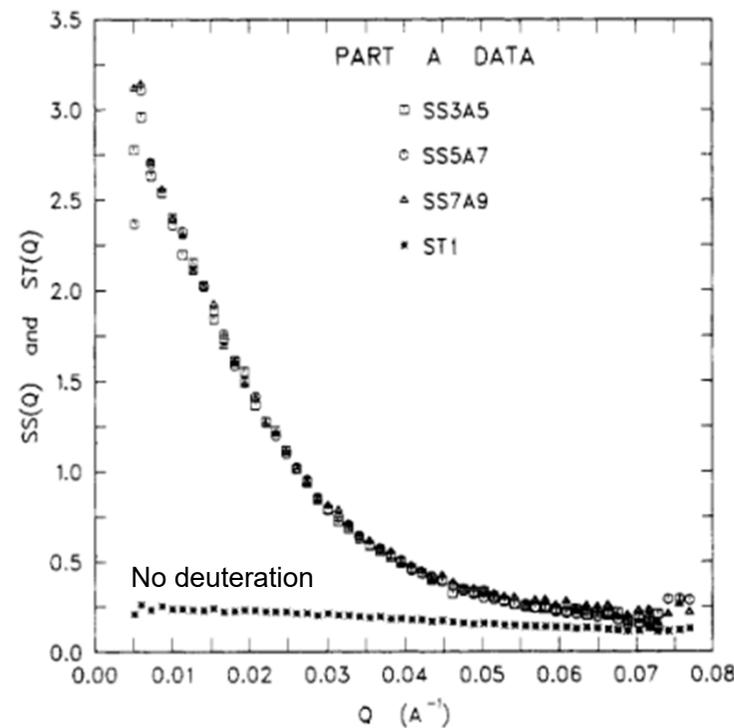
$$I(q) = (b_D - b_H)^2 x(1 - x) N z^2 P(q)$$

Benoît, Higgins, "Polymers and neutron scattering",

Oxford Science Publications, Oxford, 1996

DESY. Introduction to small-angle scattering | Stephan V. Roth, 13.05.2019

- Example:  
Polystyrene ( $C_8H_{8-x}D_x$ )<sub>n</sub> in toluene ( $C_7D_8$ )
- Concentration PS: 8%

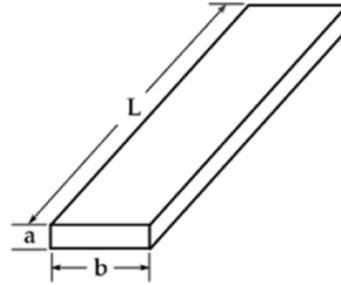


King et al., *Macromolecules* **18**, 709 (1985)

# Cellulose Nanofibrils in D<sub>2</sub>O

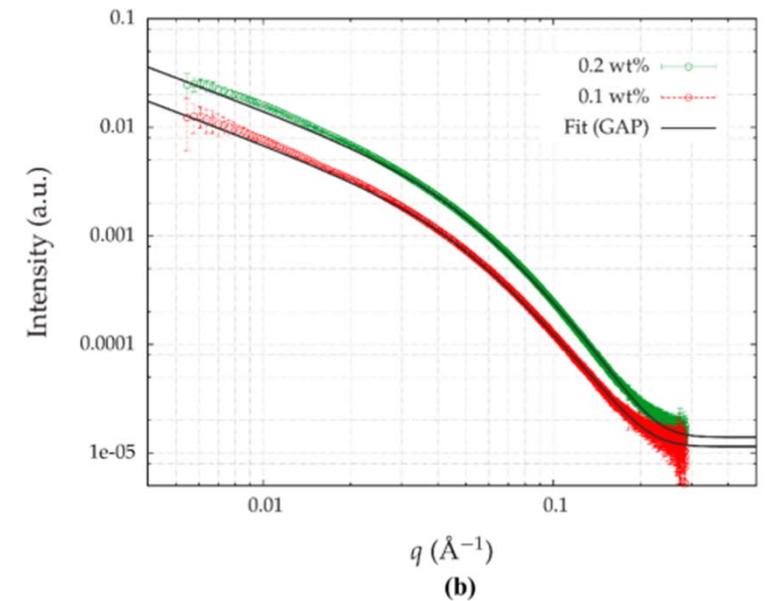
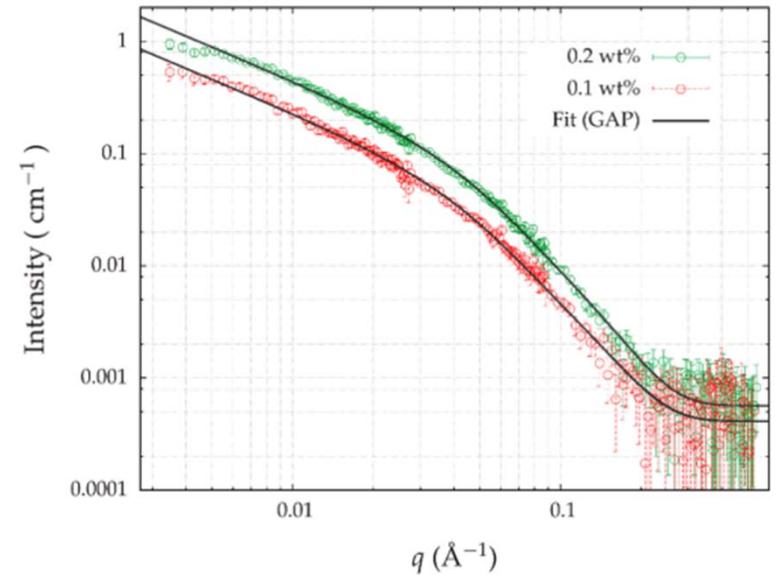
## SANS & SAXS

