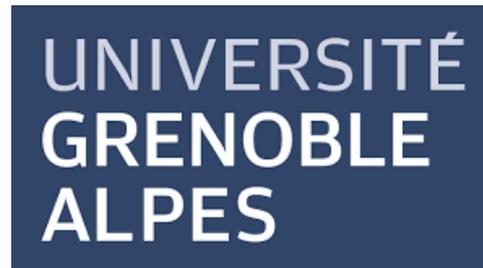


# Lattice dynamics and thermal conductivity in complex metallic alloys

Pierre-François Lory

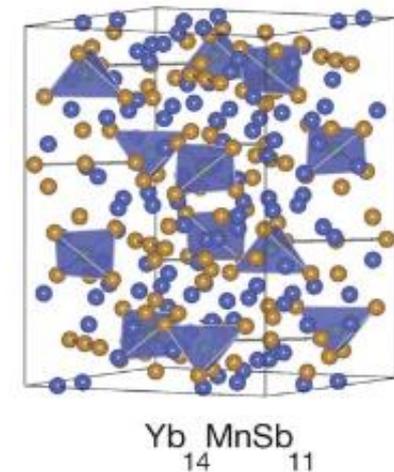
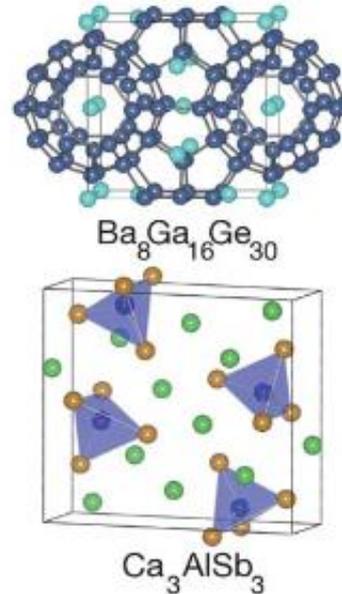
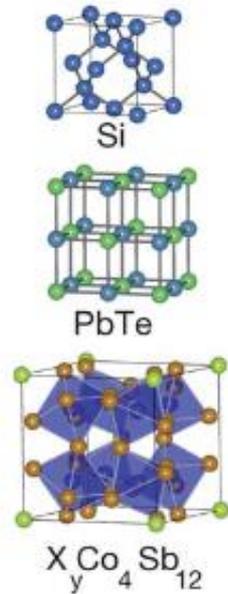
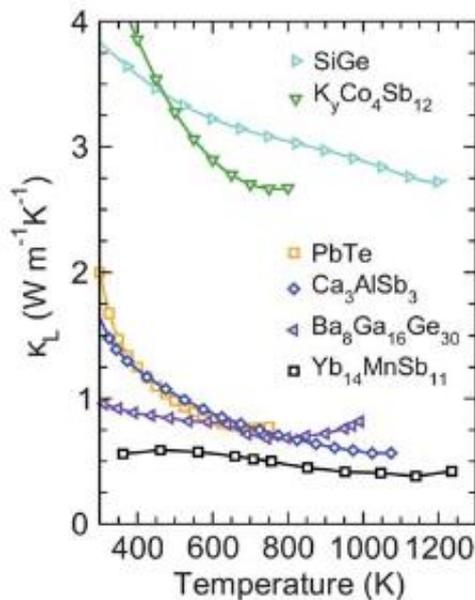
Presentation is based on my PhD work at Institute Laue-Langevin and SIMAP  
Grenoble, FRANCE



# Context and problematic

## Thermal conductivity ( $1 \text{ W/m.K}$ ) $\leftrightarrow$ Clathrates & Quasicrystals

- Effect of structural complexity and dynamics (phonons) on thermal properties
- Various ways of reducing  $K_{\text{ph}}$ : complexity, disorder, clusters/cages,...



## 1) Introduction

1.1) Lattice dynamic and thermal conductivity

1.2) Phonon lifetime

1.3) Inelastic Neutrons Scattering

1.4) Simulations - methods

1.5) DFT - Pair potentials

## 2) Approximant o-Al<sub>13</sub>Co<sub>4</sub>

3.1) Structure (complexity)

3.2) Lattice dynamics (phonons propagations)

3.3) Thermal conductivity (Simulation)

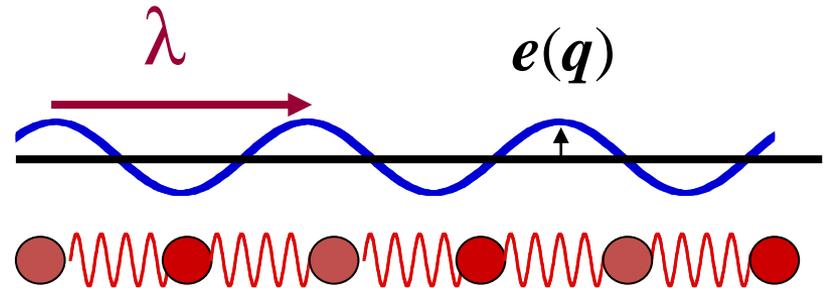
## 3) Conclusion & perspectives

# 1) Introduction

- 1.1) Lattice dynamic (Phonons) and thermal conductivity
- 1.2) Phonon lifetime
- 1.3) Inelastic Neutrons Scattering
- 1.4) Simulations - methods
- 1.5) DFT - Pair potentials

# 1.1) Lattice dynamic (Phonons) and thermal conductivity

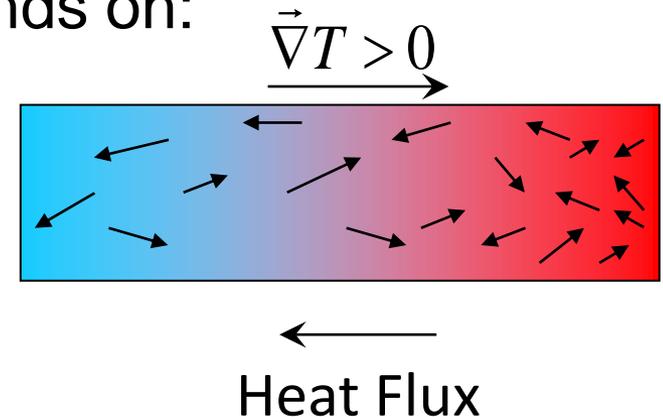
- Phonon or lattice **wave**:



- Lattice thermal conductivity ( $\kappa_{ph}^l$ ) depends on:

a) **DISPERSION** relation :  $v_{ph}(\omega)$

b) **LIFETIMES** of modes :  $\tau(\omega)$



$$l_{ph}(\omega) = v_{ph}(\omega) \tau_{ph}(\omega)$$

$$\kappa_{ph}^l = \frac{1}{3} \int C_V(\omega) v_{ph}^2(\omega) \tau_{ph}(\omega) n(\omega) d\omega$$

## 1.2) Phonon lifetime

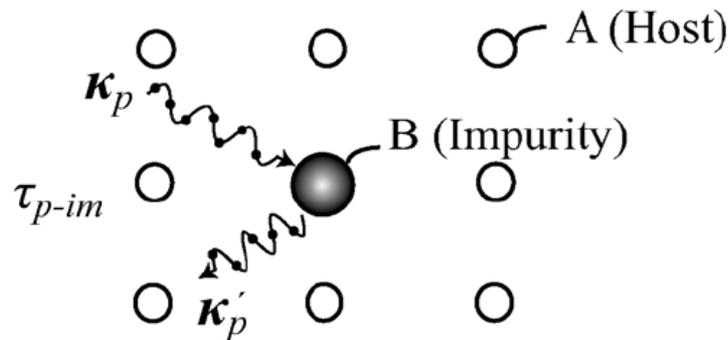
Structure (complexity, disorder) → limits thermal conductivity

