

PAUL SCHERRER INSTITUT



Stavros Samothrakis & Camilla B. Larsen :: Applied Materials Group :: Paul Scherrer Institute
Laue and time-of-flight neutron diffractive imaging methods for 3D
grain mapping of polycrystalline materials

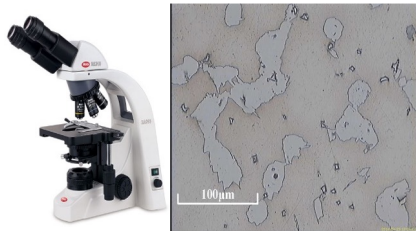
ESS ILL Joint User Meeting 2022

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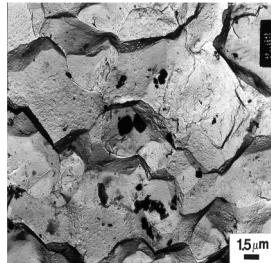
Introduction & Motivation

- Grain orientation distribution, size and shape, among other microstructural features are directly connected to macroscopic properties (mechanical, magnetic, etc.) of polycrystalline materials.
- Detailed knowledge of the crystallographic grain network, not only on the surface but also in the bulk of samples, is essential for tuning and manipulating macroscopic properties for applications.
- It is thus important to develop methods/tools that provide detailed information:
 - 2D and 3D,
 - Destructive and non-destructive,
 - Optical light, electrons, X-rays, neutrons

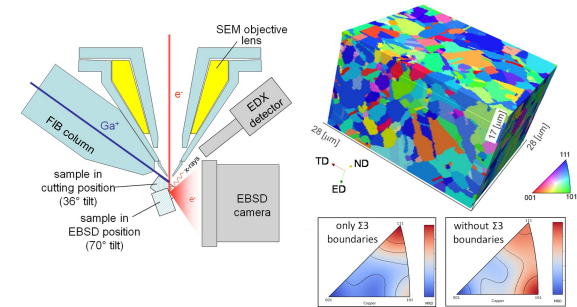
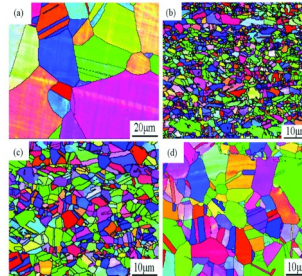
Optical Microscopy



SEM



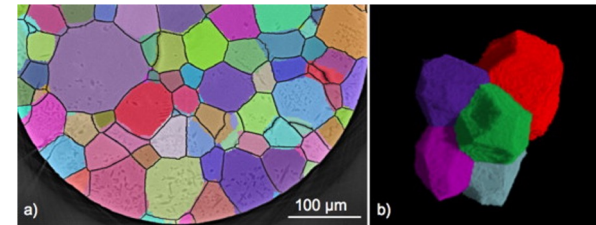
EBSD



3D EBSD

Figures taken from:
<https://www.mpie.de/3093283/Techniques3DEBSD>

3DXRD

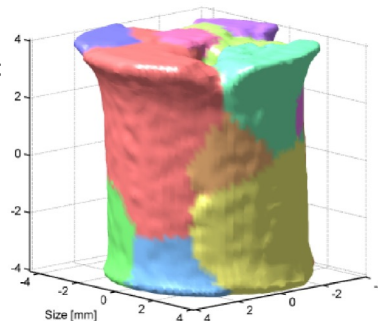


Materials Science and Engineering: A, Vol. 524, (2009), Pp. 69-76

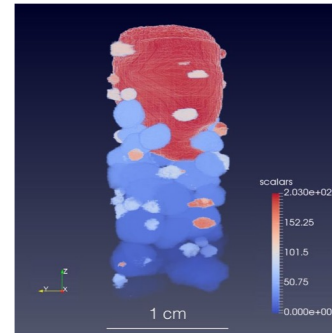
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Cold neutron
diffraction contrast
tomography



Analyst, 2014, 139(22), 5765-5771.



Time-of-Flight Three
Dimensional
Neutron Diffraction in
Transmission Mode

Scientific Reports, 2017, 7(1), 9561.

Introduction & Motivation

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- **Detailed knowledge of the crystallographic grain network, not only on the surface but also in the bulk of samples, is essential for tuning and manipulating macroscopic properties for applications.**
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 - **2D and 3D,**
 - **Destructive and non-destructive,**
 - **Optical light, electrons, X-rays, neutrons**

Laue 3D neutron diffraction tomography

6D X-ray & neutron diffraction

Laue 3D Neutron Diffraction Tomography

Characteristics

FALCON - E11 - HZB



- Not wavelength resolved ----> High Flux (also at continuous sources)
- Fast acquisition time (a few hours even minutes vs days)
- Use of simple Laue setup; easy to perform

Comparison with Cold neutron diffraction contrast tomography:

Monochromatic
neutron beam with 3
full rotations



+ Flux
+ Information
→

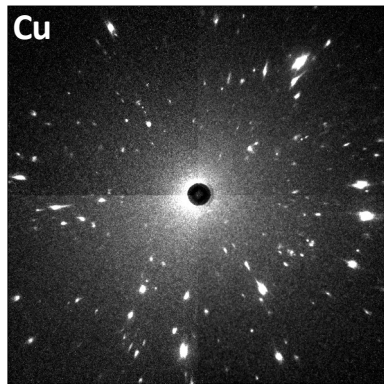
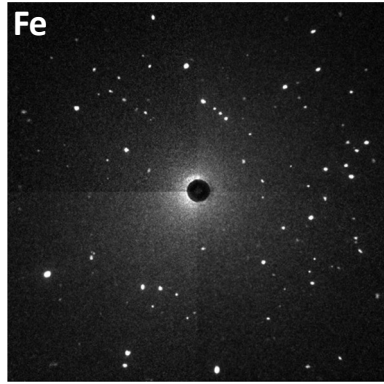
White THERMAL
neutron beam with a
single rotation