Mao et al., J. Phys. Chem. B 121, 1340 (2017)



$$I(q) = \frac{(ab)^2 L}{2} \frac{1}{q} \text{sinc}^2(qa/2) \int_0^{2\pi} \text{sinc}^2(qb \sin \phi/2) d\psi$$

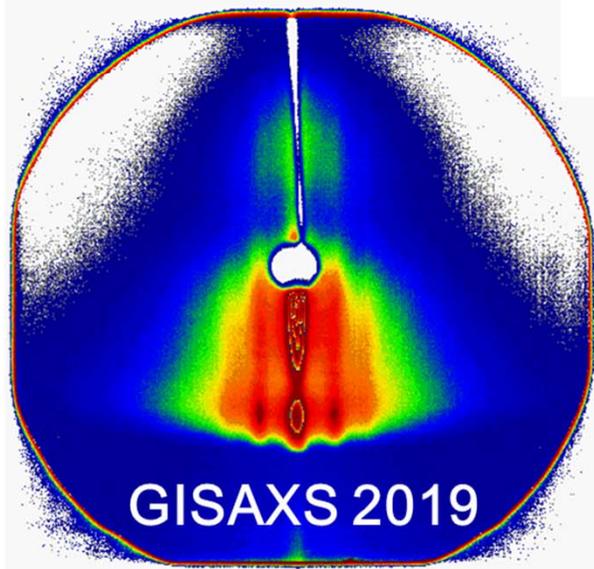
$a$ (nm)/SANS	$a$ (nm)/SAXS	$b$ (nm)/SANS	$b$ (nm)/SAXS
2.7 (0.18)	3.5 (0.45)	22.0 (0.37)	22.3 (0.40)
2.7 ( $9.2 \times 10^{-2}$ )	3.1 (0.36)	20.8 (0.21)	24.0 (0.31)
2.9 ( $3.9 \times 10^{-2}$ )	3.0 (0.31)	19.4(0.12)	24.8 (0.27)



# Summary

## Introduction to small-angle scattering – the use of X-rays and Neutrons

- Determination of nanoscale lengths, shape and orientation
- Guinier radius, Porod law, Porod constant, Orientation
- Kinetics: Drying, microfluidics (orientation), ...
- SAXS/SANS: identical quantitative results
- Contrast variation



organized by  
S.V. Roth (DESY, KTH)

&

P. Müller-Buschbaum (TU München)

# November 20-22, 2019

## Keynote speakers

*Harald Ade*, Northwestern University (US)

*Alexander Hexemer*, ALS, Berkeley (US)

*Moonhor Ree*, POSTECH (KOR)

*Frank Schreiber*, Univ. Tübingen (GER)

## Invited speakers

*Philippe Fontaine*, SOLEIL, Gif-Sur-Yvette (FRA)

*Philipp Gutfreund*, ILL, Grenoble (FRA)

*Emanuel Kentzinger*, FZ Jülich (GER)

*Peter Müller-Buschbaum*, TU München (GER)

*Gennady Pospelov*, JCNS (GER)

*Adrian Rennie*, Univ. Uppsala (SWE)

*Stephan V. Roth*, KTH, Stockholm, & DESY, Hamburg (SWE&GER)

*Mark Rutland*, KTH, Stockholm (SWE)

*Matthias Schwartzkopf*, DESY, Hamburg (GER)

*Peter Siffalovic*, Slovak Academy of Sciences, Bratislava (SK)

Keynote & Invited lectures, contributed poster sessions, visit to PETRA III & hands-on training: From the effective interface approximation to theory and simulation

<http://gisaxs2019.desy.de>