Mechanism 1:  
disorder/impurity

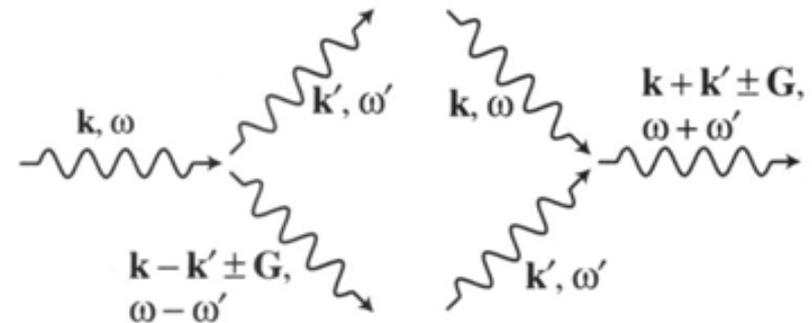
- T independent

Mechanism 2:  
3 phonons

- Due to anharmonicity
- T dependent,  $\kappa_{ph} \sim 1/T$



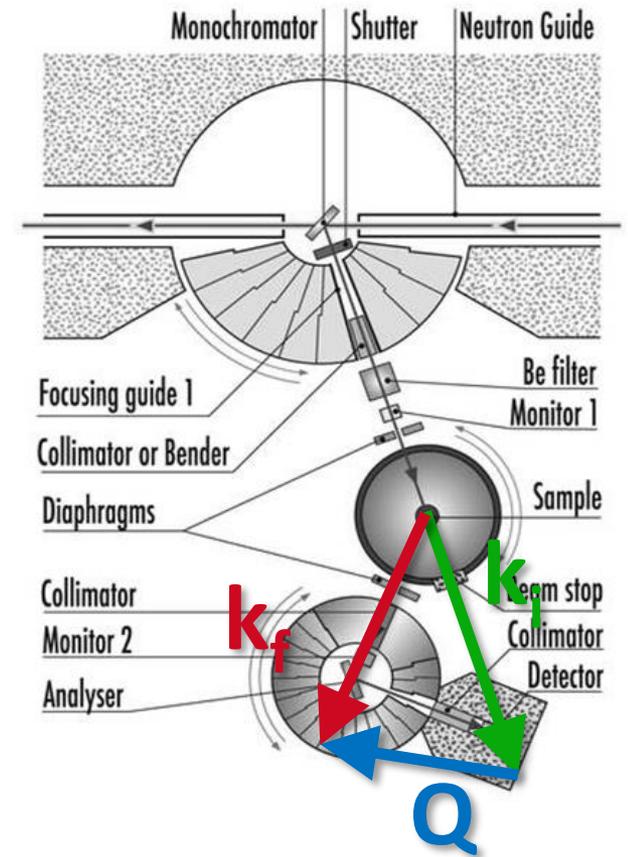
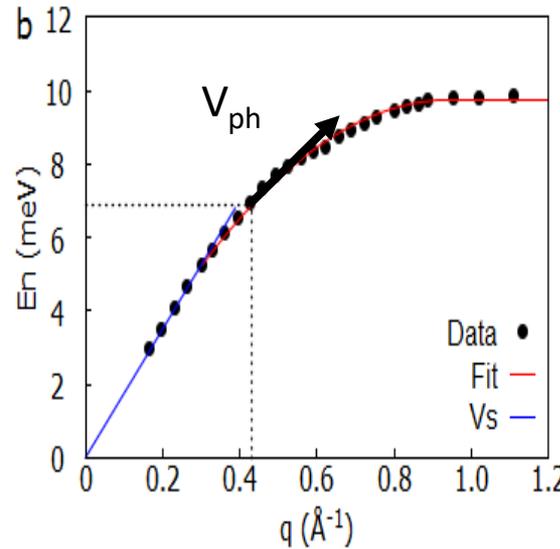
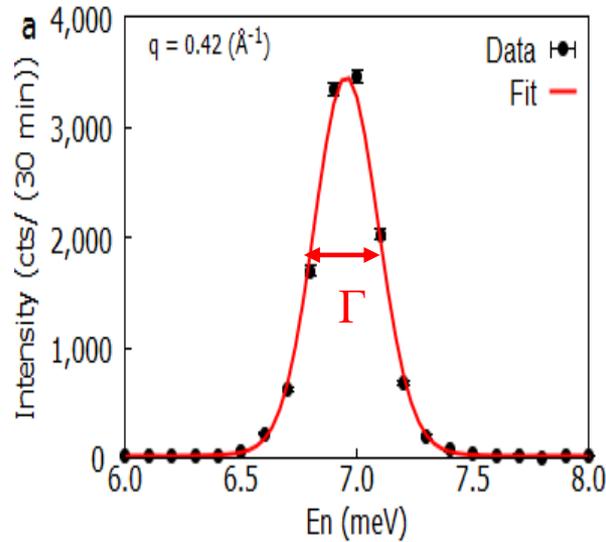
(b) Phonon-Impurity (B) Scattering



# 1.3) Inelastic Neutron Scattering

- Width  $\Gamma$   $\rightarrow$  Phonon lifetime  $\tau$
- $E(q)$ : dispersion  $\rightarrow V_{ph}$

$$\tau(\omega) = \frac{2\hbar}{\Gamma(E)_{FWHM}}$$



# 1.4) Simulations - methods

Goal : Calculate Phonon and Thermal conductivity

- 1) Normal modes - Harmonic approximation
- 2) Molecular Dynamics - includes Anharmonic terms

Normal modes	Molecular Dynamic (MD)
T = 0 K (optimized structure)	80 K < T
Normal modes / phonons	Trajectories of atoms
Supercell (~10 Å) or < 800 atoms	Superbox (~10 Å to few nm) or 10 <sup>6</sup> atoms
Quantum effects	Classic approximation
<b>Calcul S(Q,ω) ∀ q point</b>	<b>Calcul S(Q,ω) ∀ q point commensurate with the superbox</b>



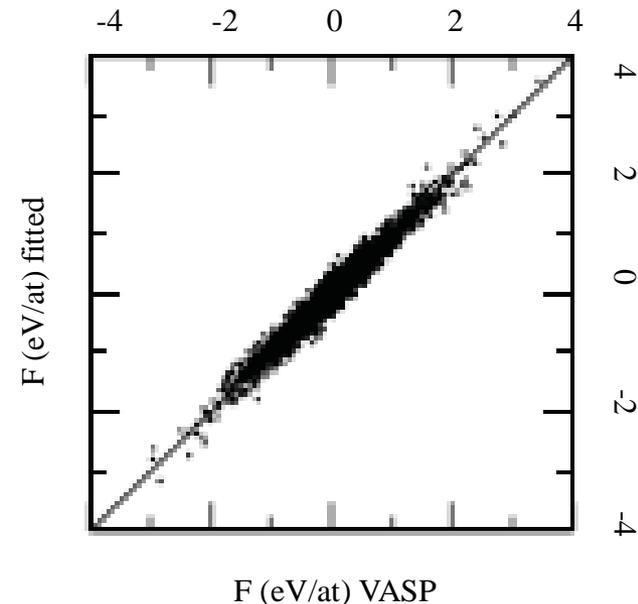
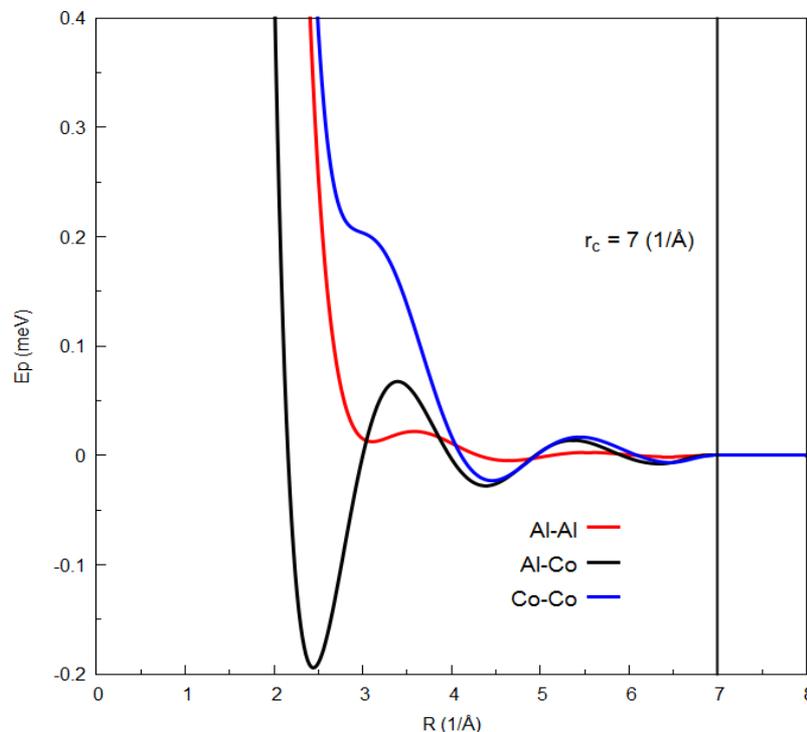
Molecular  
Dynamics  
Simulator



# 1.5) DFT and Pair potentials

- DFT  $N_{\text{atoms}} < 800$
- Pair Potentials : to describe interactions of the Al-TM
- “Force Matching” method from M. Mihalkovic and *al.*[2]

$$V(r) = \frac{C_1}{r^{\eta_1}} + \frac{C_2}{r^{\eta_2}} \cos(k_* r + \phi_*)$$



- **Use with MD simulations**

[2] M. Mihalkovic & C. L. Henley, Phys. Rev. B. 85, 9, 092- 102, (2012)