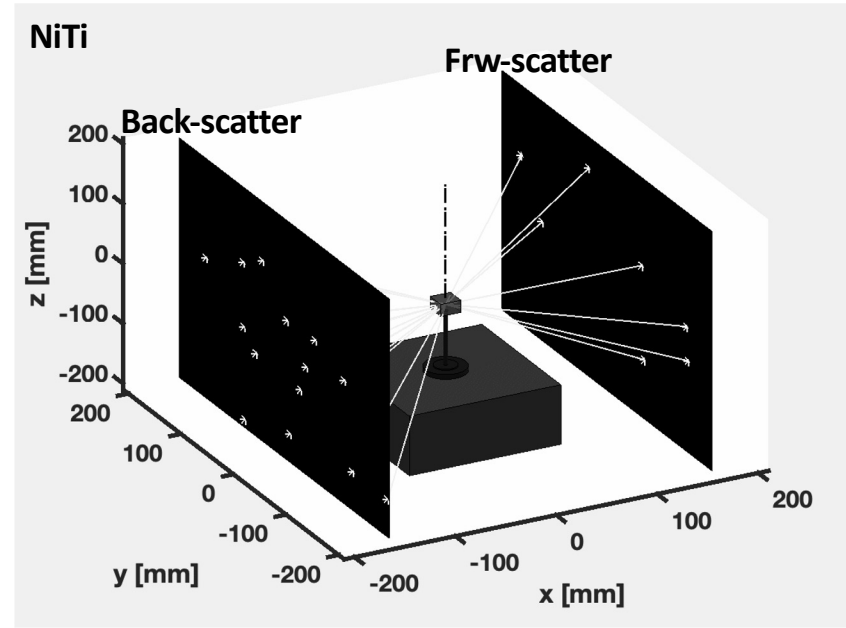
FALCON - E11 - HZB



Characteristics



Tomographic Measurements



Characteristics

In a diffraction experiment the main equation that provides information is the Bragg equation:

$$2d \cdot \sin\theta = n\lambda$$

Cubic:

$$\frac{1}{d^2} = \frac{h^2 + k^2 + l^2}{a^2}$$

Tetragonal:

$$\frac{1}{d^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2}$$

Hexagonal:

$$\frac{1}{d^2} = \frac{4}{3} \left(\frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}$$

In **polychromatic** case ($\lambda_1 < \lambda < \lambda_2$), the recorded data is integrated over the full wavelength range available and thus information is lost.

~~$$2d \cdot \sin\theta = n\lambda$$~~

Laue 3D Neutron Diffraction Tomography

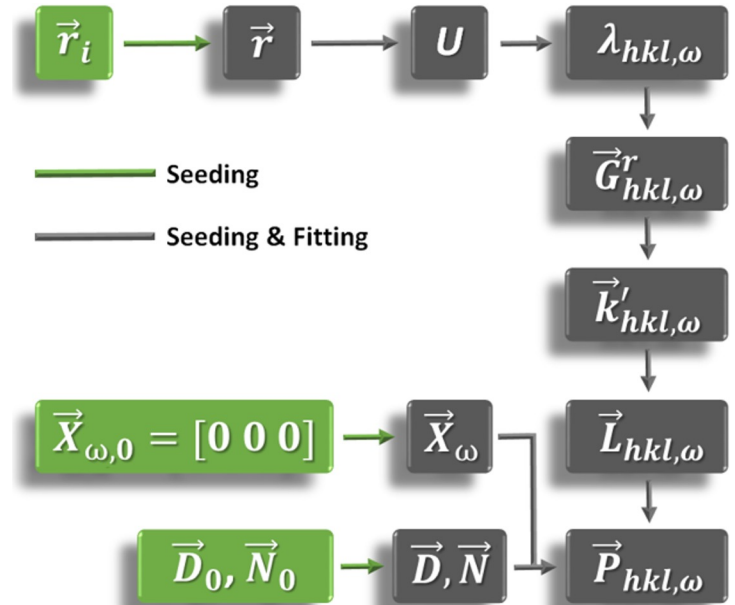
Characteristics

How to solve the problem and find potential solutions ???

Use of a *forward model* !!

What is required?

- Sample information: lattice parameters and space group that will be used with Mantid to generate the samples *hkl* planes and the corresponding **d-spacing** (also the multiplicity and structure factors squared).
- Possible **grain orientations**, originally input as **Rodrigues vectors**, r , and then transformed into **orientation matrices**, U , (total number of orientations are user defined but the values are confined by the fundamental zone of the crystal system).
- Possible **grain position**, originally assumed at (0,0,0) and refined later.
- The **omega range**, **detector parameters**, and **wavelength range**.



Original Development

SCIENTIFIC REPORTS

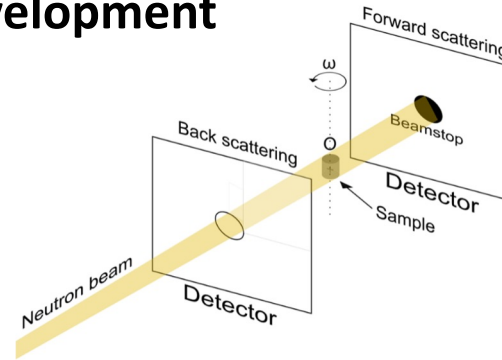
Article | [Open Access](#) | Published: 18 March 2019

Laue three dimensional neutron diffraction

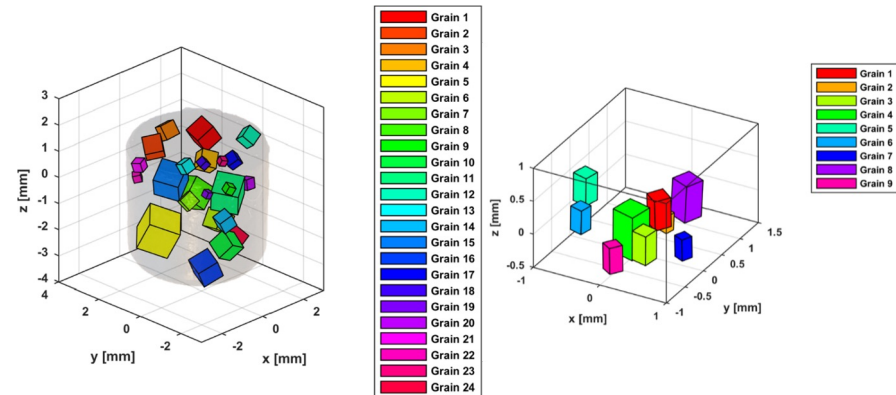
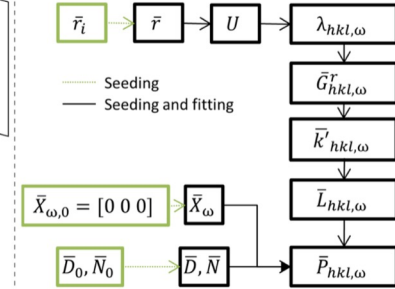
Marc Raventós , Michael Tovar, Marisa Medarde, Tian Shang, Markus Strobl, Stavros Samothrakitis, Ekaterina Pomjakushina, Christian Grünzweig & Søren Schmidt 

Scientific Reports **9**, Article number: 4798 (2019) | [Cite this article](#)

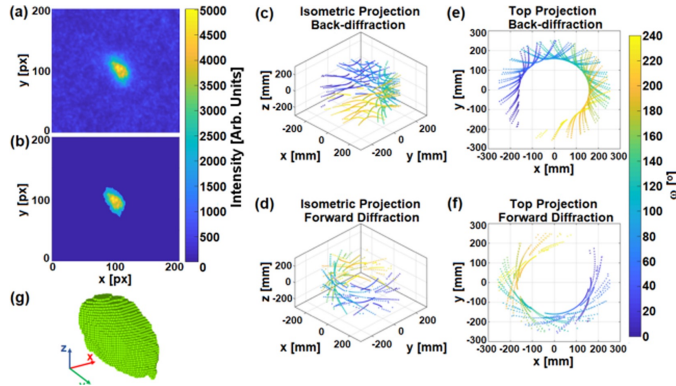
901 Accesses | [3](#) Altmetric | [Metrics](#)



Forward model flowchart



Grain Morphology Reconstruction

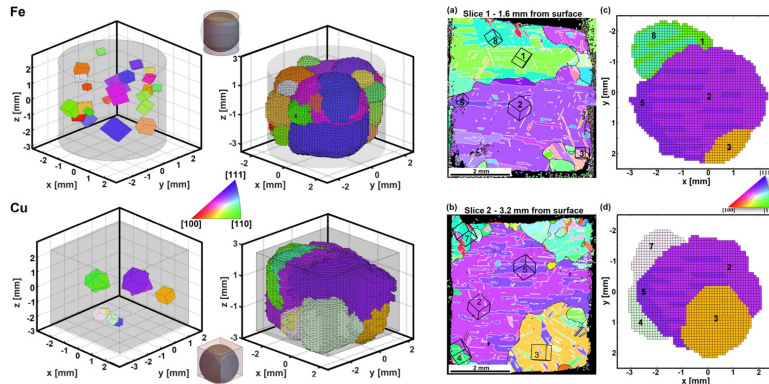


ASTRA toolbox

+



+



SCIENTIFIC REPORTS

Article | [Open Access](#) | Published: 28 February 2020

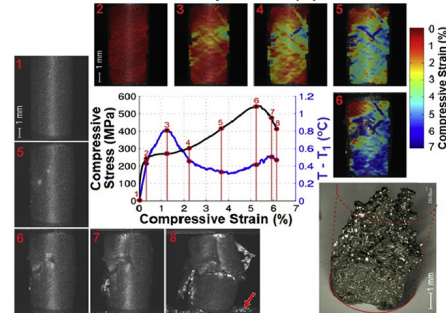
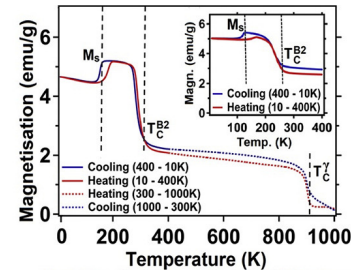
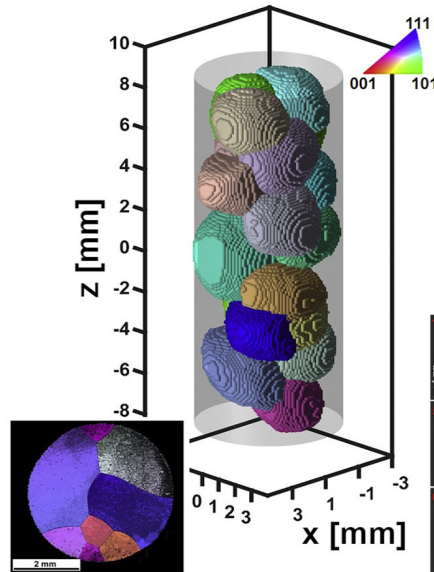
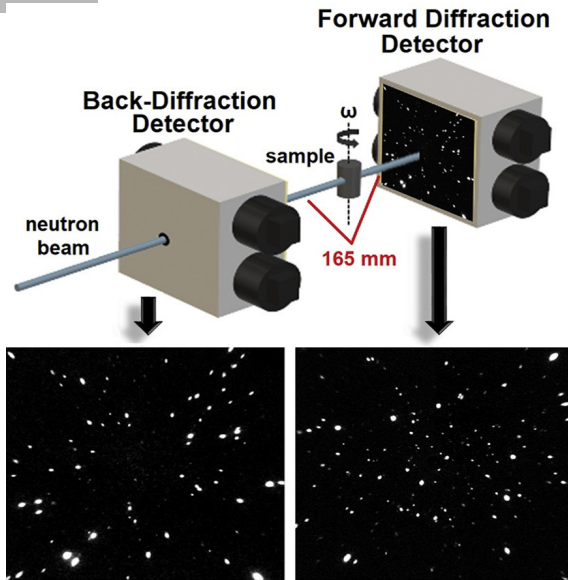
Grain morphology reconstruction of crystalline materials from Laue three-dimensional neutron diffraction tomography

Stavros Samothrakitis, Marc Raventós, Jan Čapek, Camilla Buhl Larsen, Christian Grünzweig, Michael Tovar, Marina Garcia-Gonzalez, Jaromír Kopeček, Søren Schmidt ✉ & Markus Strobl ✉

Scientific Reports **10**, Article number: 3724 (2020) | [Cite this article](#)