# Approximant-crystal



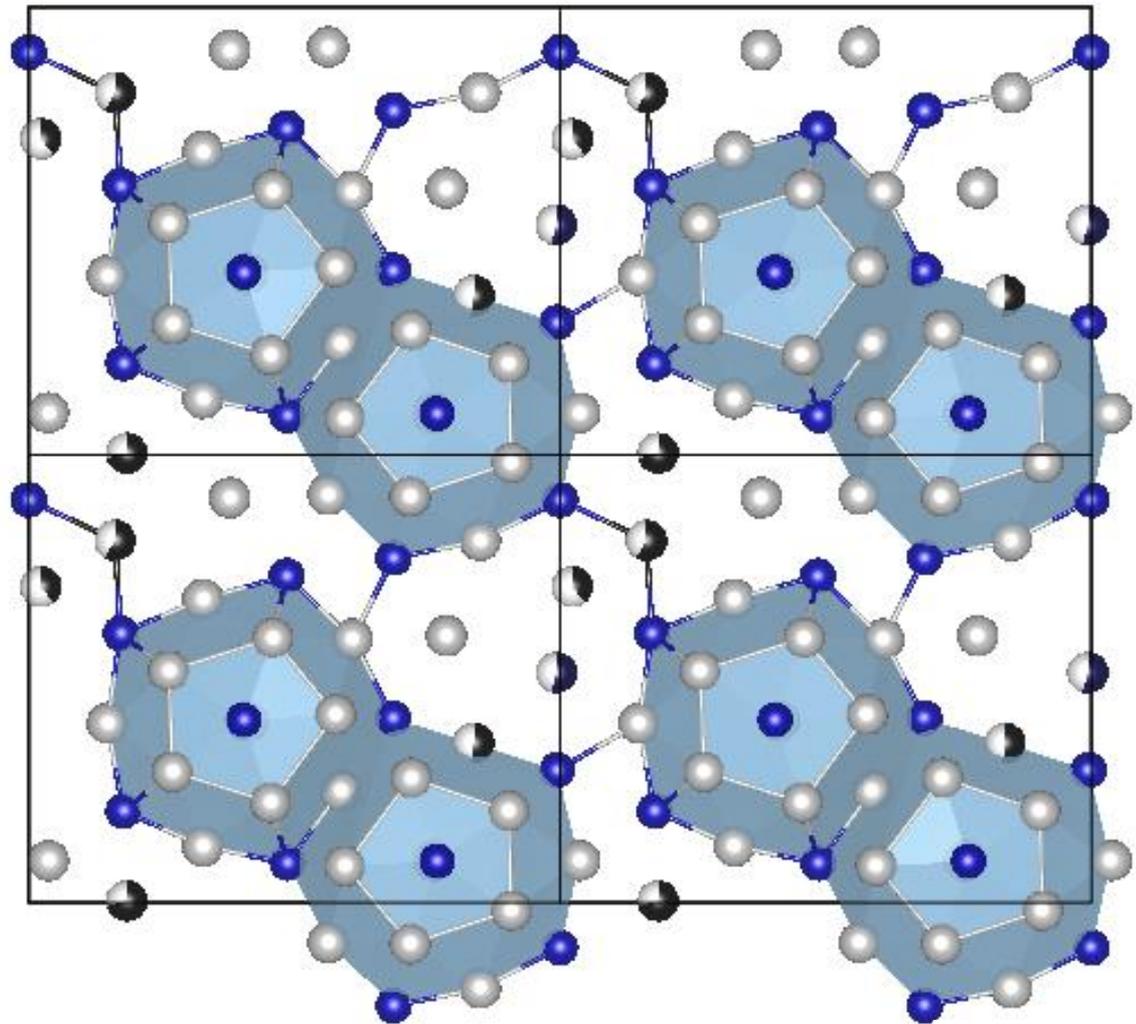
2.1) Structure of approximant  $\text{o-Al}_{13}\text{Co}_4$

2.2) Lattice Dynamics  
(experiment and simulation)

2.3) Thermal conductivity model

## 2.1) Structural models for o-Al<sub>13</sub>Co<sub>4</sub>

- Local structure: cluster (4.6 Å)
- [100] **periodic** and [010] and [001], **pseudo-quasiperiodic**
- Unit cell: orthorhombic (a = 8.15 Å, b = 12.34 Å, c = 14.45 Å)

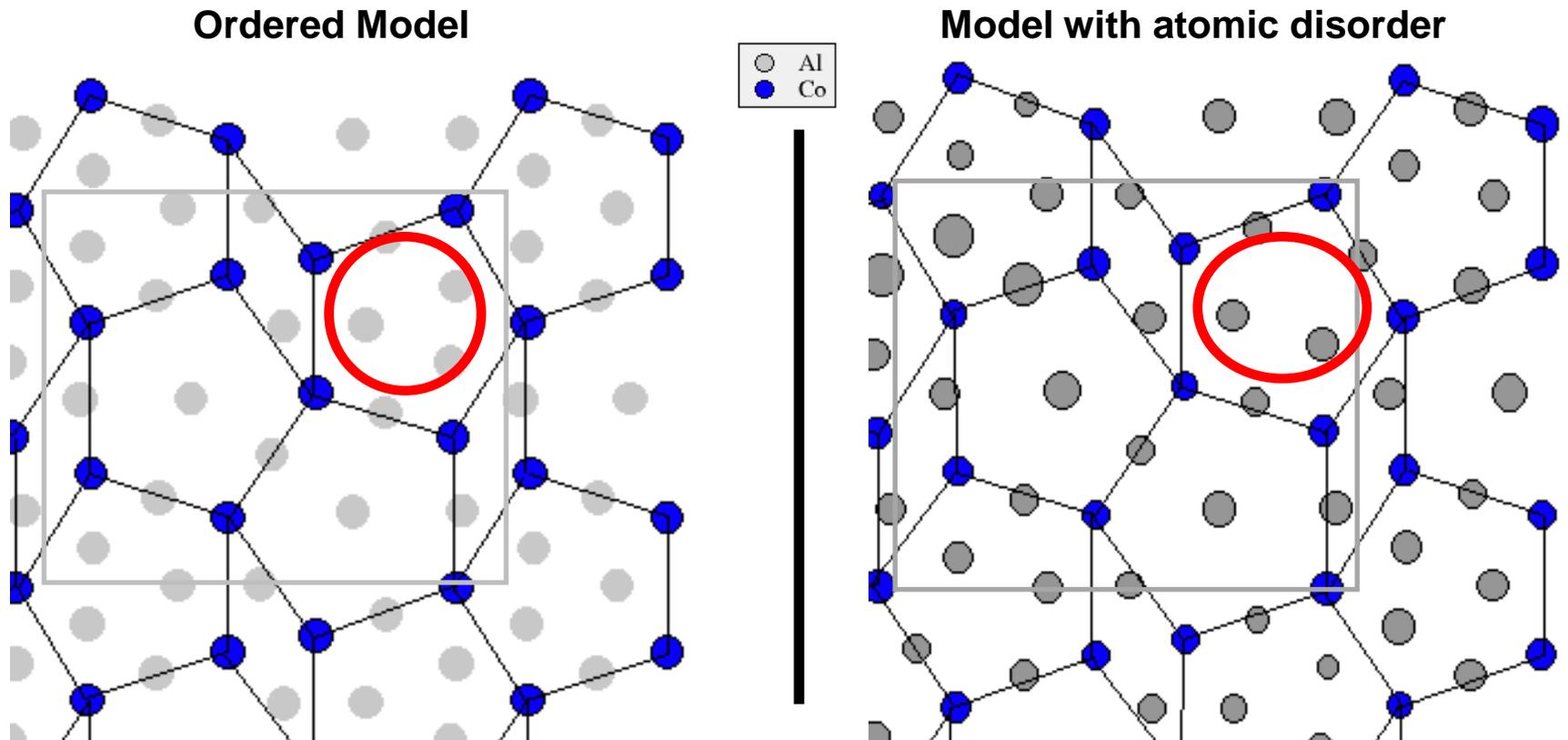


Picture of the (b,c) plane, it is the pseudo quasiperiodic plane



## 2.1) Numerical model with disorder (J. Grin 2015)

- Model 1: SC(1×1×1): 102 atoms  
**ORDERED MODEL [1994]**
- Model 2, 1 atom/site : SC(3×2×2): *1205 atoms or 100 atoms unit cell* **ATOMIC DISORDER [2015]**



# Approximant-crystal



2.1) Structure of approximant  $\text{o-Al}_{13}\text{Co}_4$

2.2) Lattice Dynamics

(experiment and simulation)