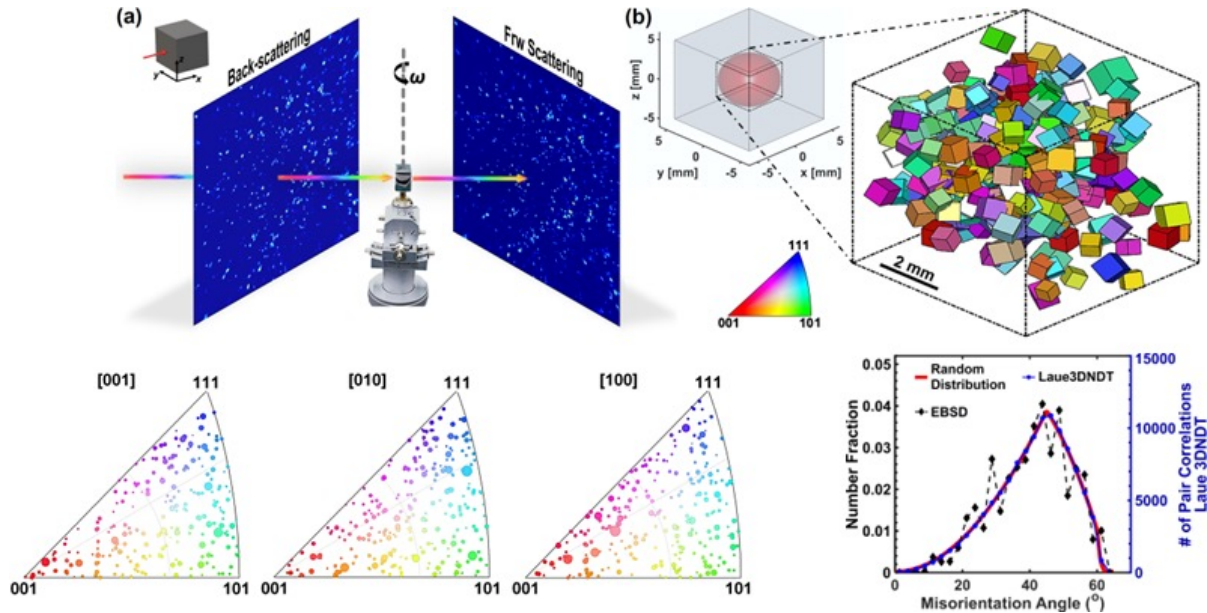
Applications

A multiscale study of hot-extruded CoNiGa ferromagnetic shape-memory alloys



Applications

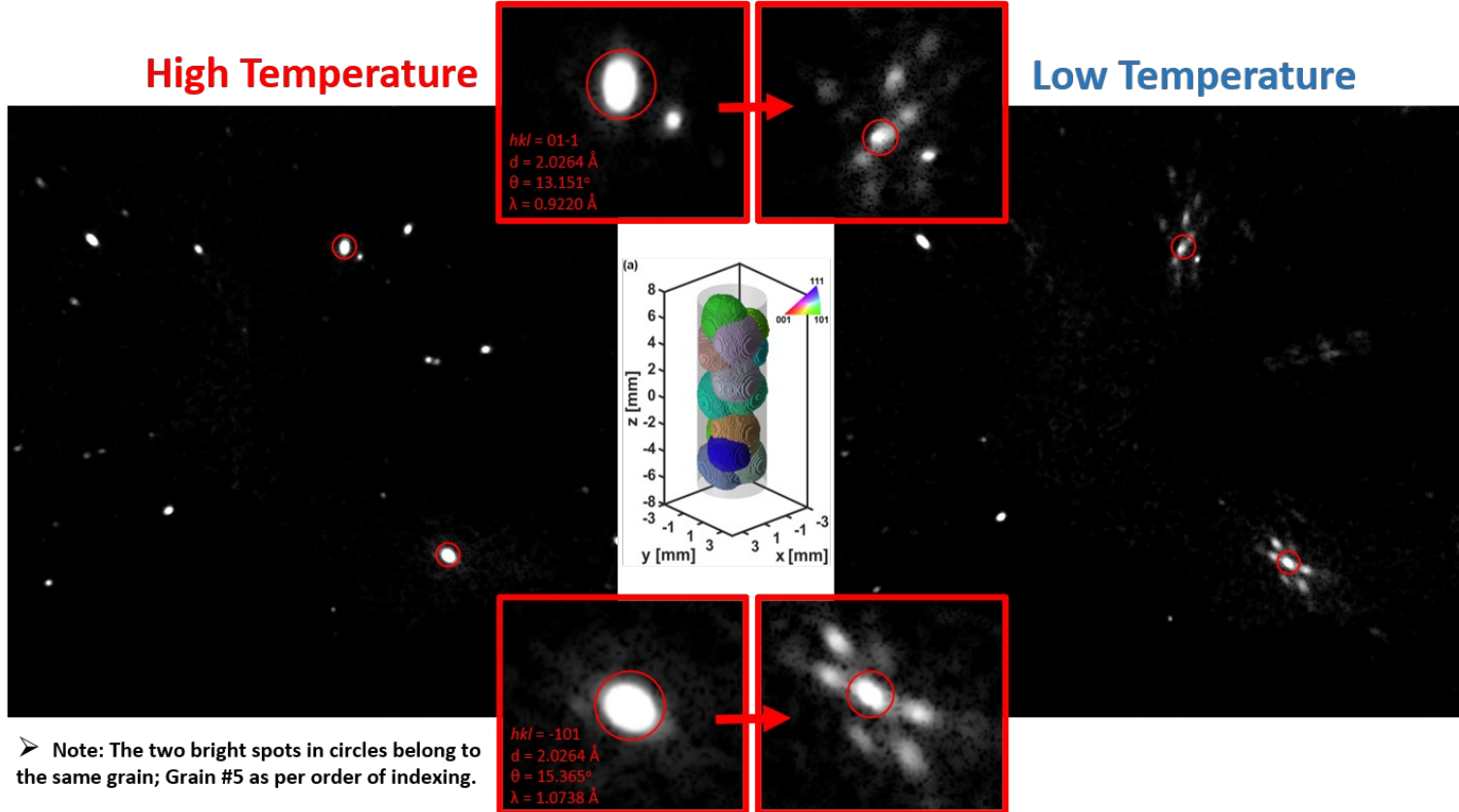
Microstructural characterization through grain orientation mapping with Laue three-dimensional neutron diffraction tomography



Applications

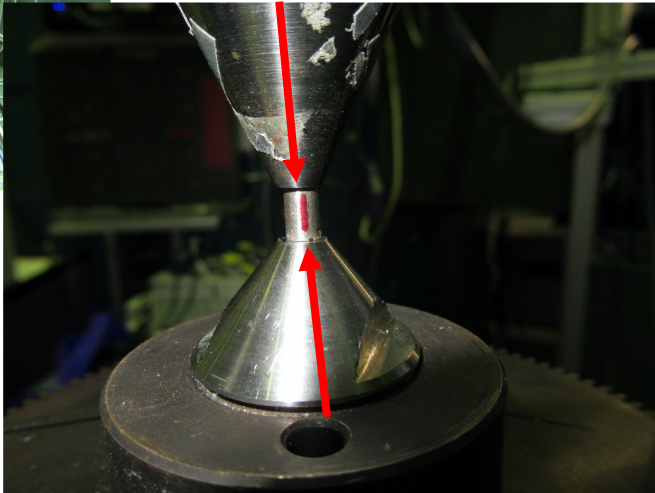
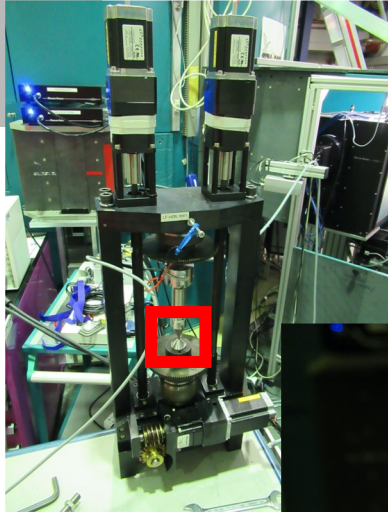
High Temperature

Low Temperature



Laue 3D Neutron Diffraction Tomography

Applications



Journal of
**Applied
Crystallography**
ISSN 0021-8898

Received 9 September 2009
Accepted 7 April 2010

Determining grain resolved stresses in polycrystalline materials using three-dimensional X-ray diffraction

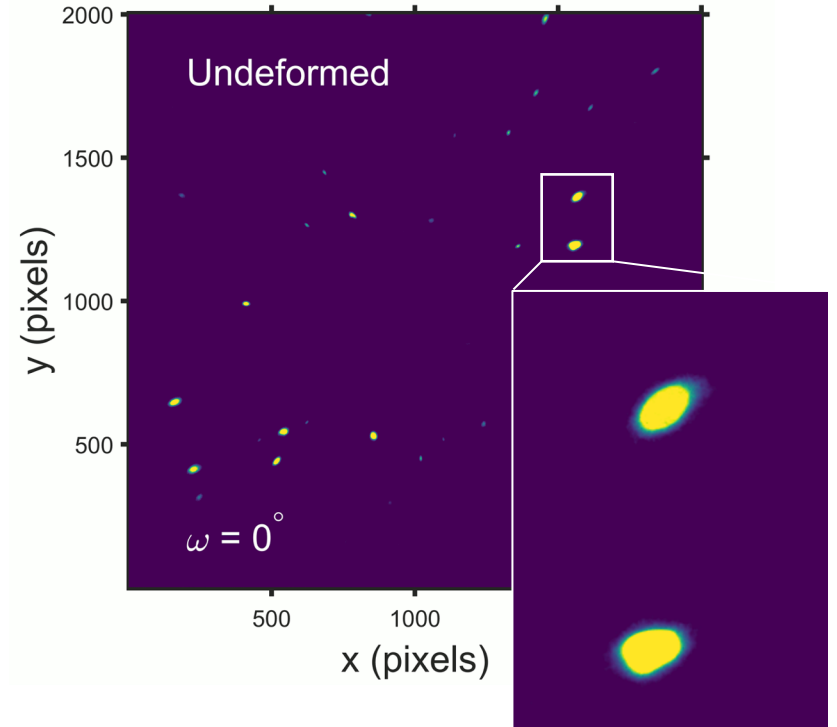
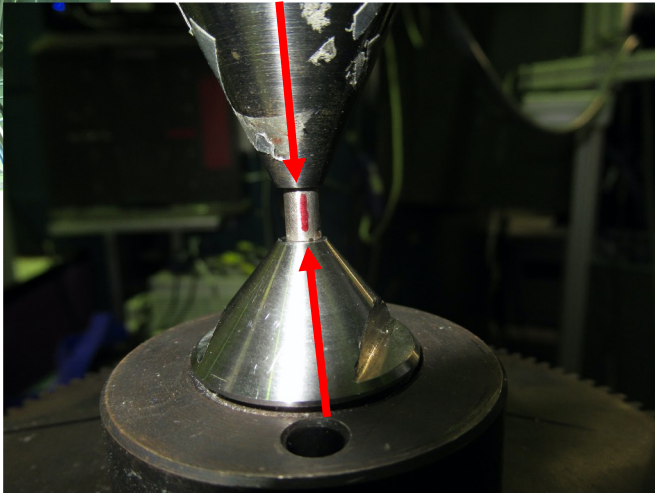
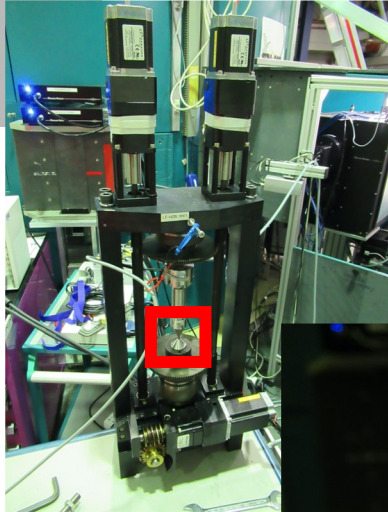
Jette Oddershede,^{a*} Søren Schmidt,^a Henning Friis Poulsen,^a Henning Osholm Sørensen,^a Jonathan Wright^b and Walter Reimers^c

^aCentre for Fundamental Research: 'Metal Structures in Four Dimensions', Riso DTU, DK-4000 Roskilde, Denmark, ^bESRF, Grenoble, France, and ^cTU Berlin, Germany. Correspondence e-mail: jeto@risoe.dtu.dk

$$\frac{\Gamma_{ij}^{-1} \bar{G}_{ij}}{\text{Experimental}} = \frac{\lambda}{2\pi} \frac{U_i B_i \bar{G}_{hkl,ij}}{\text{Theoretical}}$$