Method 1 : Harmonic approximation

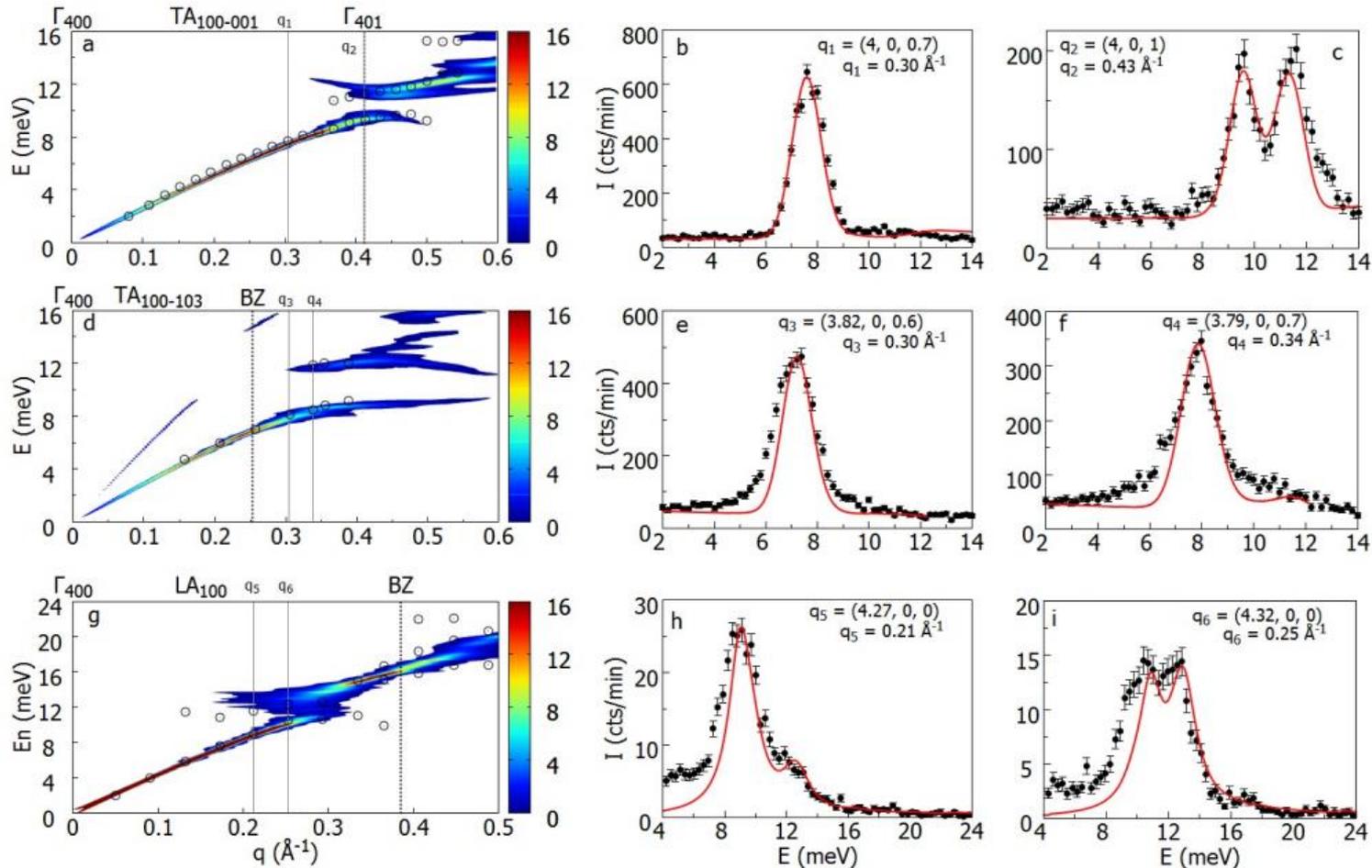
Method 2 : Classical MD

2.3) Thermal conductivity model

# 2.2) INS and harmonic simulation (Method 1)

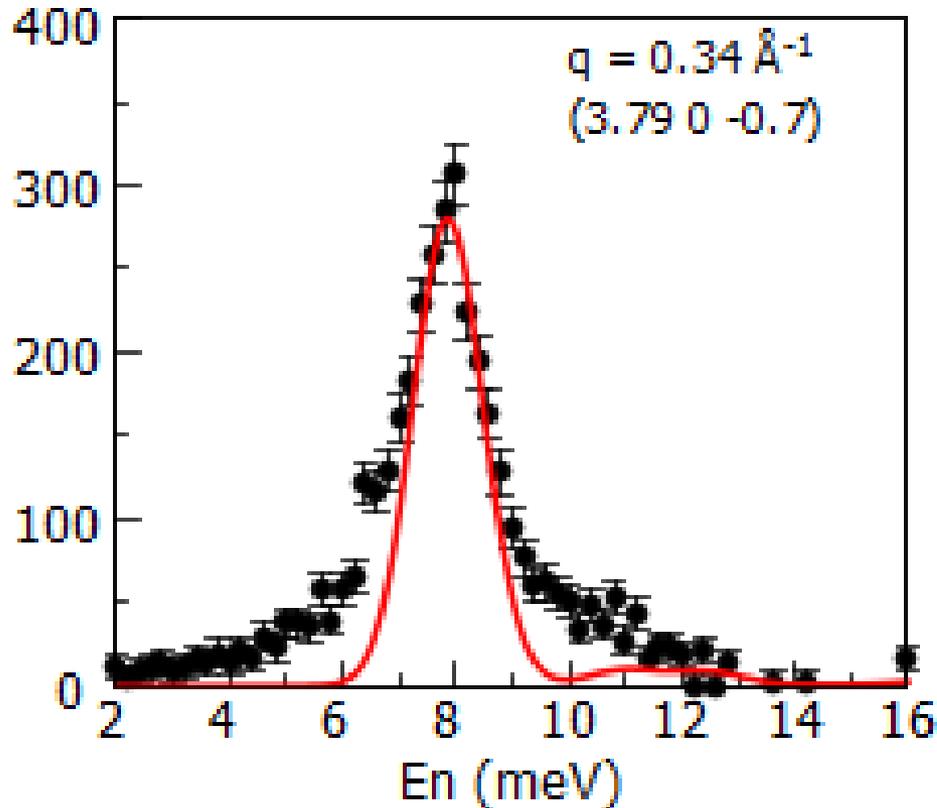
Experiment (INS, dots)  
Harmonic simulation (red lines)  
SIMULATION  $\otimes$  RESOLUTION

**GOOD AGREEMENT**  
experiment-simulation

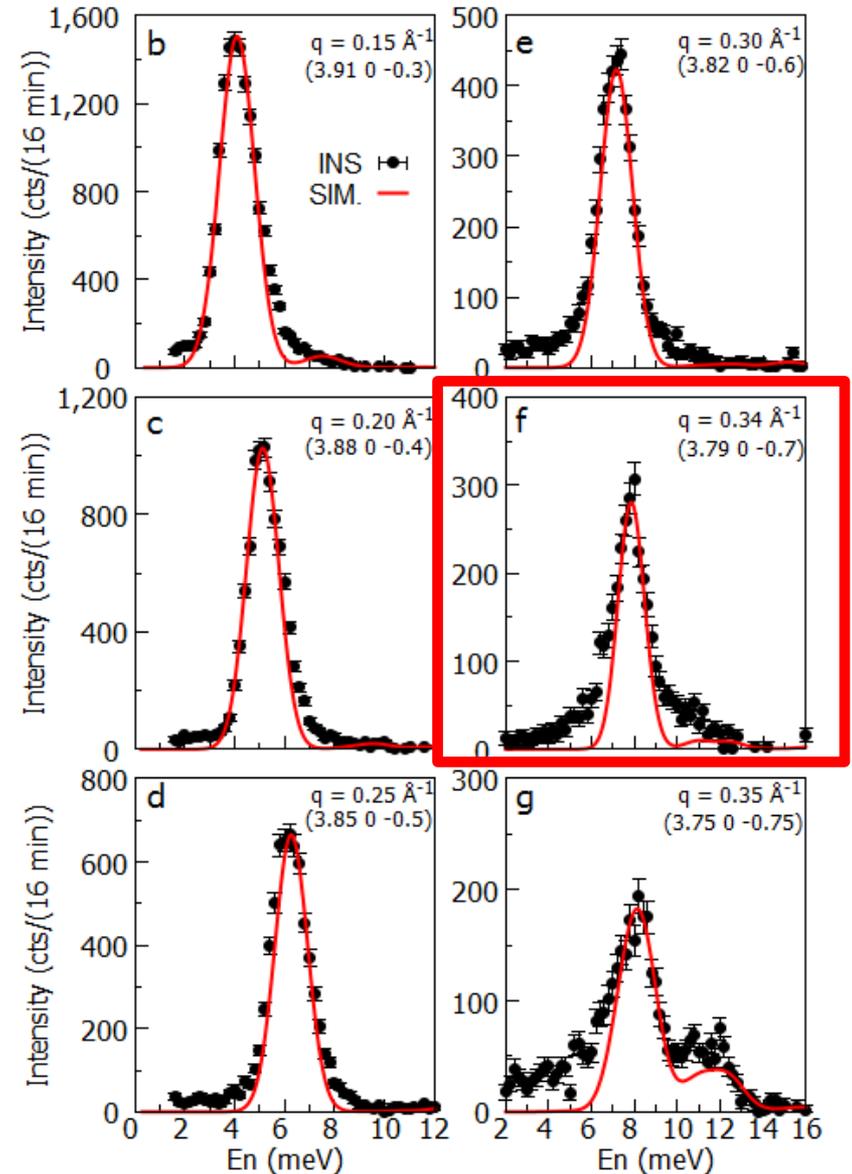


## 2.2) Low symmetry (400)-(206)

Experiment (INS, dots)  
Harmonic simulation (red lines)



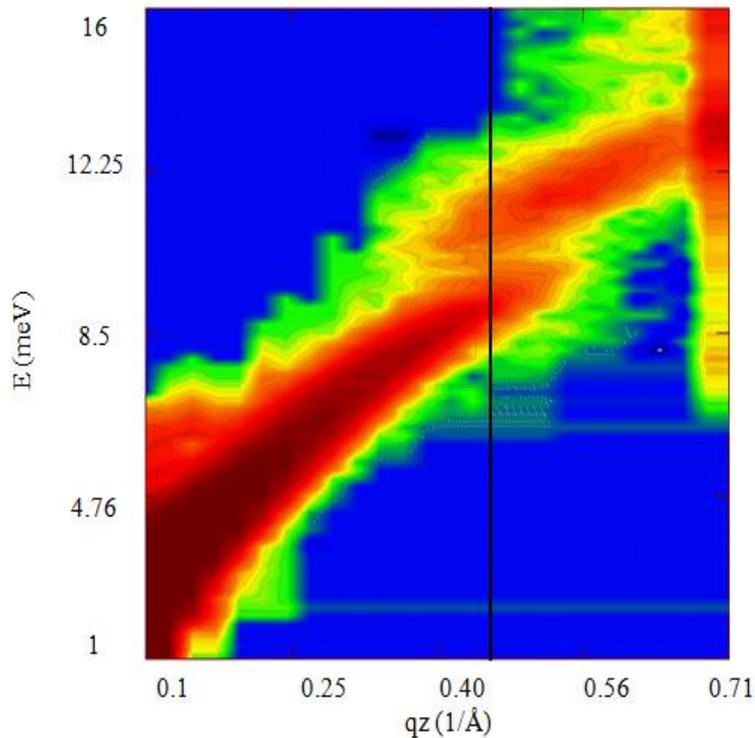
Experiment: PHONON DHO at  $q=0,34 \text{ \AA}^{-1}$   
 $\Gamma_E = 0.5 \text{ meV} \approx \tau = 2.63 \text{ ps}$



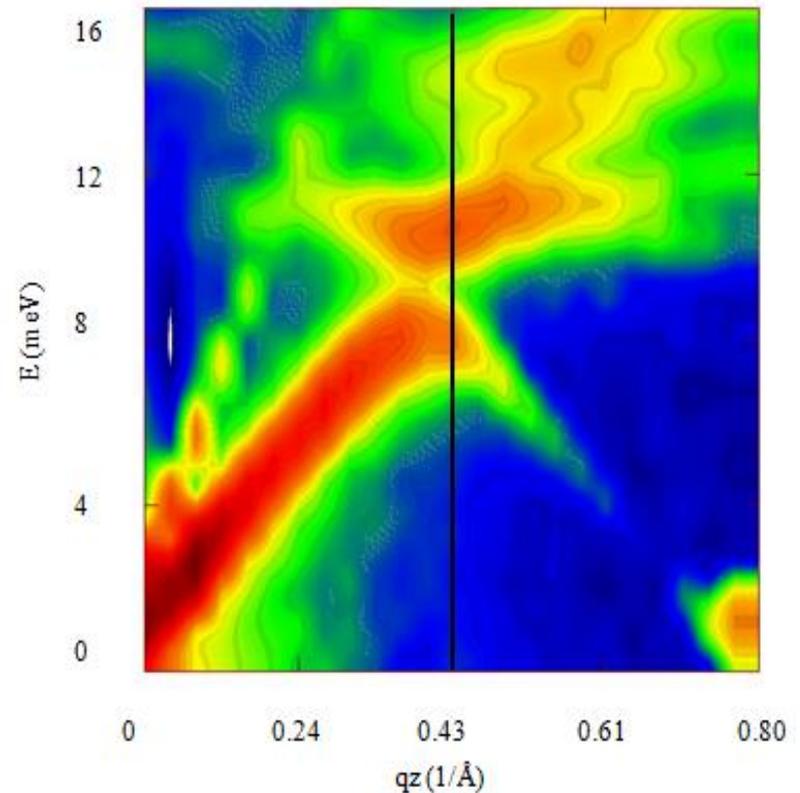
Lattice dynamics:  
Experiment and  
Molecular dynamics simulations  
(LAMMPS + data analysis with  
nMoldyn

## 2.2) INS and MD Simulations

Inelastic Neutron Scattering  
(IN22)



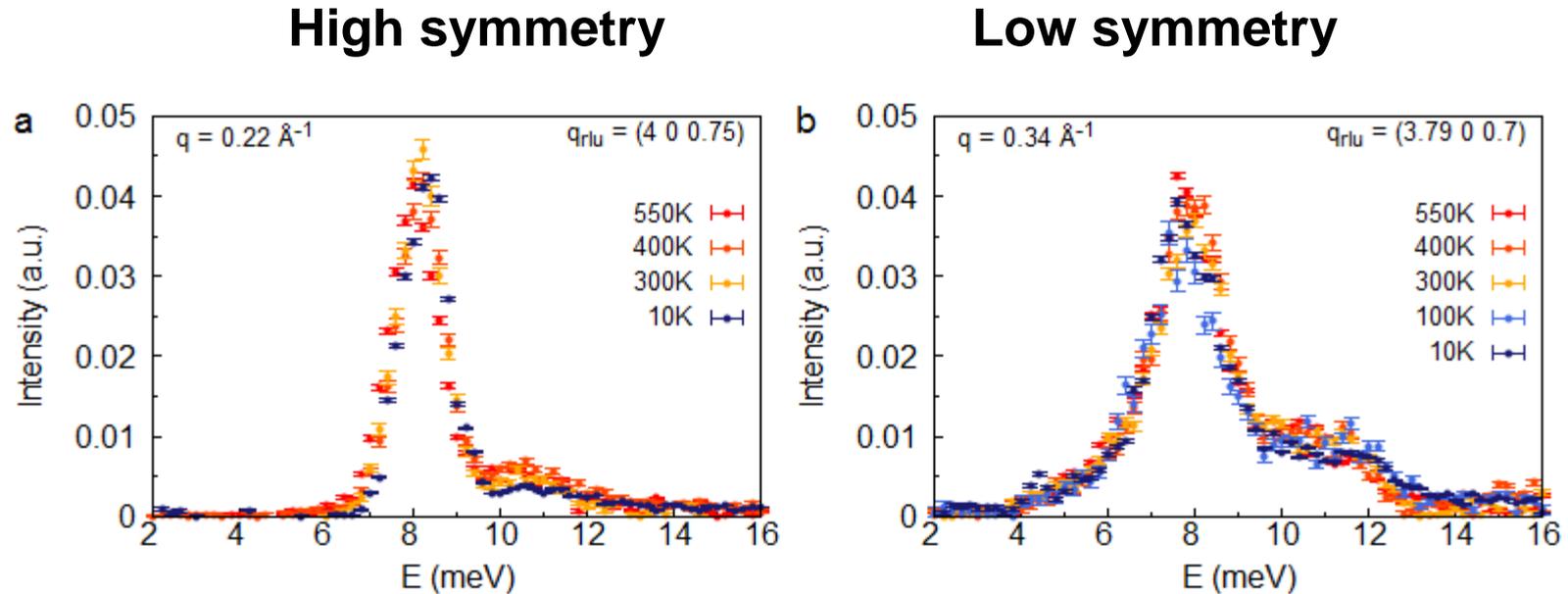
Molecular simulation  
(LAMMPS and nMoldyn)



- MD simulation reproduce very well the phonon dispersion mapping (high symmetric direction)
- MD mapping has been convoluted by the instrumental resolution (Gaussian function)