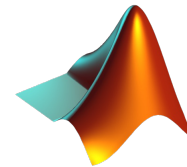
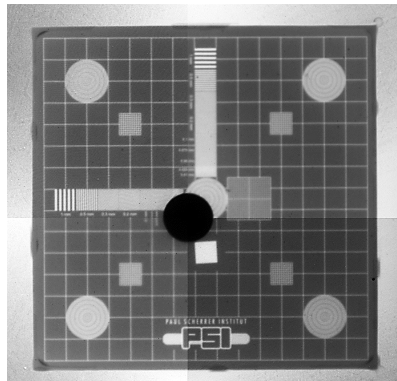
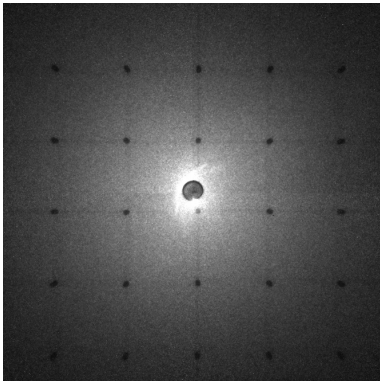
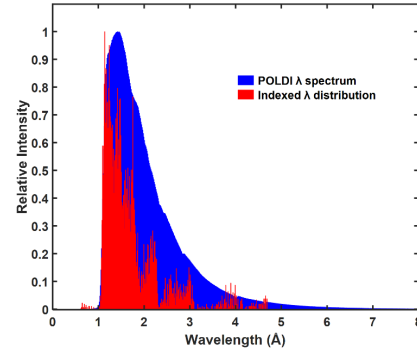
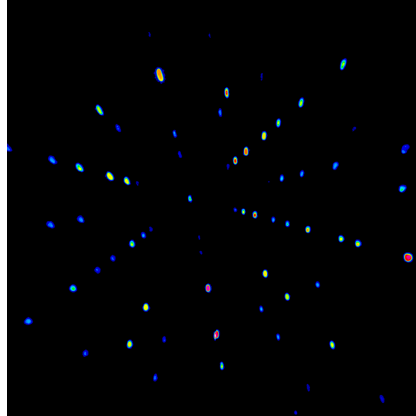
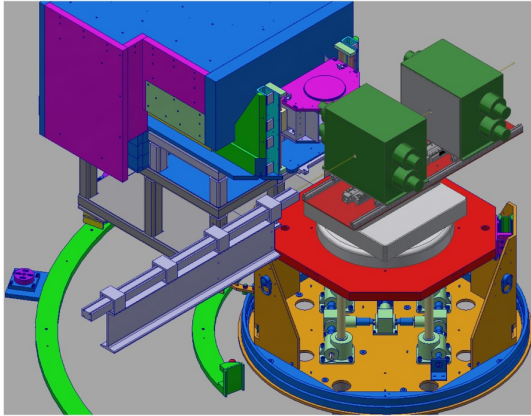
Laue 3D Neutron Diffraction Tomography

Applications



Laue 3D Neutron Diffraction Tomography

Future @ PSI

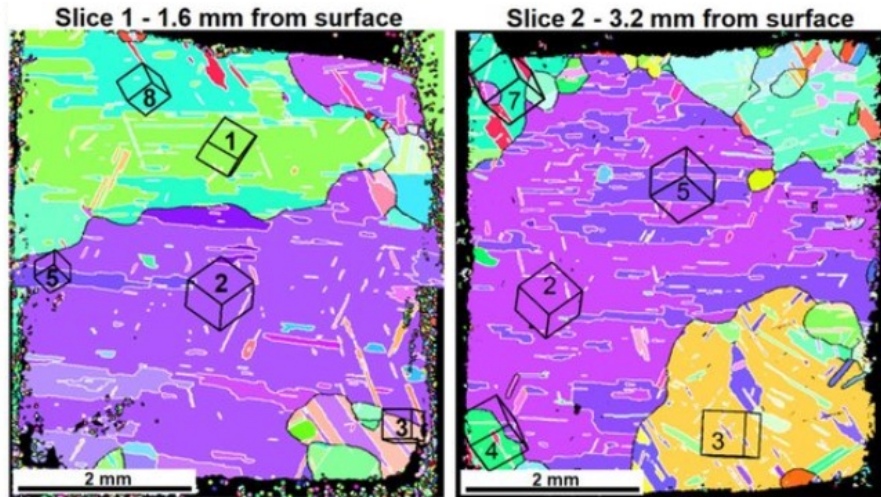


Funding: SDSC 4th call "Robust and scalable Machine Learning algorithms for Laue 3-Dimensional Neutron Diffraction Tomography" (2021).

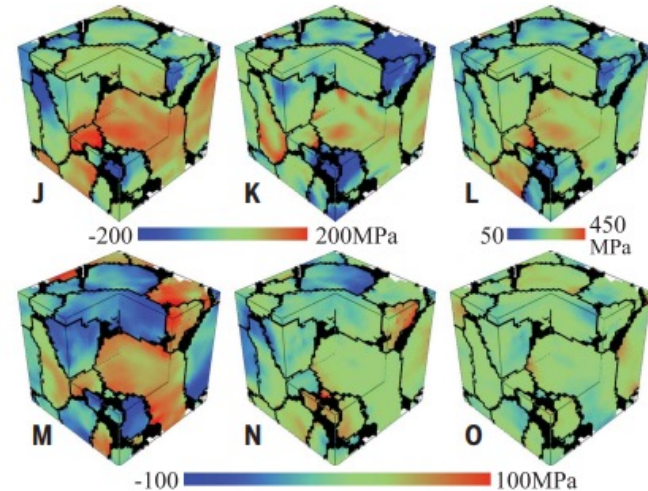
Future Development: 6DXND Tomography

6DXND: 6-Dimensional X-ray and Neutron diffraction

- Why 6D: 3D (direct space) + 3D (orientation space)
- Motivation: Grain-resolved → subgrain spatial resolution
- Potentially better spatial resolution than Laue 3DNDT, but requires more experimental information (wavelength)



Adapted from [S. Samothrakitis et al., *Scientific Reports* 10, 3724 (2020)]