## 2.2) Experiments vs temperature

$S(Q,w)$  from MD simulations, where we observe the DHO signature from INS



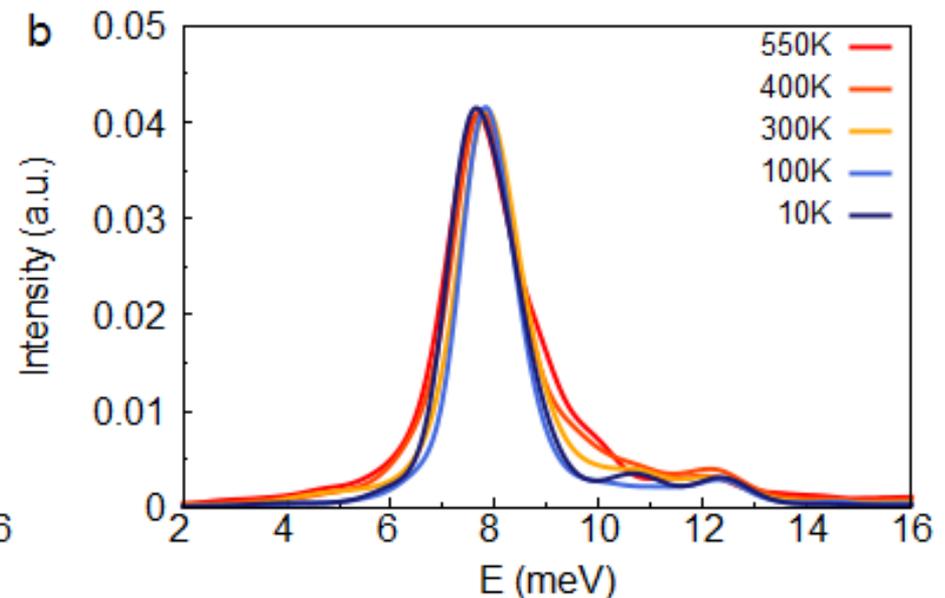
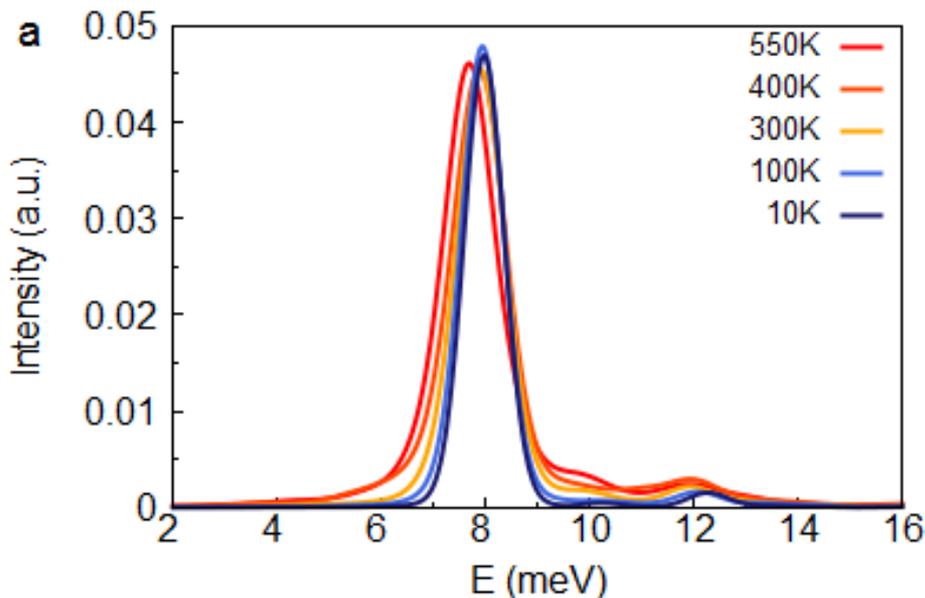
- Measure in large range of temperatures [10 K to 550 K], (TAS-IN22 + cryofurnace)
- Conclusion :  $\Gamma$  is not dependent on the temperature

## 2.2) MD Simulations vs temperature

MD of  $S(Q,\omega)$  low symmetry :  $q = 0.34 \text{ \AA}^{-1}$  (measured DHO)

**Model 1: ordered**

**Model 2: disordered**

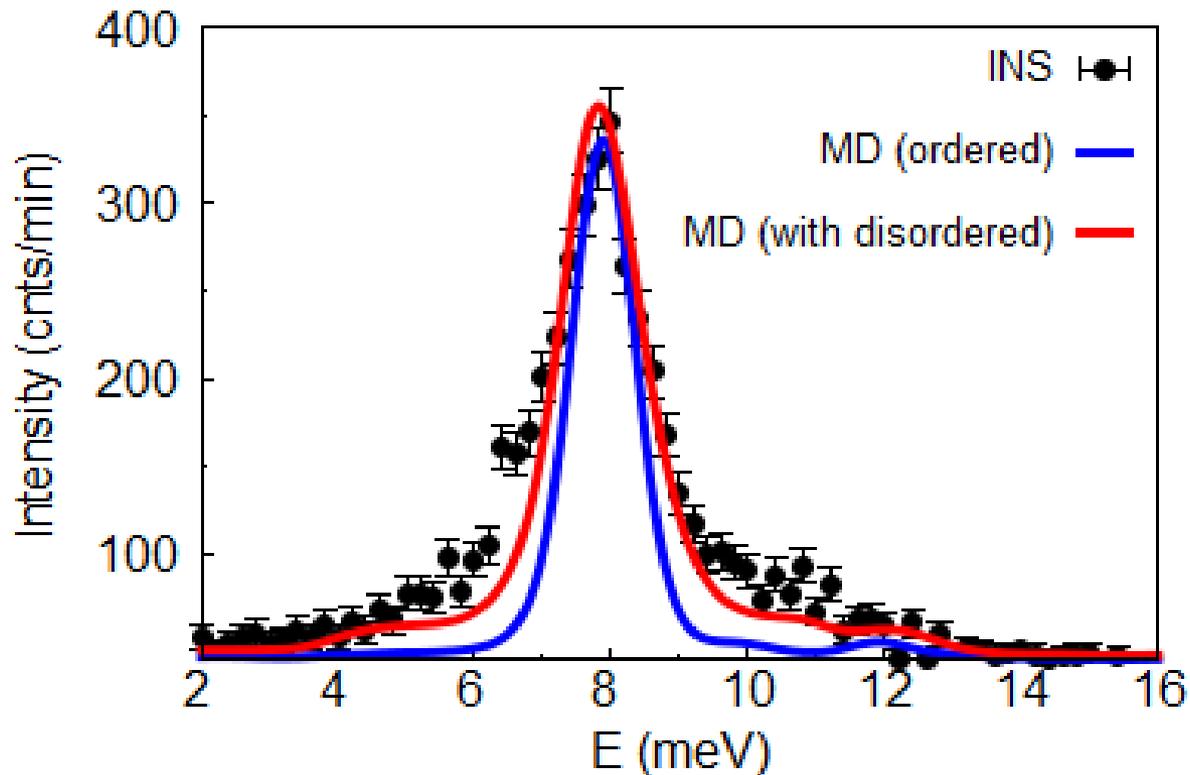


- Effect of T more important in model 1
- Weak temperature dependence

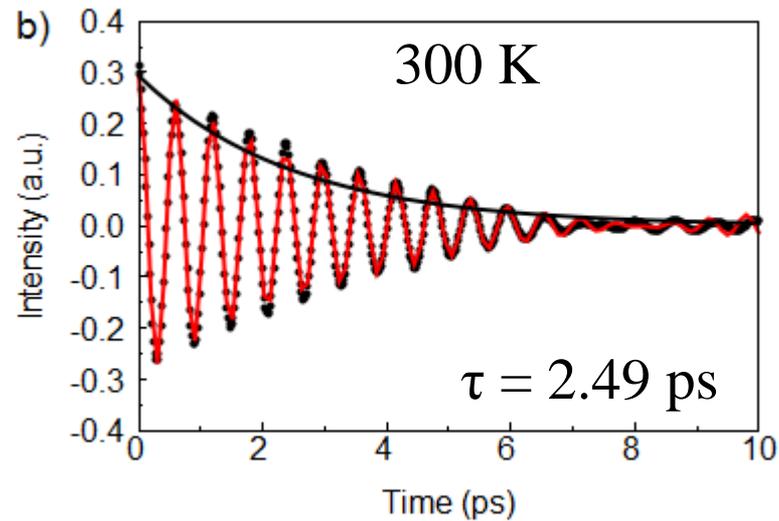
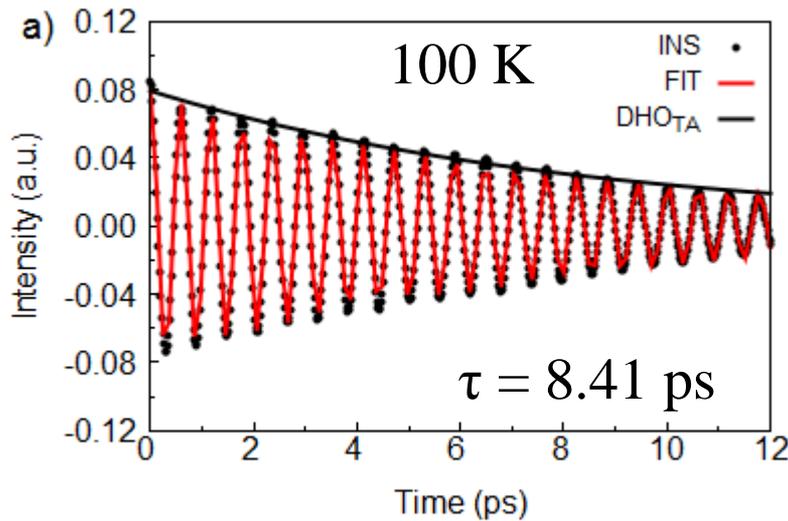
## 2.2) MD Simulations

- MD with **Anharmonic terms** using pair potentials
- Phonon with **finite lifetime** → **Model 2 with atomic disorder**

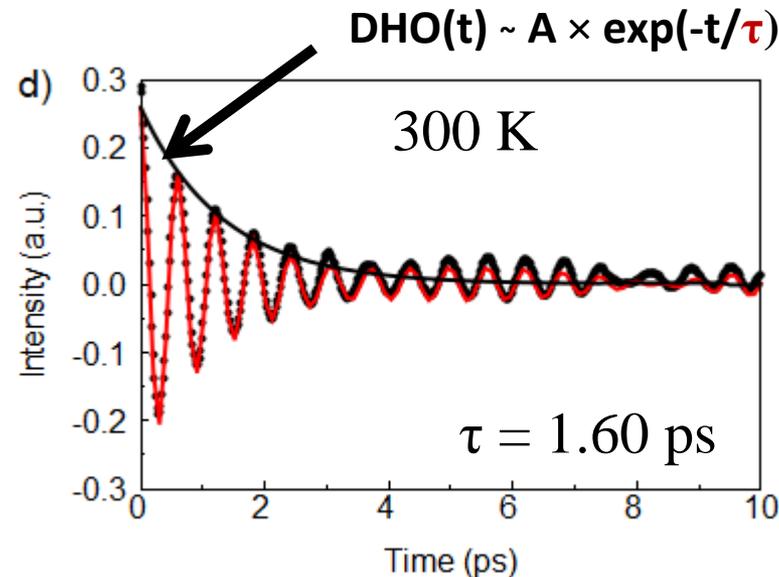
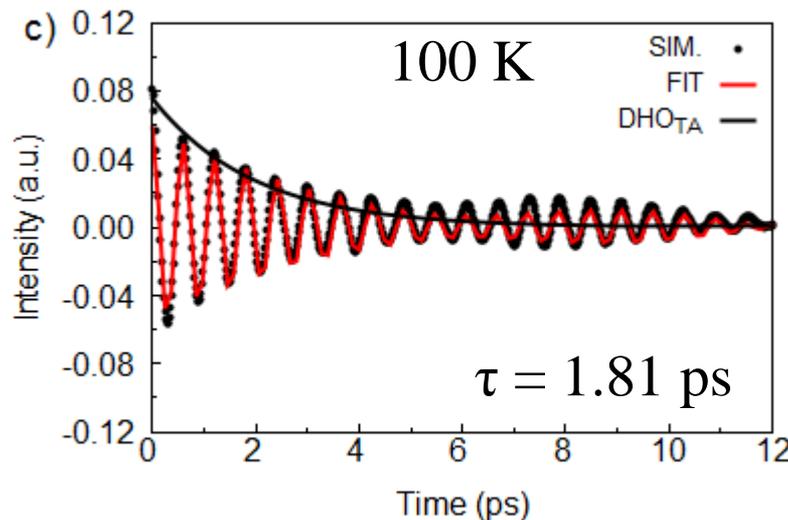
$S(Q, \omega)$  **low symmetry**:  $q = 0.34 \text{ \AA}^{-1}$



## 2.2) MD Simulations $S(Q,t)$ vs temperature



(ordered)  
Model 1



(disordered)  
Model 2

$S(Q,t)$  from MD simulations at  $q = 0.34 \text{ nm}^{-1}$ , where we observe the DHO signature in INS