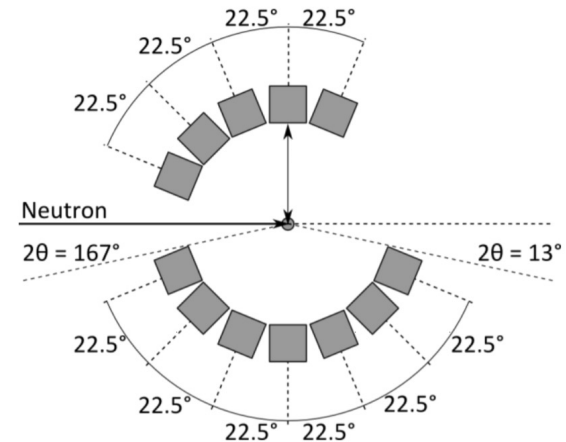
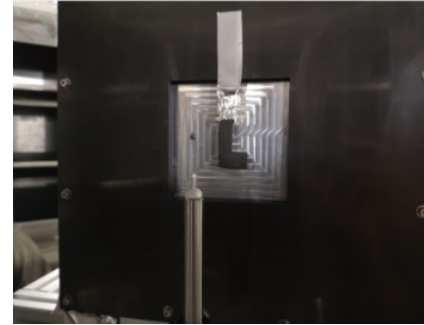
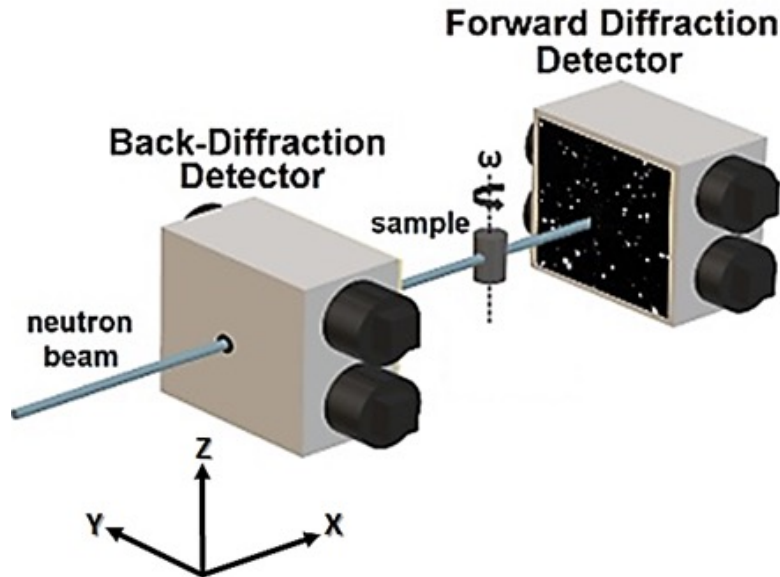


From [Y. Hayashi et al., *Science* 366, 1492 (2019)]

6DXND Tomography: Experimental setup

Similar experimental setup to Laue 3DNDT

- Sample fully illuminated by beam
- Collect diffraction images at different sample rotations ω



High resolution orientation distribution function (HRODF)

A discrete spherical x-ray transform of orientation distribution functions using bounding cubes

I G Kazantsev, S Schmidt and H F Poulsen

Center for Fundamental Research: 'Metal structures in four dimensions', RISØ National Laboratory for Sustainable Energy, Technical University of Denmark, Roskilde, DK-4000, Denmark

E-mail: kazantsev1@netzero.net

Received 15 November 2008, in final form 9 August 2009

Published 16 September 2009

Online at stacks.iop.org/IP/25/105009

Abstract

We investigate a cubed sphere parametrization of orientation space with the aim of constructing a discrete voxelized version of the spherical x-ray transform. For tracing the propagation of a unit great circle through the \mathbb{S}^3 partition subsets, the frustums of the cubed sphere, a fast procedure is proposed. The circle's parts in each frustum are gnomonically mapped into line segments inside the bounding cubes. The line segments constitute a convex polygon with vertexes indicating frustum exit–entry points. Thus the problem of system matrix calculation is reduced to the tracing of line segments within rectangular voxel arrays partitioning the bounding cubes. Hence algebraic reconstruction techniques can be used in a comprehensive way for orientation distribution function estimation from diffraction data.

Materials Science Forum Vols. 702-703 (2012) pp 536-539
 Online available since 2011/Dec/06 at www.scientific.net
 © (2012) Trans Tech Publications, Switzerland
 doi:10.4028/www.scientific.net/MSF.702-703.536

HIGH RESOLUTION ORIENTATION DISTRIBUTION FUNCTION

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 Bernd Dammann^{2,c}, and Ivan G. Kazantsev^{3,d}

¹Materials Research Division, Risø DTU, Technical University of Denmark, Denmark.

²Dept. of Informatics and Mathematical Modelling, Technical University of Denmark, Denmark.

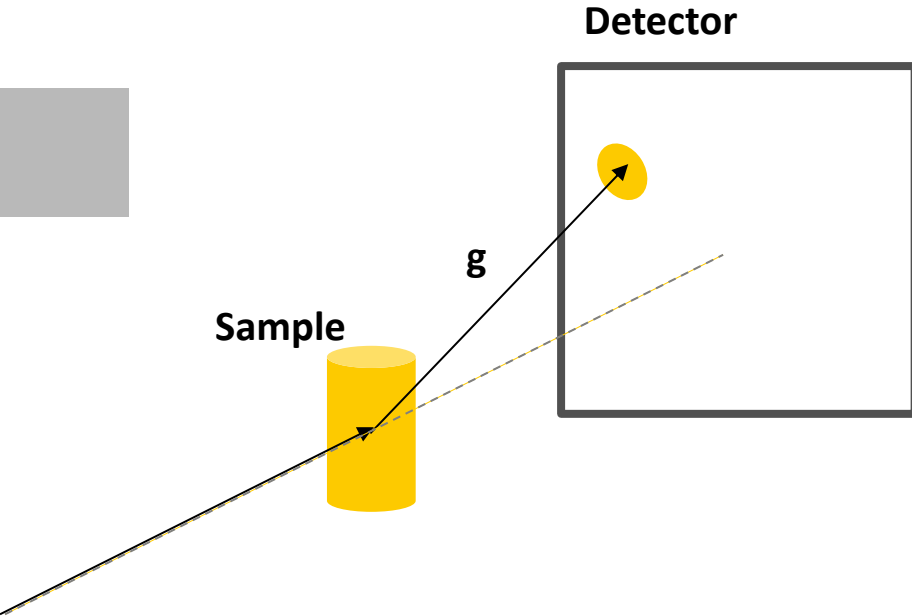
³Institute of Computational Mathematics and Mathematical Geophysics, 630090 Novosibirsk, Russia.

^assch@risoe.dtu.dk, ^bnfga@imm.dtu.dk, ^cbd@imm.dtu.dk, ^dkazantsev.ivan6@gmail.com

Keywords: ODF, x-ray, high resolution, inverse problems.