# Approximant-crystal

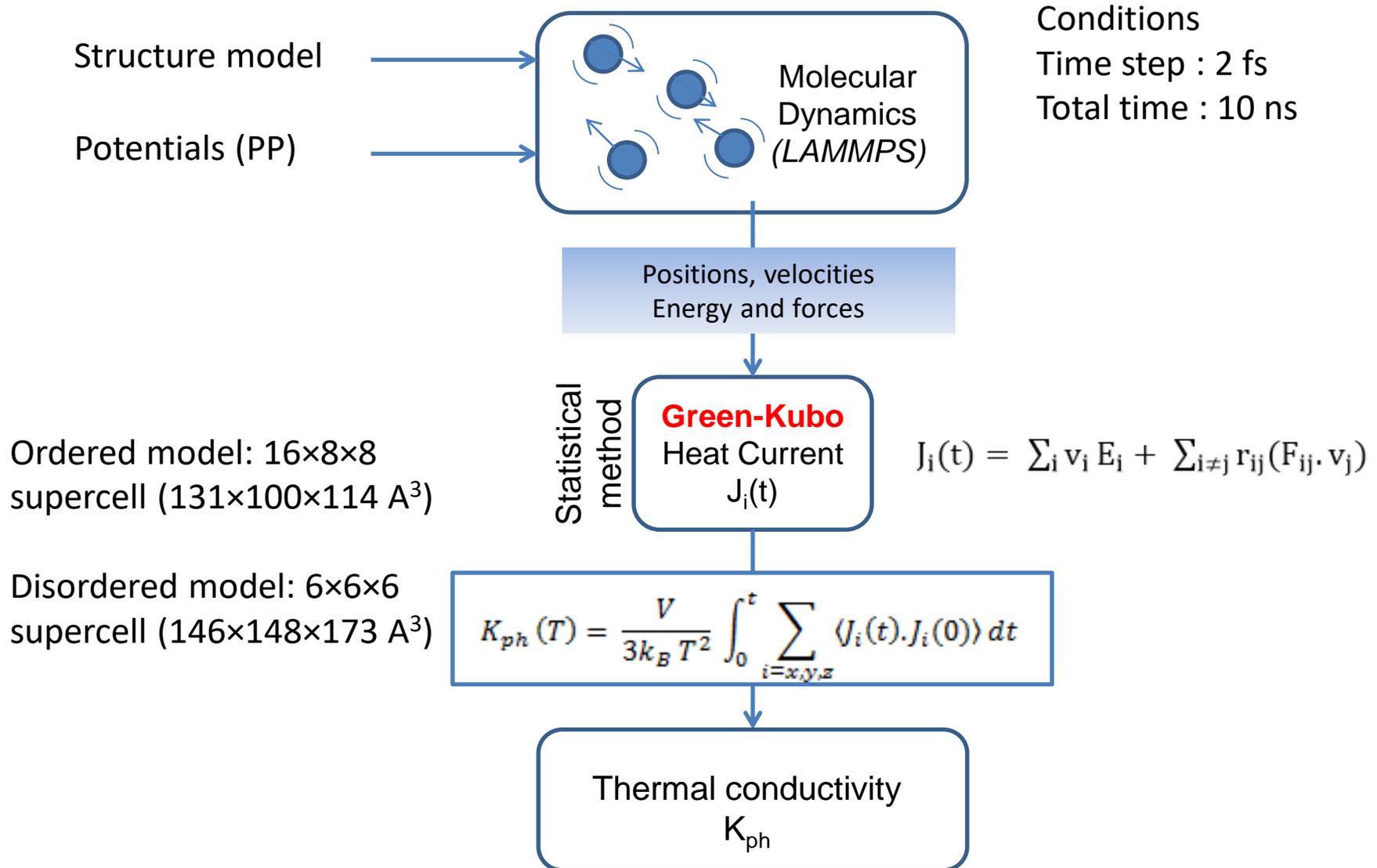


2.1) Structure of approximant  $\text{o-Al}_{13}\text{Co}_4$

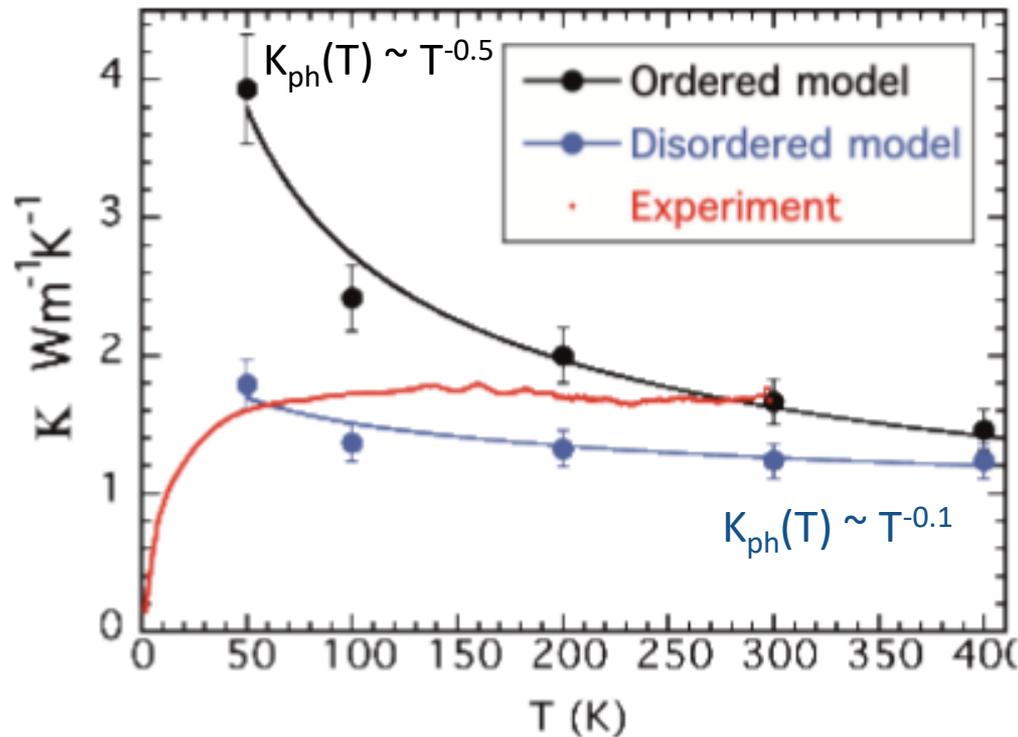
2.2) Lattice Dynamics  
(experiments and simulations)

2.3) Thermal conductivity simulation

# 3.3) Method of prediction of thermal conductivity MD-GK



### 3.3) Thermal conductivity MD-GK in o-Al<sub>13</sub>Co<sub>4</sub>



- For  $\mathbf{K}_{\text{total,ph}}$ , simulations coherent with experiment
- **Weak temperature** dependence (disorder model)

Measure: J. Dolinšek, M. Komelj, P. Jeglič, S. Vrtnik, D. Stanić, P. Popčević, J. Ivkov, A. Smontara, Z. Jagličić, P. Gille, Yu. Grin, *Phys. Rev. B*, **79**, 184-201, (2009)

# Conclusion

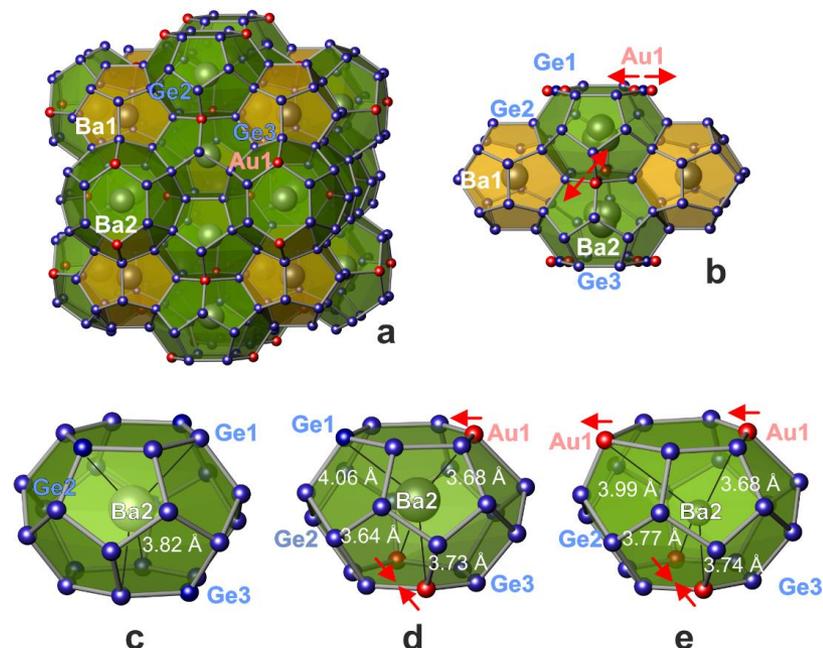
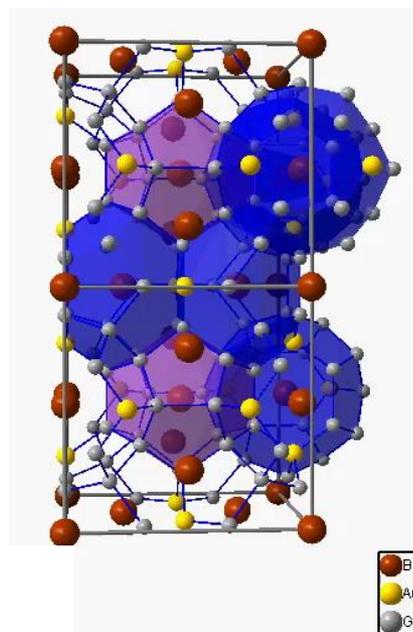
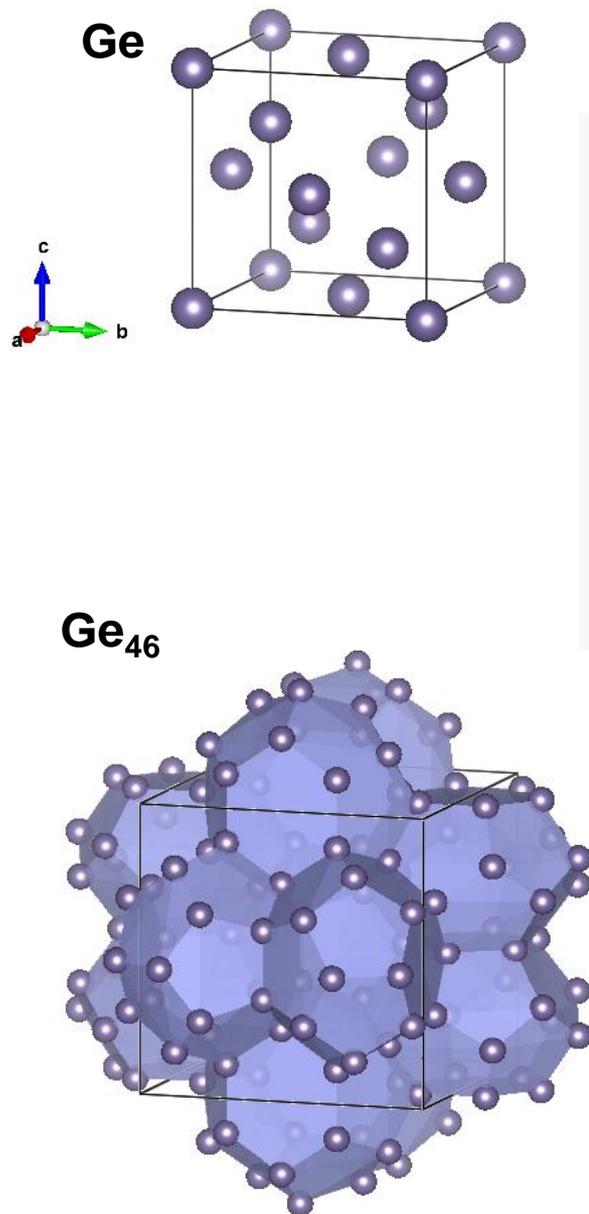
Structure → Lattice Dynamics → Thermal conductivity

For approximant-crystal o-Al<sub>13</sub>Co<sub>4</sub>

- **Finite phonon lifetime** due to **disorder**
- **Phonon lifetime** for acoustic mode : **33 ps**
- Mean free path **~25 nm >>> a = 10.8 Å**
- **Cluster structure and atomic disorder** → low and weak T-dependence of thermal conductivity in o-Al<sub>13</sub>Co<sub>4</sub>

Perspective : Investigate the thermal properties on the Clathrate Ba<sub>8</sub>Ge<sub>40.6</sub>Au<sub>5.25</sub> by Classical simulation and potentials

P.-F. Lory, V. M. Giordano, P. Gille, H. Euchner, M. Mihalkovič, E. Pellegrini, M. Gonzalez, L.-P. Regnault, P. Bastie, H. Schober, S. Pailhes, M. R. Johnson, Yu. Grin, and M. de Boissieu, Phys. Rev. B **102**, 024303, (2020)



- Investigate the thermal properties on the clathrate series by Classical molecular simulation
- Compare with the experimental work (Measure  $\tau_{ph} \sim [30 - 53]$  ps)  
Lory, P. F., Pailhès, S., Giordano, V. M., Euchner, H., Nguyen, H. D., Ramlau, R., ... & de Boissieu, M. (2017). Direct measurement of individual phonon lifetimes in the clathrate compound Ba 7.81 Ge 40.67 Au 5.33. *Nature communications*, 8(1), 1-10.

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**Thank you for your attention**