Abstract. A new method for reconstructing a High Resolution Orientation Distribution Function (HRODF) from X-ray diffraction data is presented. It is shown that the method is capable of accom-

6DXND: Mathematical formalism



g: experimental scatter vector
h: theoretical (unrotated) scatter vector

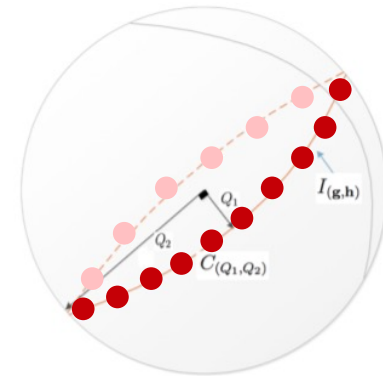
Determine orientation **q** such that h maps onto g

Use quaternion formalism:

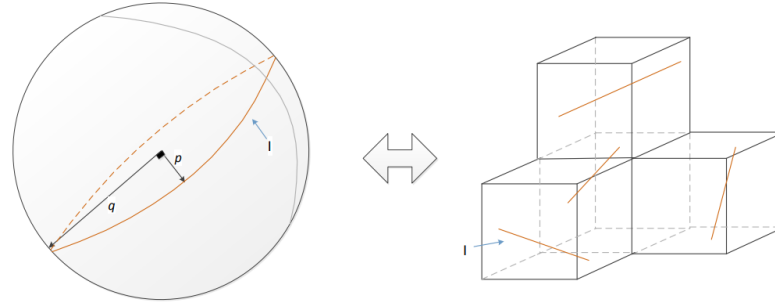
$$\mathbf{q} = (q_0; \mathbf{q}) = \left(\cos \frac{\alpha}{2}; \vec{\mathbf{n}} \sin \frac{\alpha}{2} \right)$$

Set of orientations is a sphere:

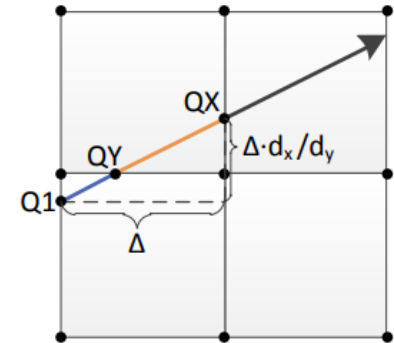
$$|\mathbf{q}| = \sqrt{q_0^2 + |\vec{\mathbf{q}}|^2} = 1$$



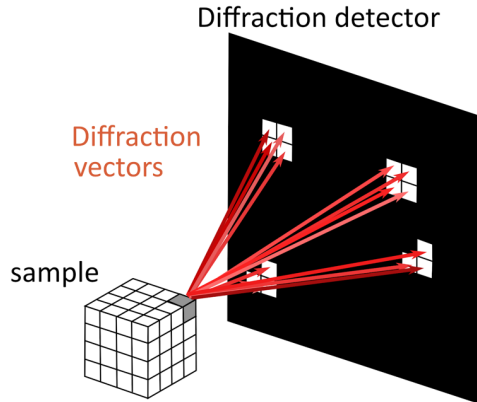
6DXND: Mathematical formalism



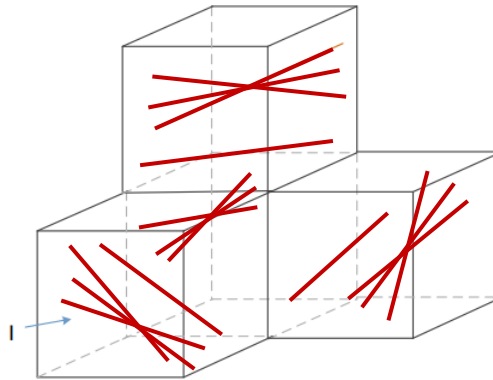
- **Set of solutions can be represented as straight lines**
 - **Problem reduces to solving a set of linear equations**
 - **Can be solved fast on gpus**
 - **Orientation space is discretized, and possible solutions explored through raytracing**



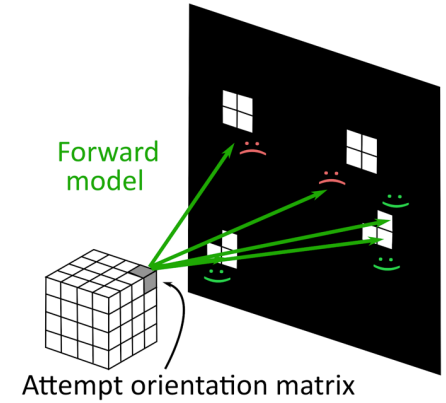
Step 1: Compute diffraction vectors



Step 2: Determine candidate orientations



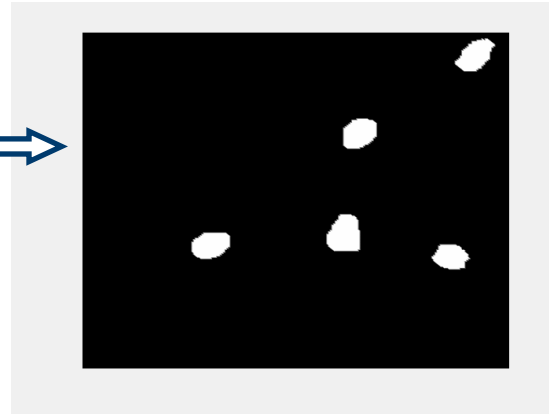
Step 3: Completeness check



6DXND: Input and Output

Input: Binarized list of lit pixels per wavelength

ω	x	y
0	1	1
0	0	1
0	0	2
0	0	3
0	0	2
0	0	3
0	0	4
\vdots	\vdots	\vdots
ω_N	N	N



Output: List of orientations and completeness per voxel

```

VOXEL (0, 0, 0) COUNT 1
Orientation:
[[ 0.9158573, -0.0323292, 0.4002003]
 [ -0.3969138, 0.0773878, 0.9145876]
 [ -0.0605385, -0.9964768, 0.0580442]]
Completeness: 99.07%
VOXEL (1, 0, 0) COUNT 1
Orientation:
[[ 0.9158573, -0.0323292, 0.4002003]
 [ -0.3969138, 0.0773878, 0.9145876]
 [ -0.0605385, -0.9964768, 0.0580442]]
Completeness: 99.07%
  
```

6DXND: Conclusions and outlook

6DXND advantages:

- Sub-grain level spatial resolution
- Spot overlap not issue
- Fast – Simulations shown take ~3 h on single graphic card (rtx 2060) ~15 min on small cluster

To be done:

- Robustness tests and further developed visualization tools

0.1 deg step:



5 deg step:



Acknowledgments



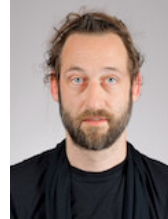
EUROPEAN
SPALLATION
SOURCE



S. Schmidt



R. Woracek



M. Strobl



E. Polatidis



J. Capek



M. Raventos



T. Kacprzak



M. Tovar

Thank you!

Questions?

